

Vorlesung Batterie- und Brennstoffzellensysteme

(Langzeit-) Stabilität von SOFC-Komponenten

André Weber

Institut für Angewandte Materialien – Elektrochemische Technologien (IAM-ET)

Adenauerring 20b, Geb. 50.40 (FZU), Raum 314

phone: 0721/608-7572, fax: 0721/608-7492

andre.weber@kit.edu

www.iam.kit.edu/et



Basic Concepts in Reliability Engineering

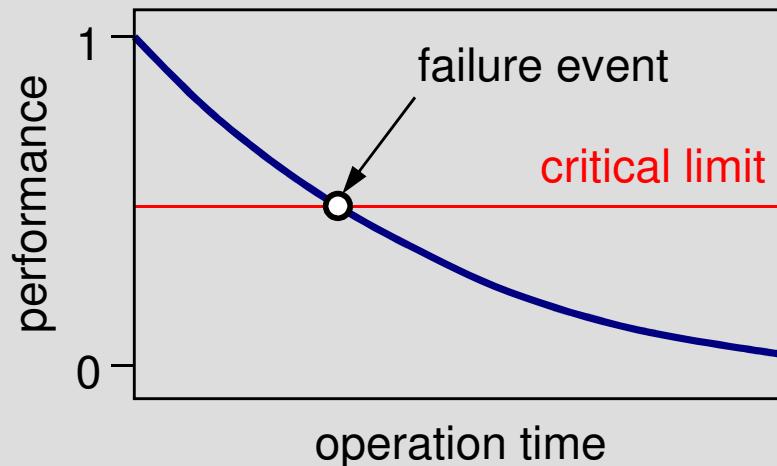
Definition of Failure Event and Time-To-Failure

soft failure

gradual performance loss until (pre-defined) critical level is reached.

example

- voltage/ power degradation in fuel cells



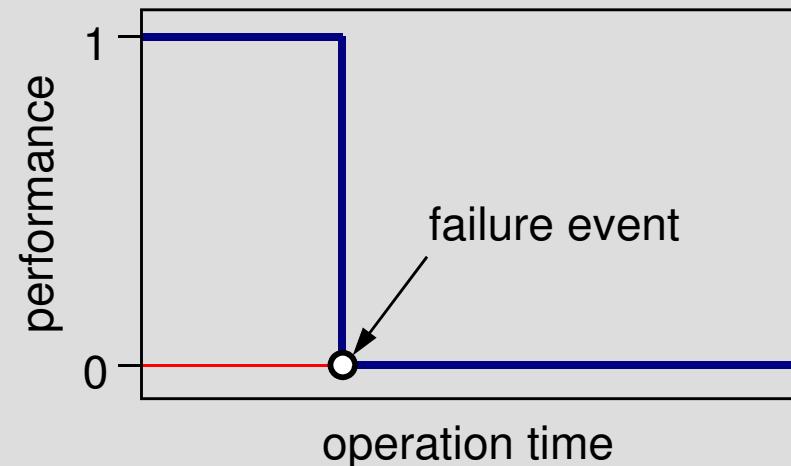
(Meeker et al., Technometrics, 40, 1998)

hard failure

sudden loss of functionality – the device stops working.

example

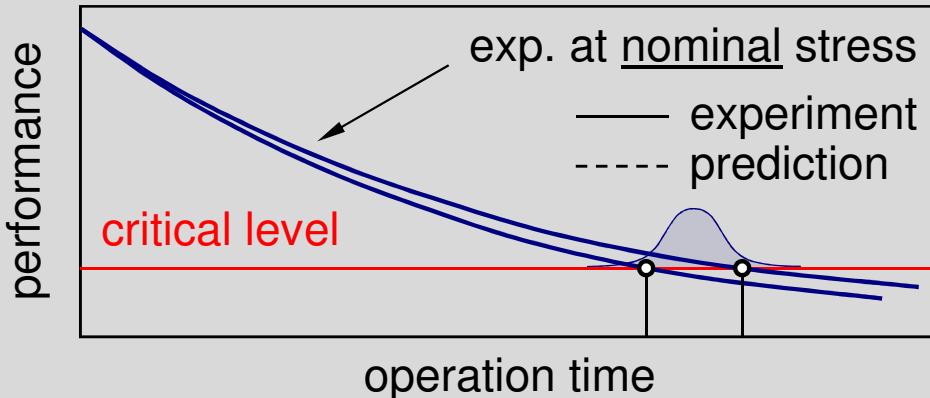
- light bulb burns out



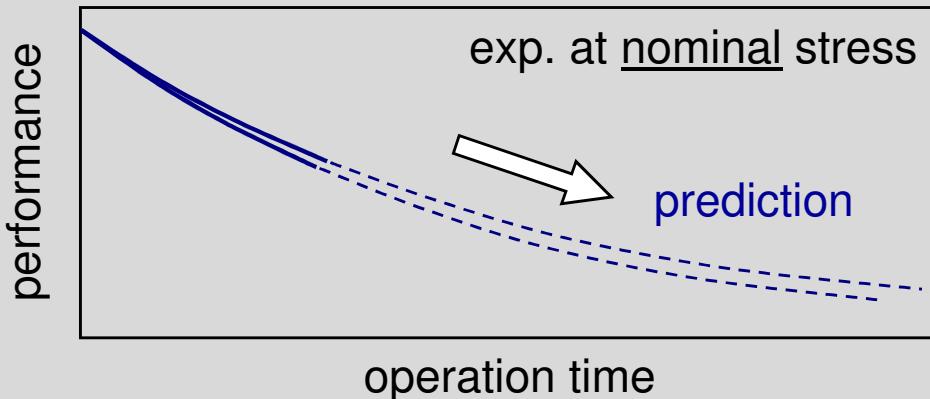
Lifetime and Degradation Assessment

Overview of Life Testing Methods

"conventional" life testing

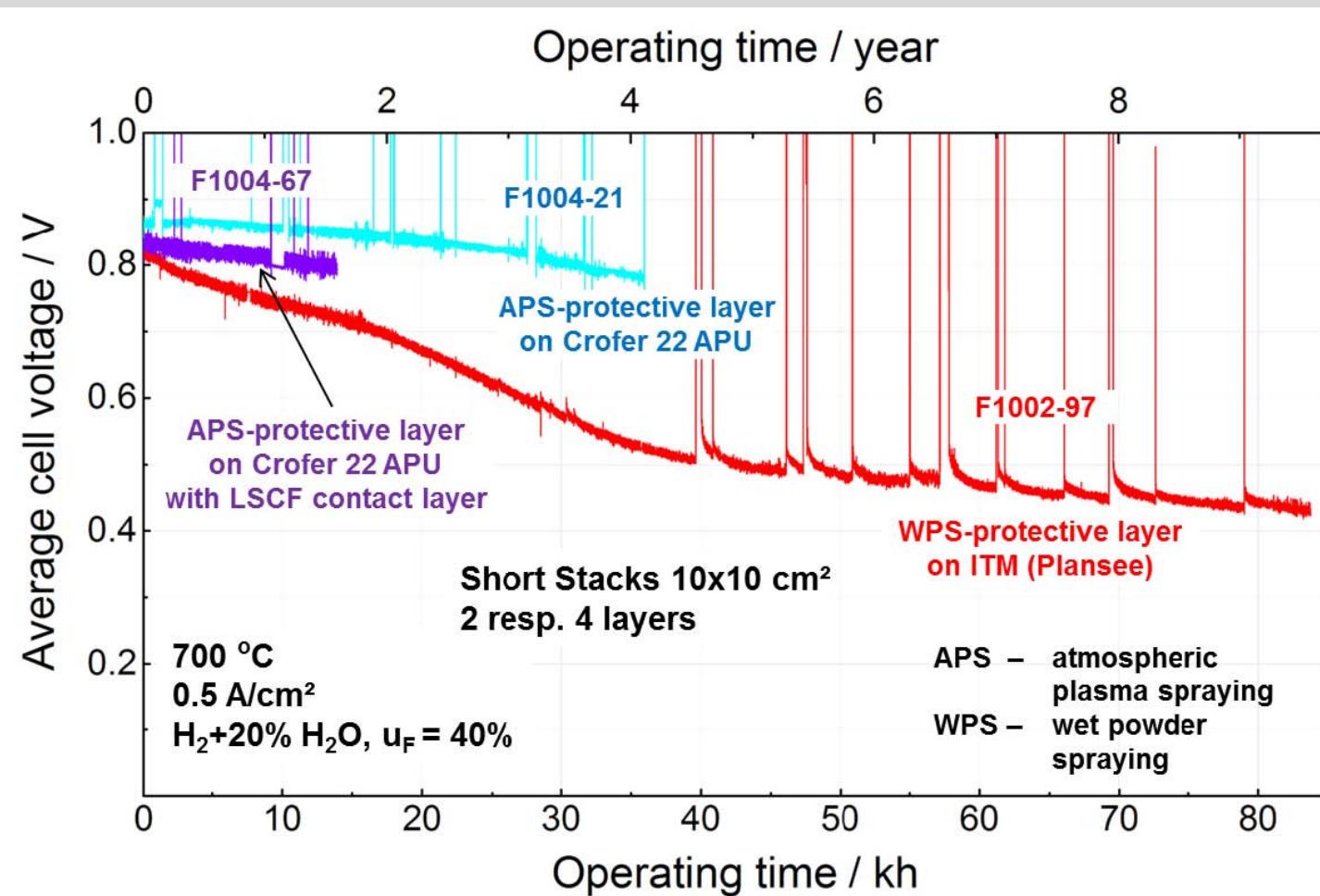


degradation testing



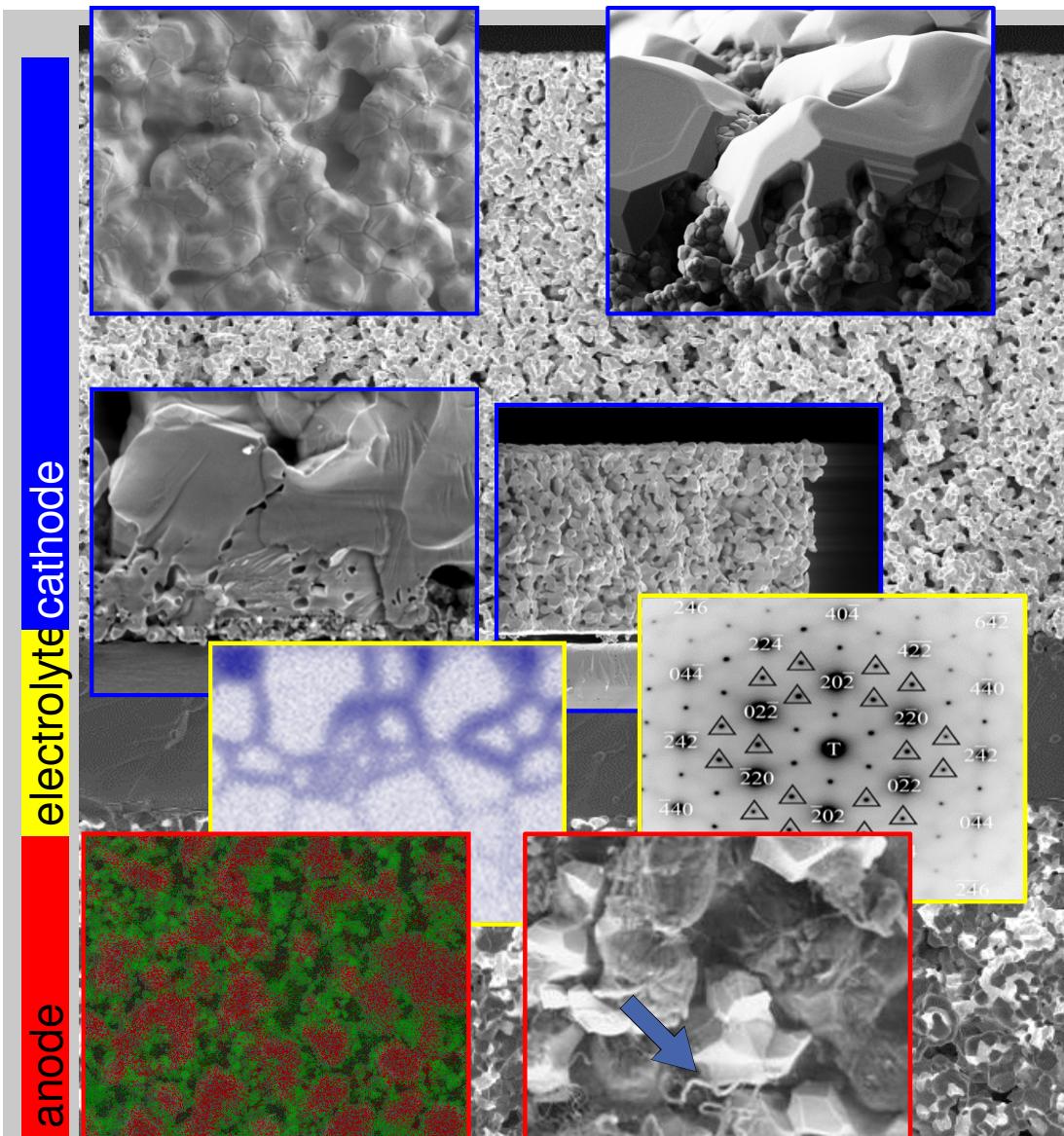
Durability Testing of Solid Oxide Cells

Research Center Jülich ASC-Stacks (2007 ff)



L. Blum et al., *ECS Trans.*, **78** (1), p. 1791 (2017)

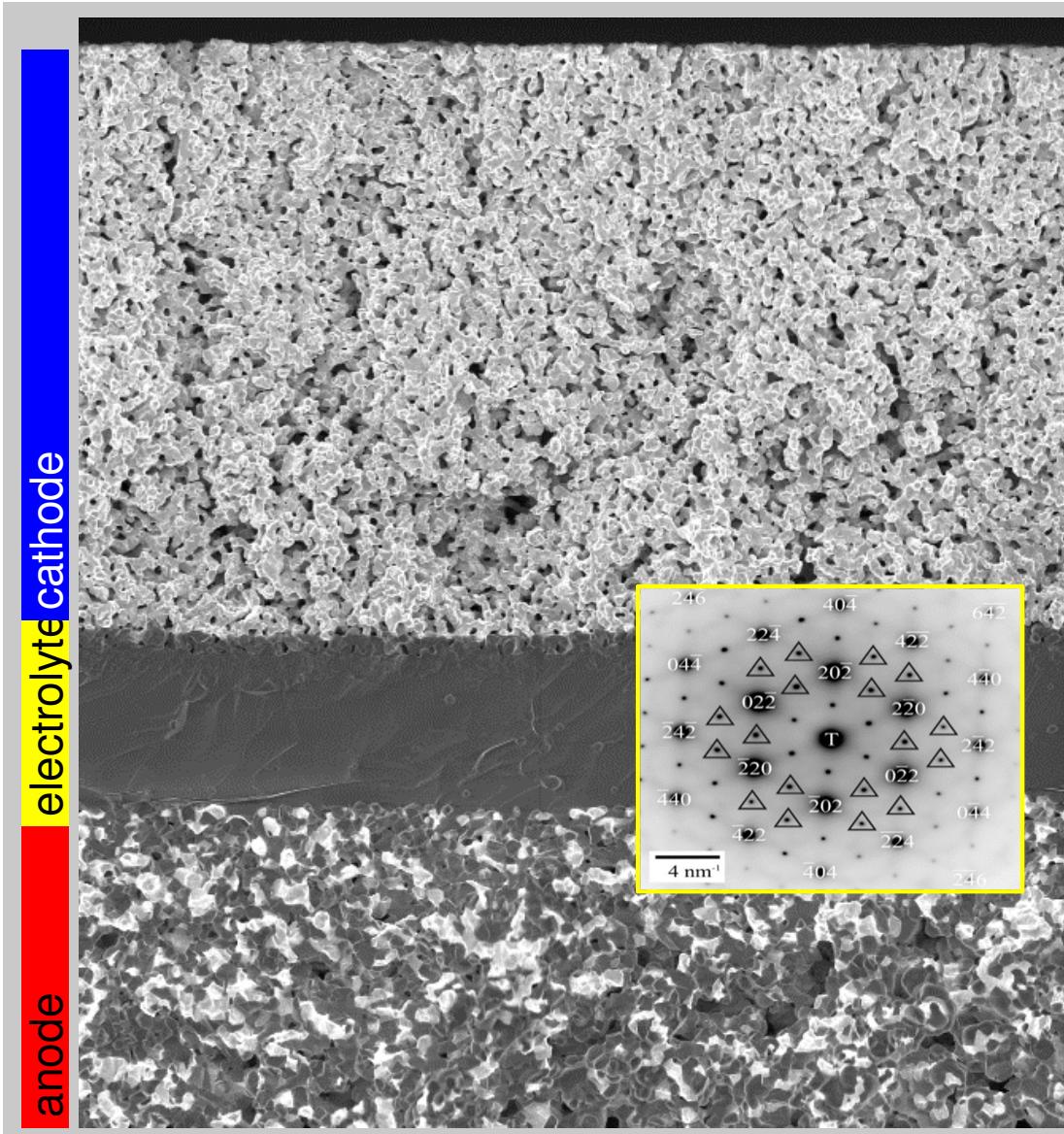
Degradation Processes in Solid Oxide Fuel Cells



in SOFC many potentially competing degradation mechanisms are known:

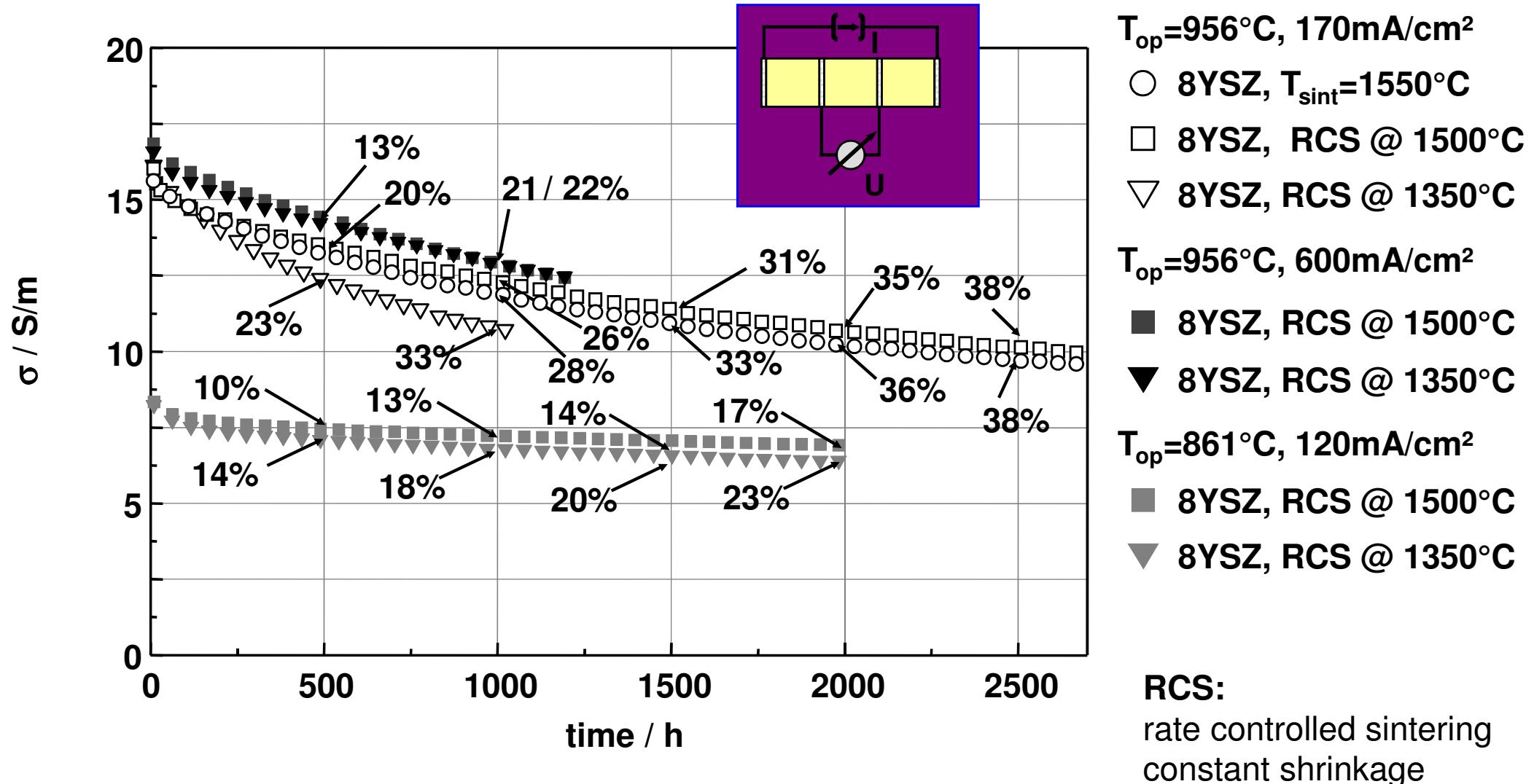
- densification of electrodes
 - formation of micropores
A. Weber et. al., *Denki Kagaku* **64**, pp. 582-589 (1996).
 - formation of chromium compounds on the cathode
 - delamination of the cathode
M. J. Heneka et. al., Proc. 9th Int. Symp. on SOFC, pp. 534-543 (2005).
 - Mn-interdiffusion
A. Weber, in J. Garche (Ed.), *Encyclopedia of Electrochemical Power Sources*, Amsterdam: Elsevier, pp. 120-134 (2009).
 - intrinsic electrolyte degradation
B. Butz et. al., *Solid State Ionics* **177**, pp. 3275-3284 (2006).
 - Ni-agglomeration
A. C. Müller et. al., Proc. 3rd European SOFC Forum, pp. 353-362 (1998).
 - carbon deposition
E. Ivers-Tiffée et. al., *Handbook of Fuel Cells – Fundamentals, Technology and Applications*, pp. 933-956 (2009).
- ⇒ in addition, activation processes that improve the performance take place
A. Weber et. al., *Denki Kagaku* **64**, pp. 582-589 (1996).

Degradation Processes in Solid Oxide Fuel Cells Electrolyte



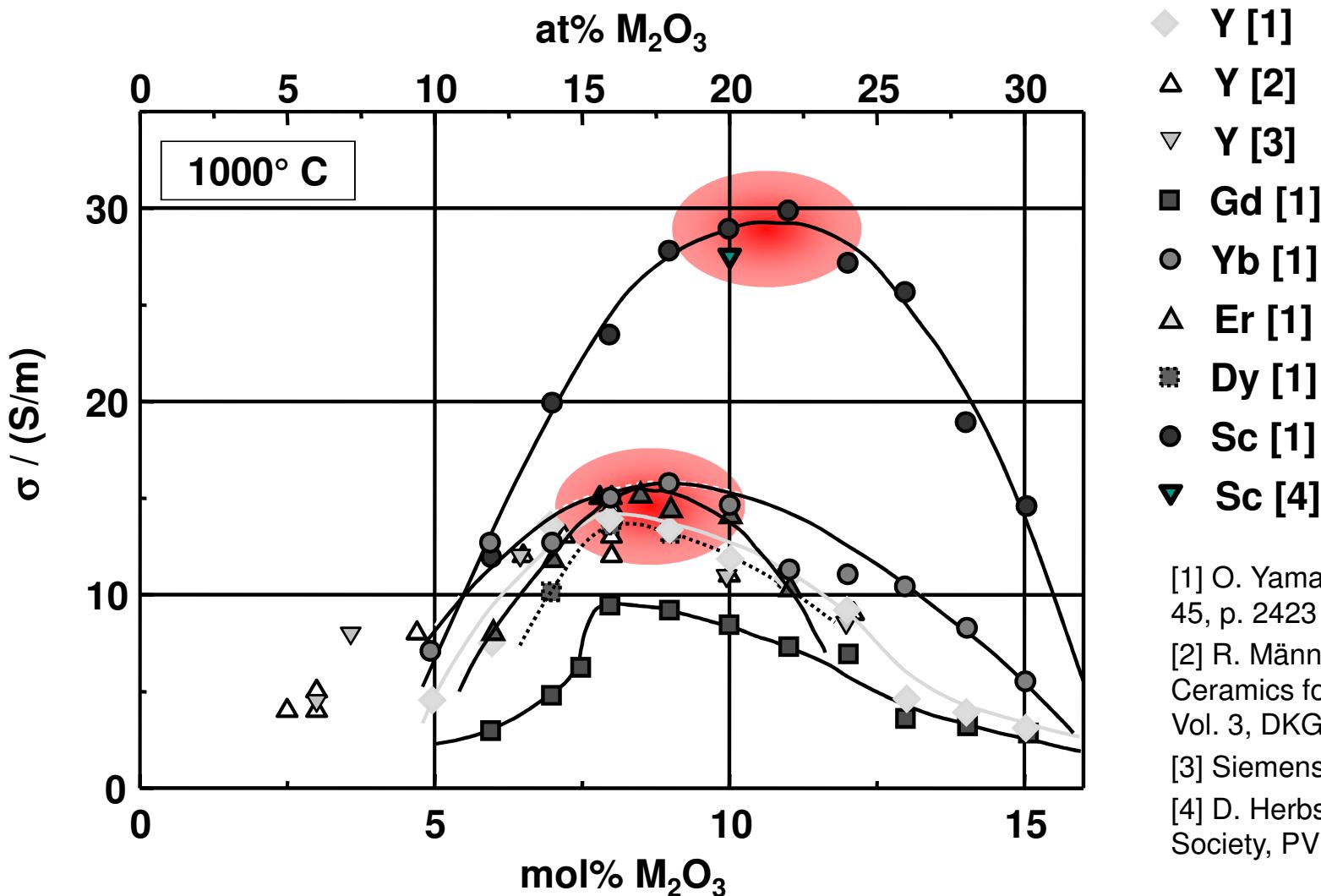
- intrinsic electrolyte degradation
B. Butz et. al., *Solid State Ionics* **177**, pp. 3275-3284 (2006).

Long term stability of ZrO₂ – Me₂O₃ Oxide Ion Conductors FSZ: 8YSZ



Electrical Conductivity of Fluorites ZrO₂ – Me₂O₃

Impact of various dopants on the conductivity of ZrO₂



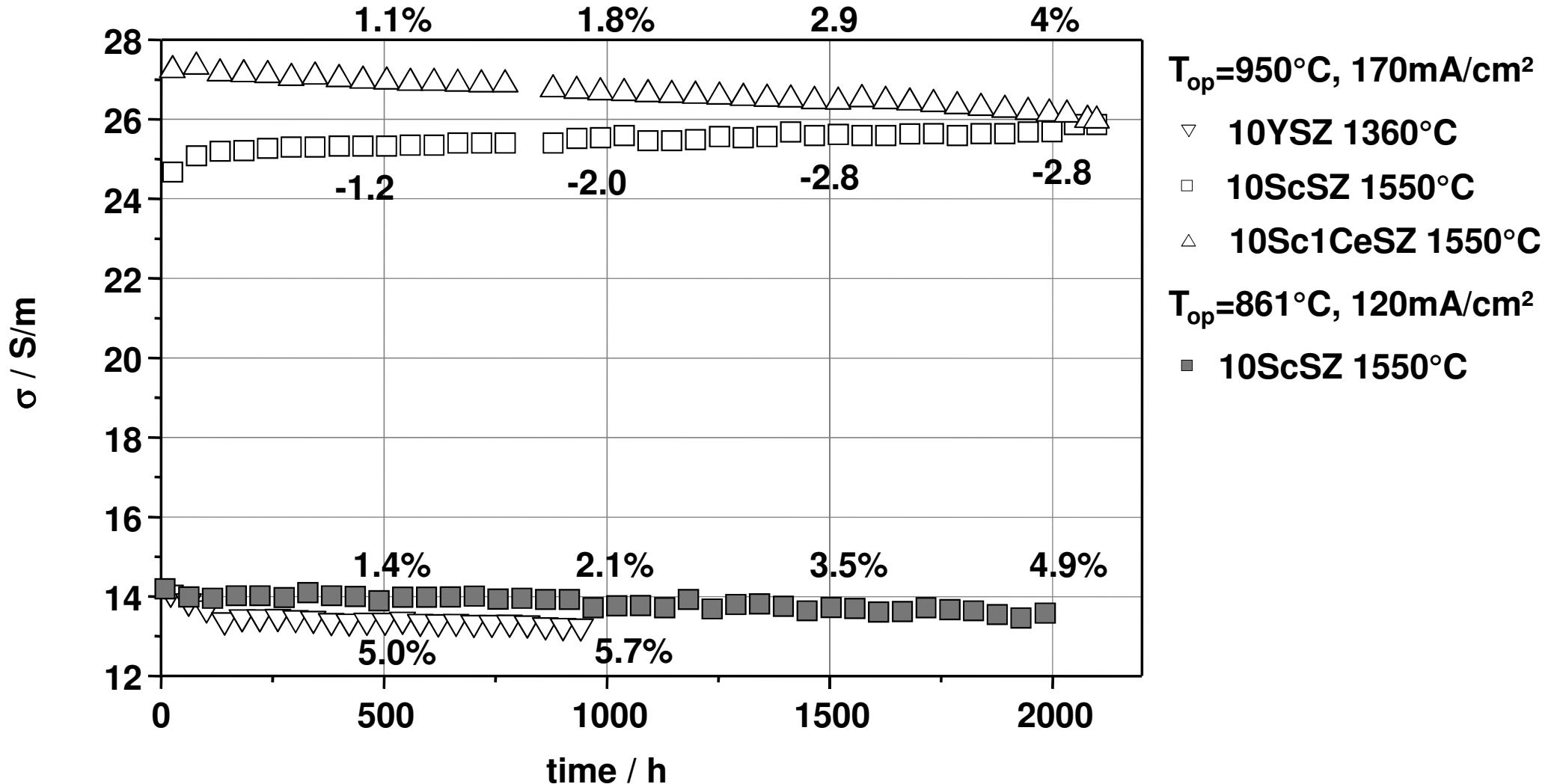
[1] O. Yamamoto, *Electrochimica Acta* 45, p. 2423 (2000)

[2] R. Männer, *Electroceramics and Ceramics for Special Applications*, Vol. 3, DKG (1992)

[3] Siemens, unpublished data

[4] D. Herbstritt, *The Electrochemical Society, PV 2001-16*, p. 349, (2001)

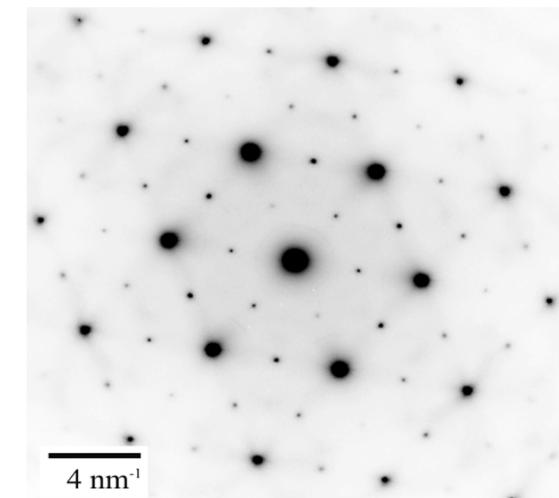
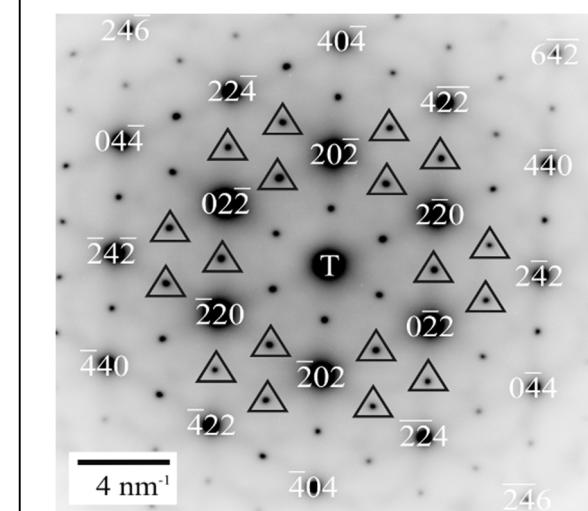
Long term stability of ZrO₂ – Me₂O₃ Oxide Ion Conductors FSZ: 10YSZ and 10ScSZ



Electron diffraction: Tetragonal Phase

[110], [111], [112], [013], [123]

- Three variants of tetragonal phase:
c-axis along all cubic $<100>$ -axes
 - Double diffraction in thicker sample regions
- Clear identification of tetragonal phase
(no superstructures!)



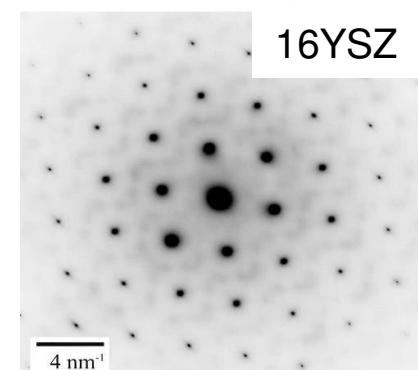
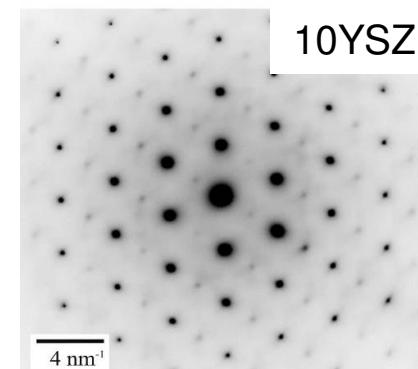
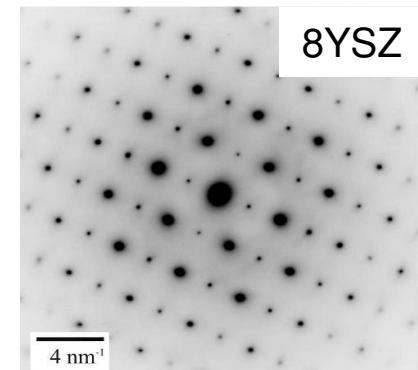
[111] 8YSZ



Electron diffraction

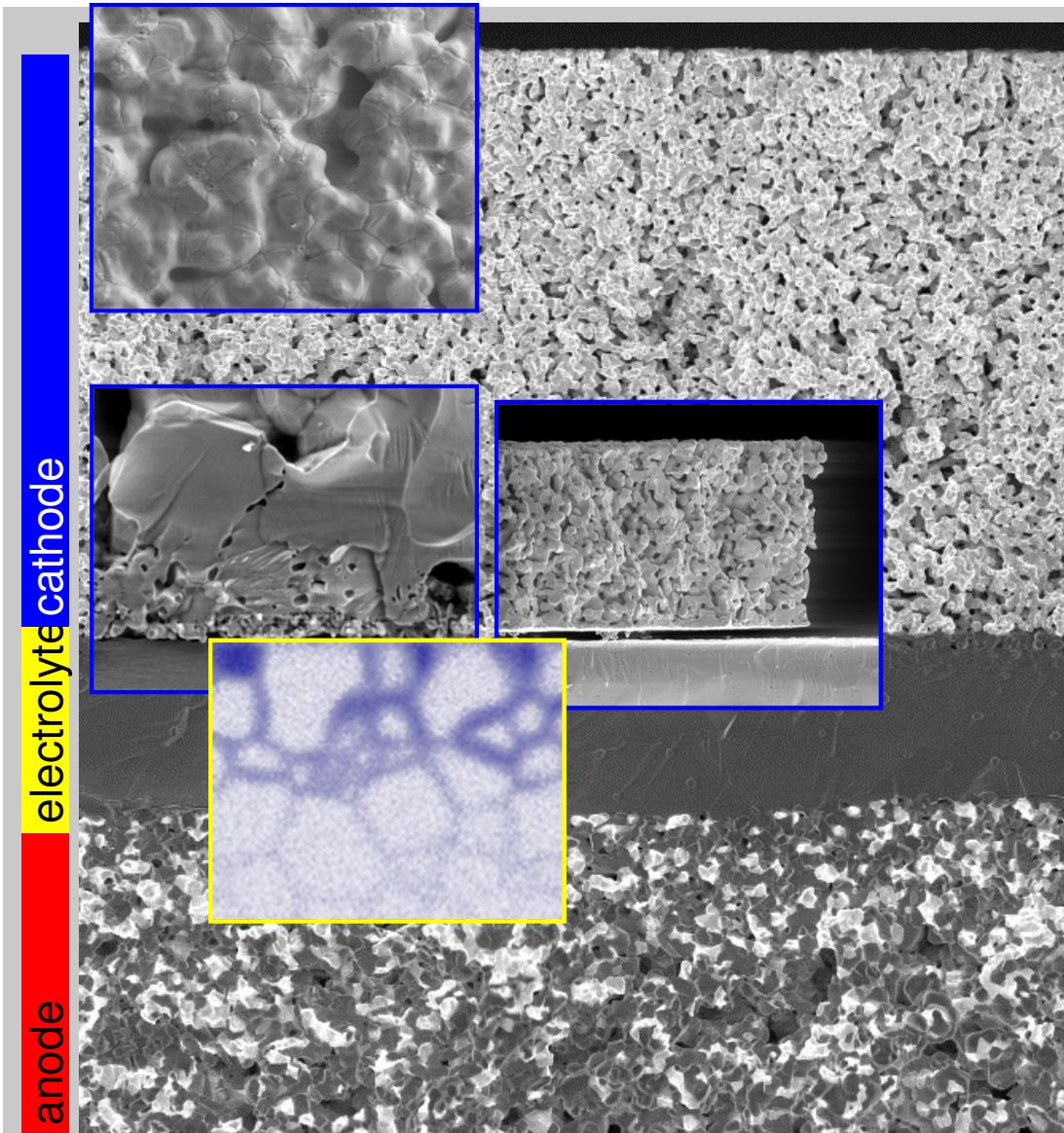
- Exists in 8YSZ and 10YSZ, not in 16YSZ
- Decreasing volume fraction with increasing doping concentration

→ Dark field images with tetragonal intensity



Degradation Processes in Solid Oxide Fuel Cells

Cathode & Cathode/Electrolyte-Interface



- densification of electrodes
- formation of micropores

A. Weber et. al., *Denki Kagaku* **64**, pp. 582-589 (1996).

- delamination of the cathode

M. J. Heneka et. al., Proc. 9th Int. Symp. on SOFC, pp. 534-543 (2005).

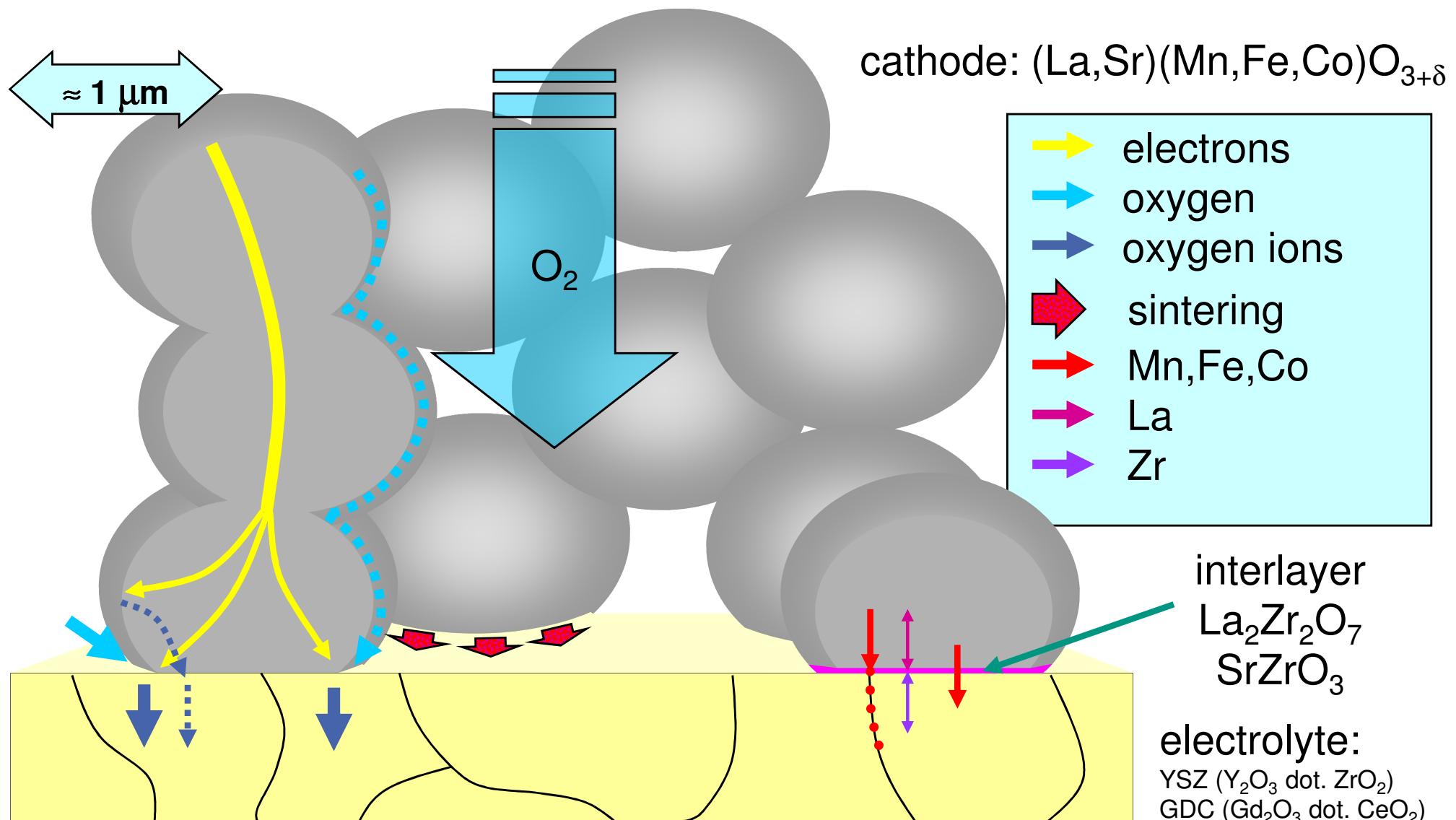
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A. Weber, in J. Garche (Ed.), Encyclopedia of Electrochemical Power Sources, Amsterdam: Elsevier, pp. 120-134 (2009).

⇒ in addition, activation processes that improve the performance take place

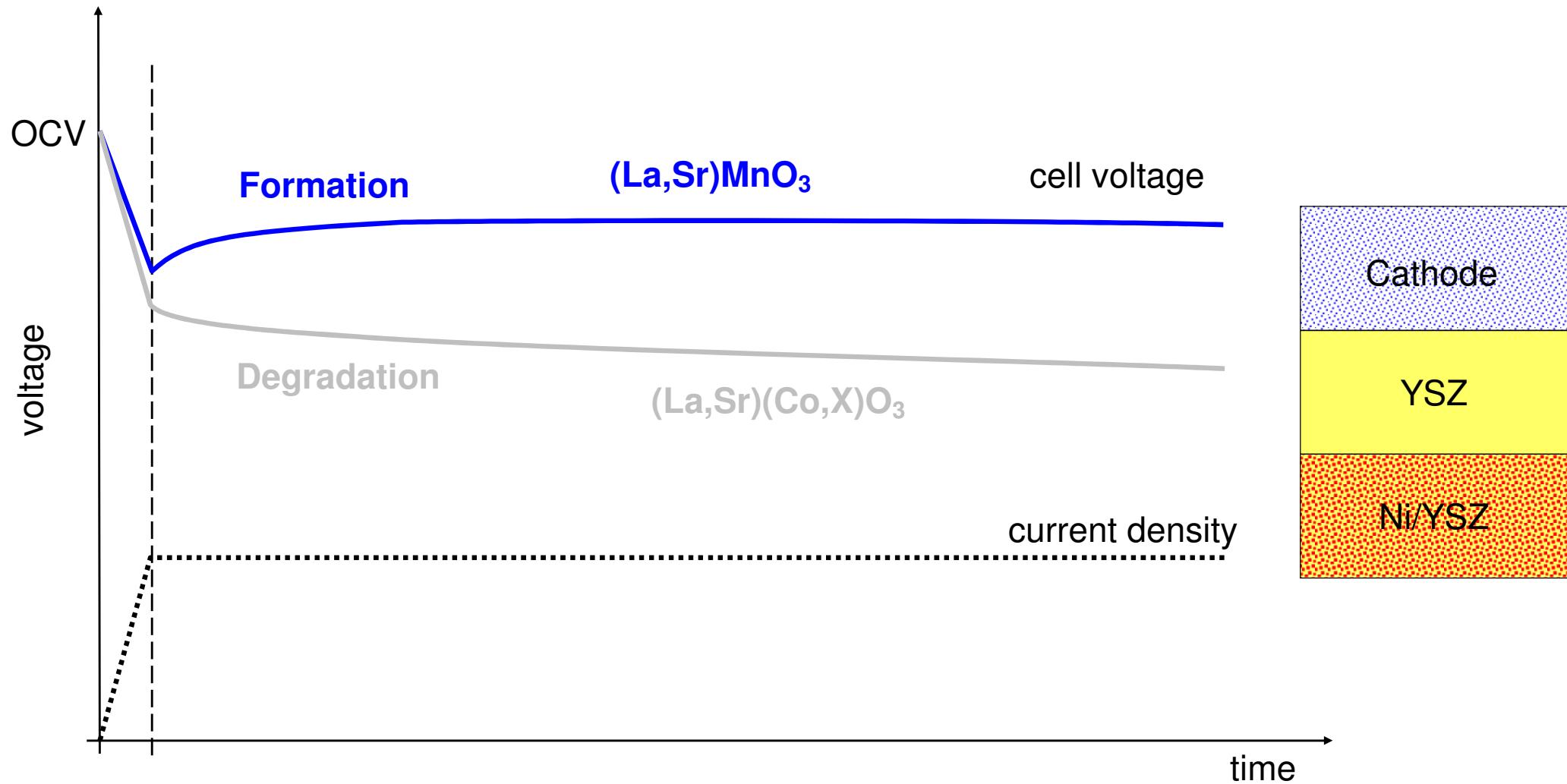
A. Weber et. al., *Denki Kagaku* **64**, pp. 582-589 (1996).

Stability of Interfaces

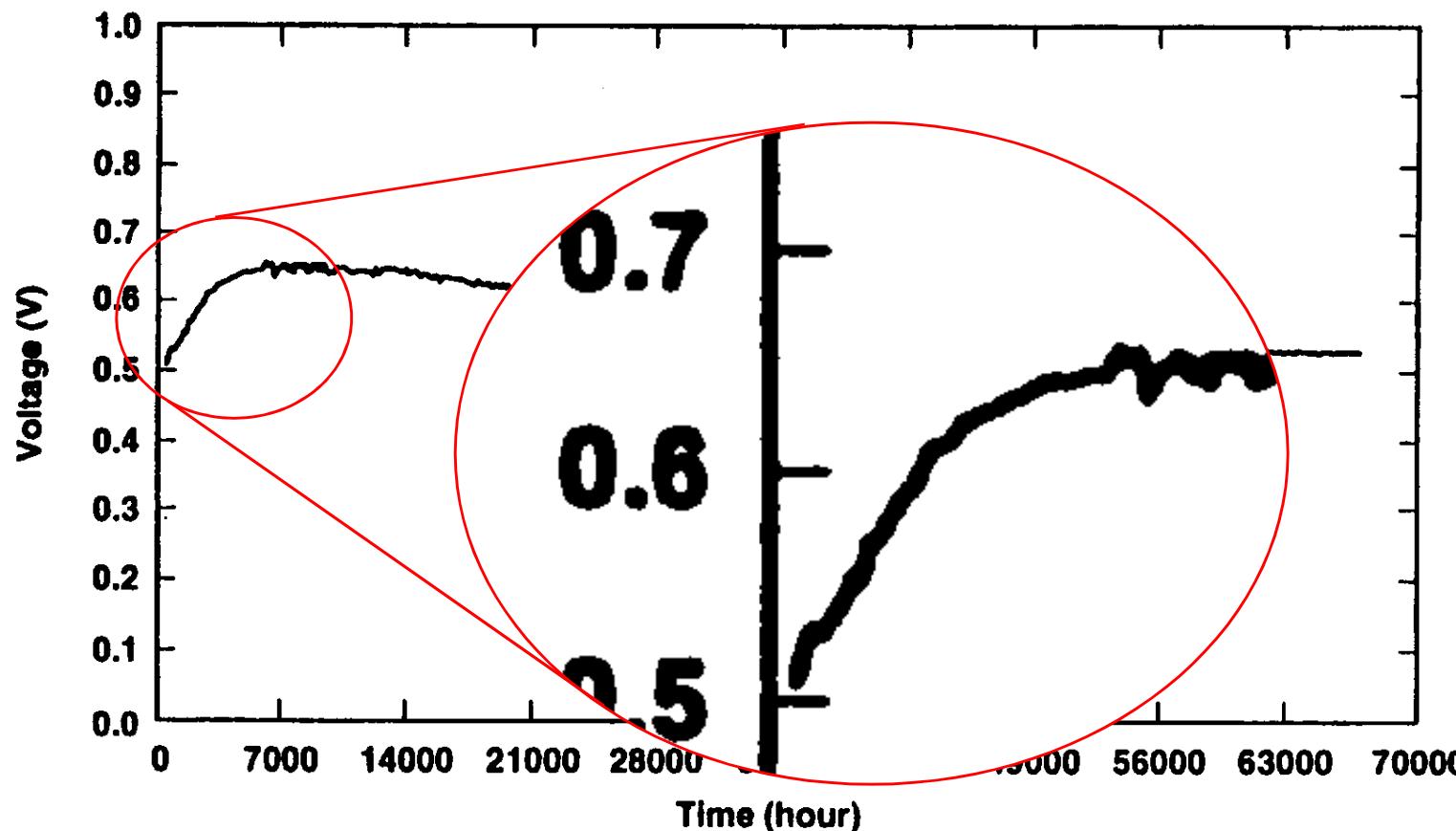


Startup Behavior of SOFC Single Cells and Stacks

Formation and Degradation



SOFC: Formation and Long Term Test of a tubular Cell Siemens-Westinghouse SWPC (1997)



$V_{\text{start}} : 0.50 \text{ V}$
 $V_{\text{max}} : 0.65 \text{ V}$

- ▶ V_{max} : Point of max. performance
- ▶ time to V_{max} :
 $\approx 5.000 \text{ hours}$

Fig. 1. Long term test of an early-technology PST cell.

S. C. Singhal, Recent Progress in Tubular Solid Oxide Fuel Cell Technology, Proc. 5th Int. Symposium on SOFC,
Ed. U. Stimming, S. C. Singhal, H. Tagawa, W. Lehnert, The Electrochemical Society, PV 97-40, 37-50, (1997)



SOFC: Formation and Long Term Test of a 5-cell Stack Sulzer Hexis (1999)

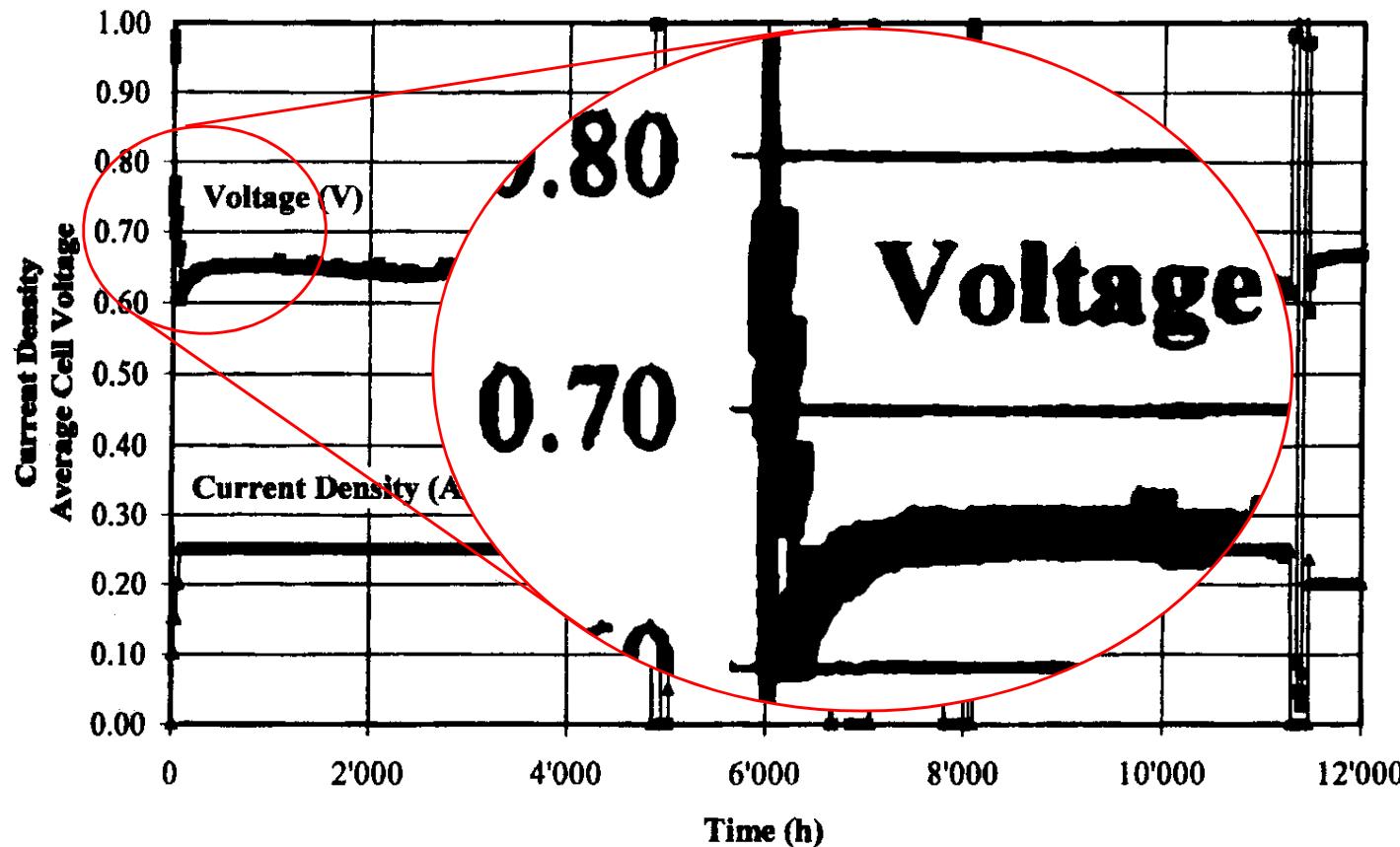


Figure 2: Negligible degradation of a five cell stack

$V_{\text{start}} : 0.60 \text{ V}$

$V_{\text{max}} : 0.65 \text{ V}$

► V_{max} : Point of max. performance

► time to V_{max} :

$\approx 200 \text{ hours}$

R. Diethelm, M. Schmidt, Status of the Sulzer Hexis Product Development, Proc. 6th Int. Symp. on SOFC,
Ed. S. Singhal und M. Dokiya, The Electrochemical Society, PV 99-19, 60-67, (1999)

SOFC: Formation and Long Term Test of a planar Cell Ceramic Fuel Cells Ld. CFCL (1999)

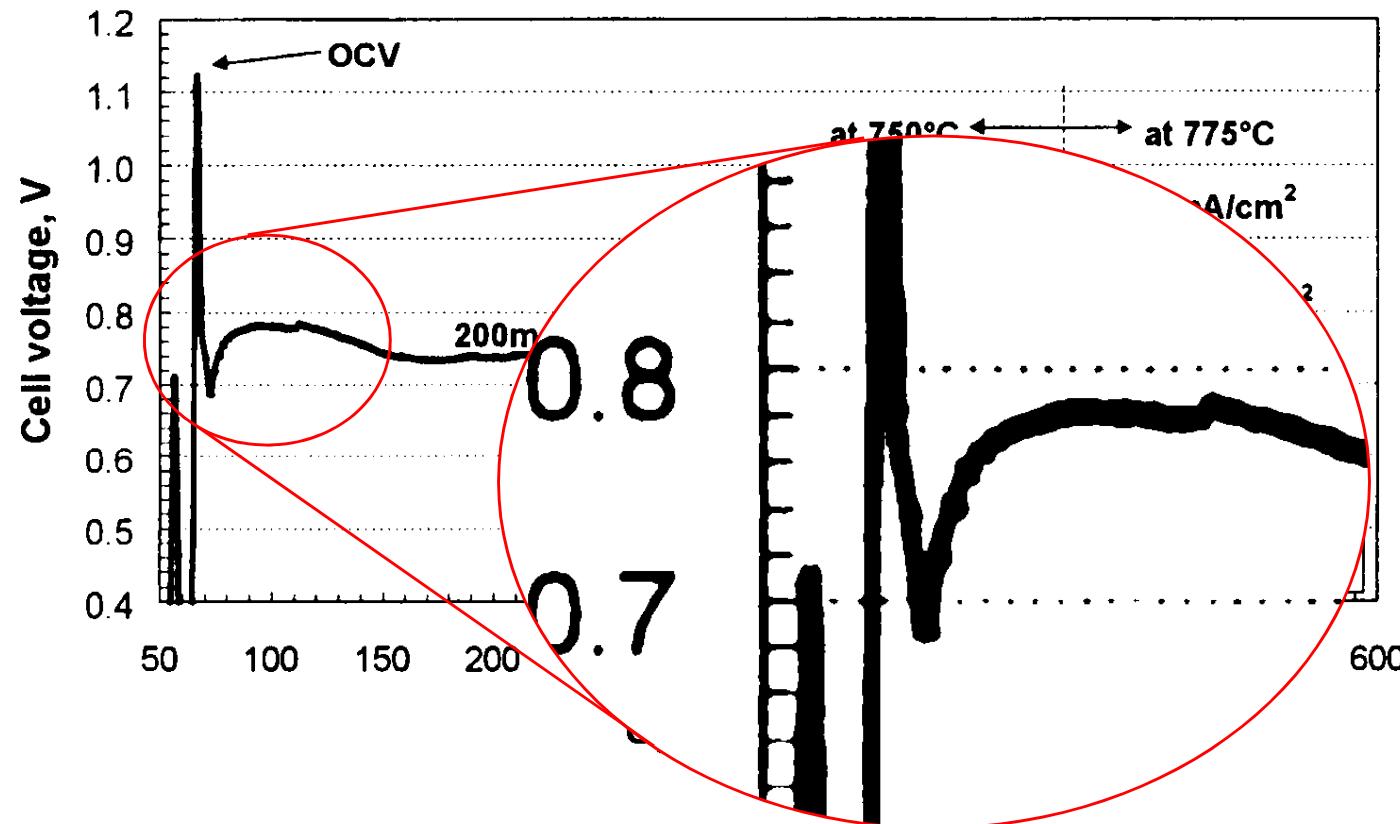


Figure 2. Cell performance at 750 and 775°C tested in alumina assembly

K. Föger et al., Demonstration of Anode Supported Cell Technology in kW Class Stack, Proc. 6th Int. Symp. on SOFC,
Ed. S. Singhal und M. Dokiya, The Electrochemical Society, 95-100, PV 99-19, (1999)



SOFC: Formation and Long Term Test of a 1 kW Stack Mitsubishi Heavy Industries MHI (1999)

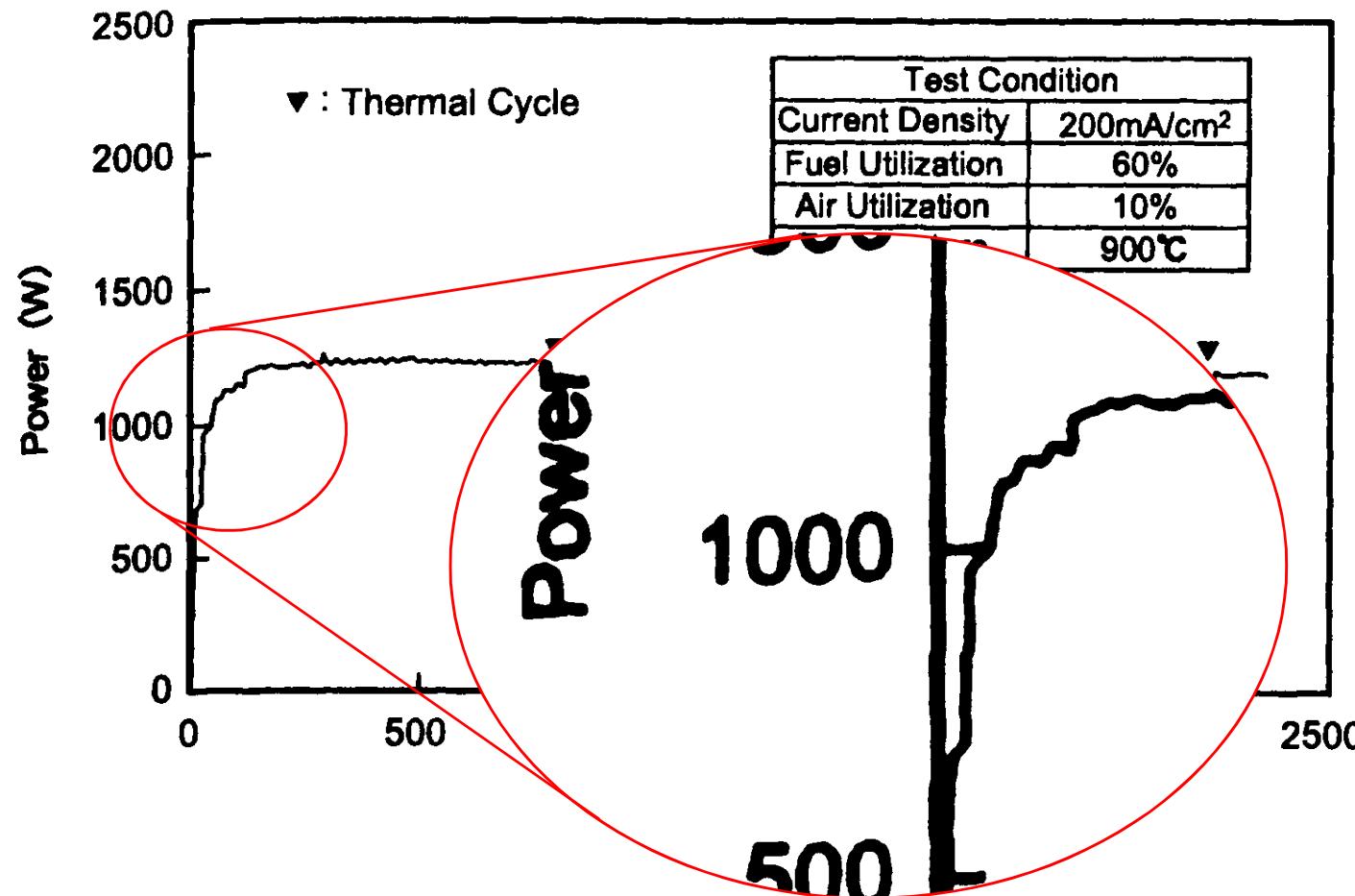


Fig. 12 Endurance Test of 1kW Module with Co-Sintered Cell Stack

H. Mori et al., Pressurized 10 kW Class Module of SOFC, Proc. 6th Int. Symp. on SOFC,
Ed. S. Singhal und M. Dokiya, - The Electrochemical Society, PV 99-19, 52-59, (1999)

P_{start} : 700 W

P_{max} : 1250 W

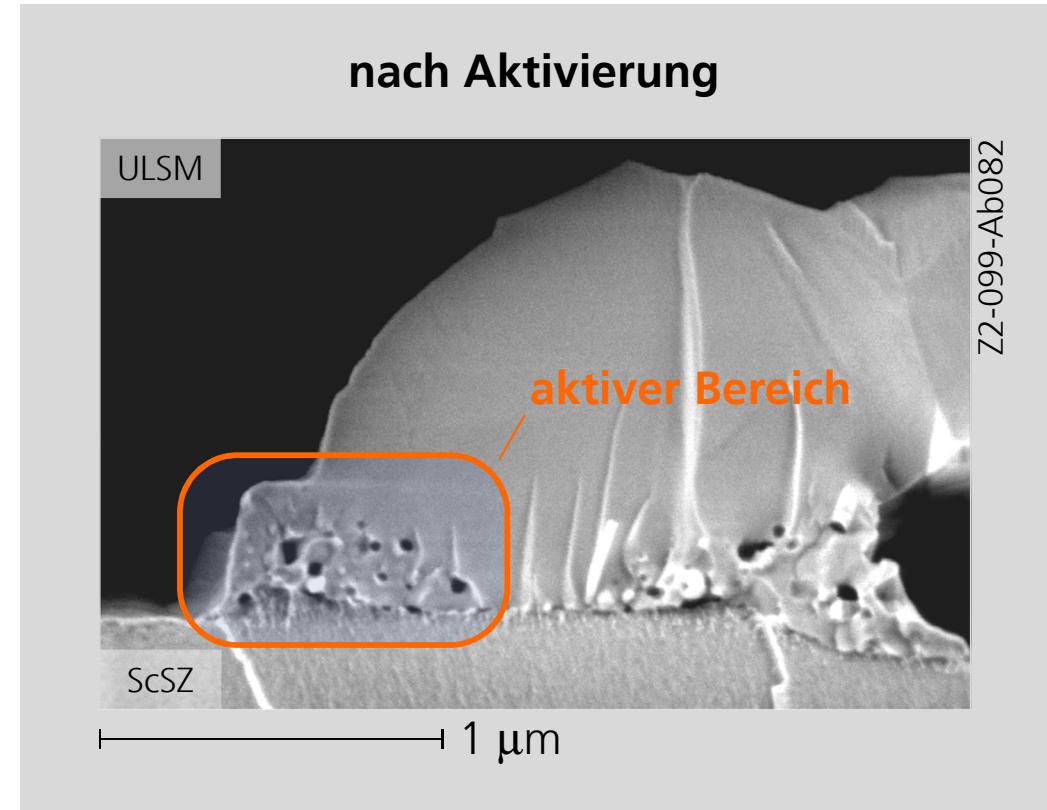
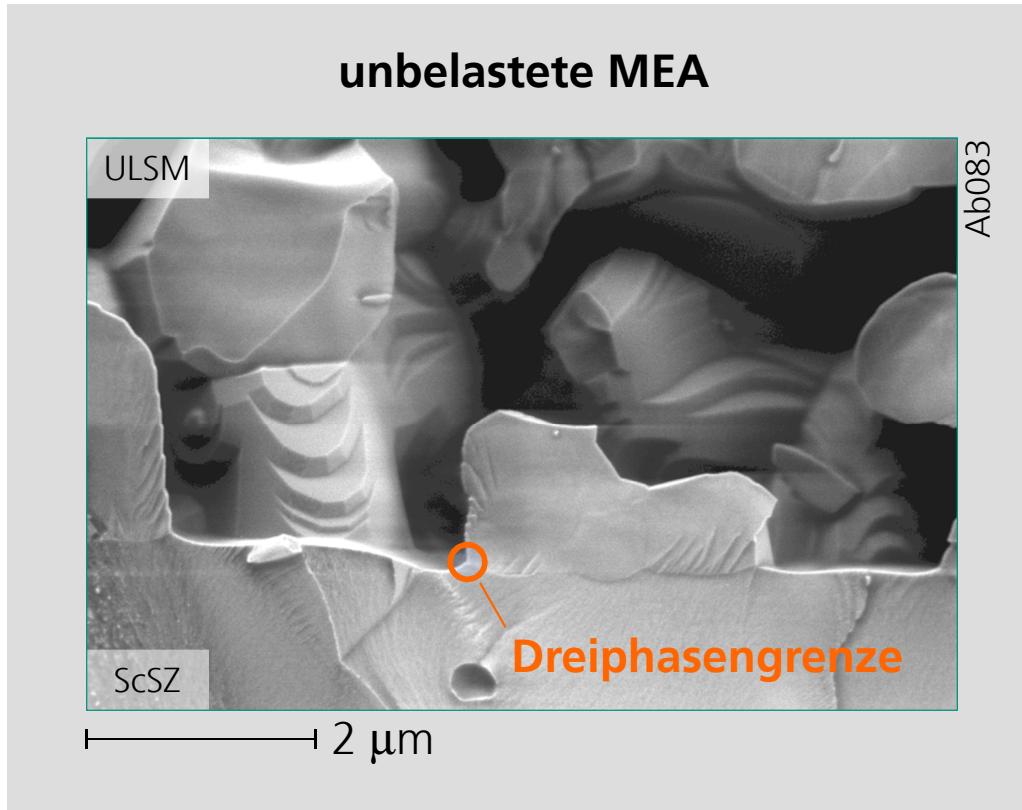
► P_{max} : Point of max. performance

► time to P_{max} :

≈ 300 hours

Grenzfläche Kathode-Elektrolyt vor und nach Aktivierung

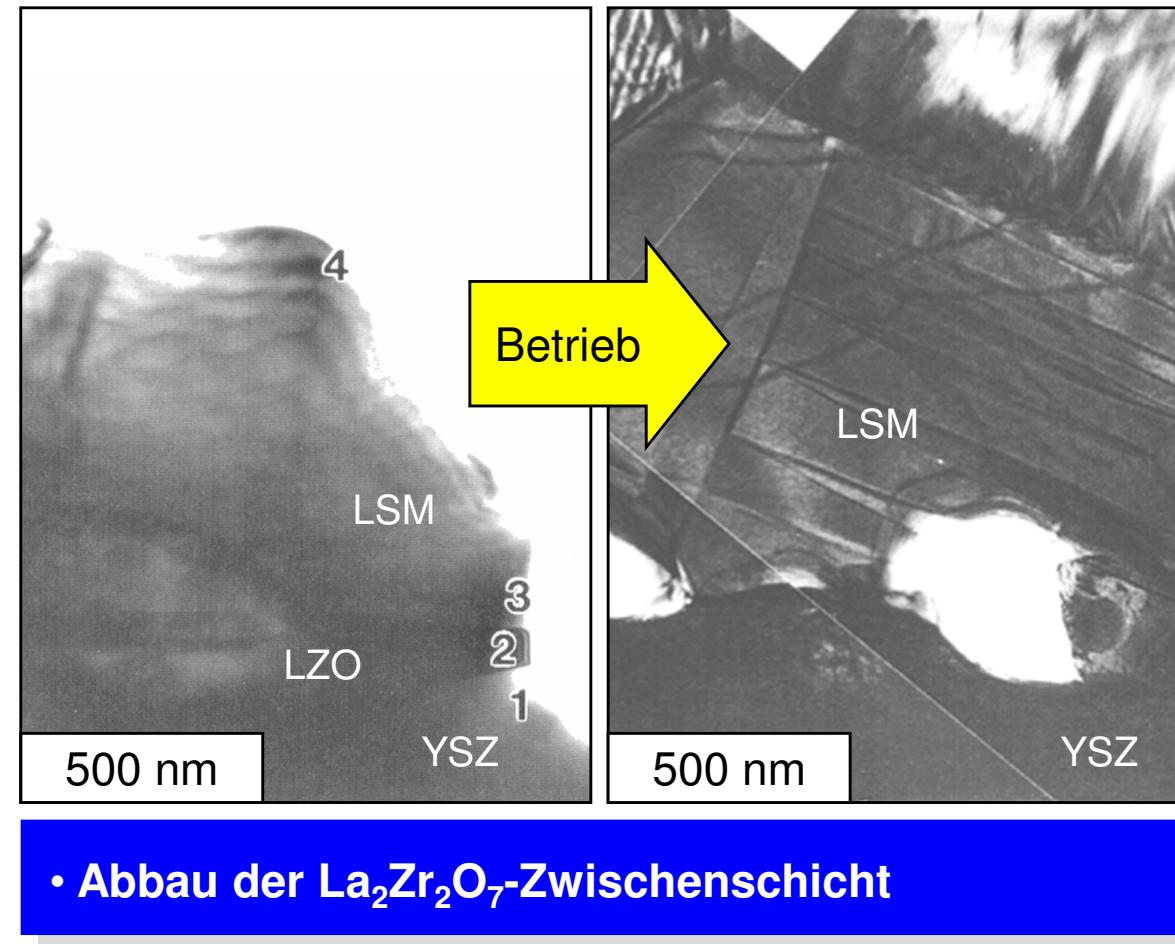
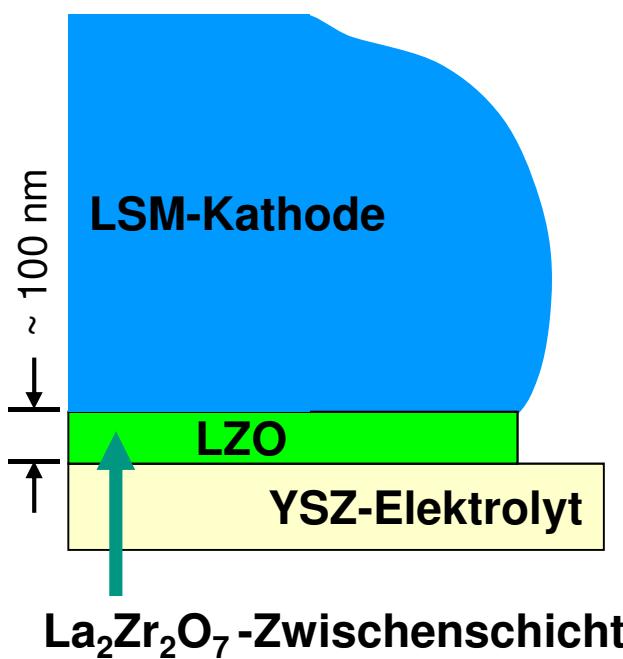
Mikrostruktur



- Bildung eines porösen Bereichs im ULSM-Korn nahe der Grenzfläche unter Belastung
- Reaktionsraum vergrößert ⇒ höhere elektrochemische Leistungsfähigkeit der Kathode

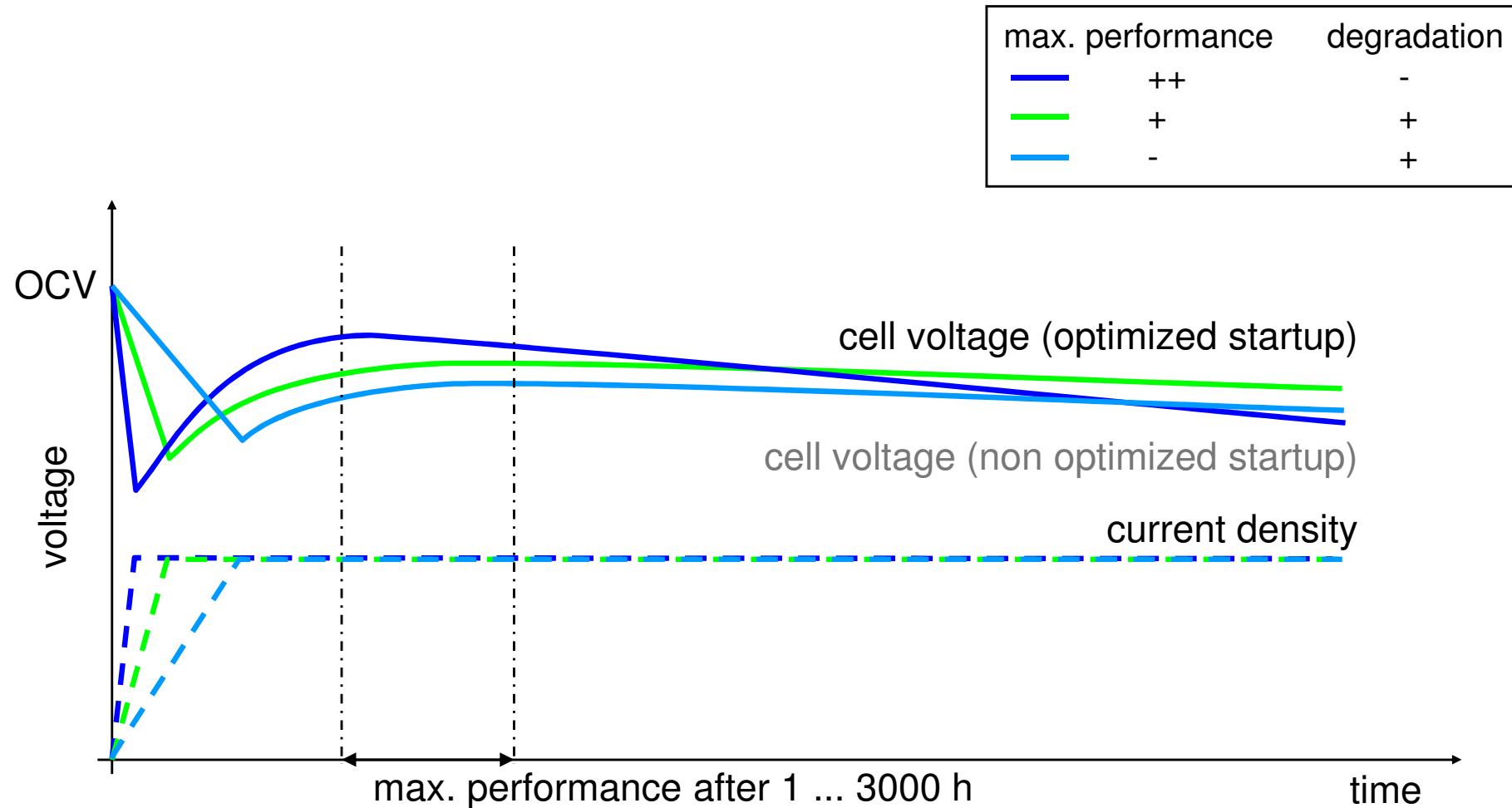
Grenzfläche Kathode-Elektrolyt vor und nach Aktivierung

Chemische Zusammensetzung



Startup Behavior of SOFC Single Cells and Stacks

Formation and Degradation



Modeling of Degradation Phenomena

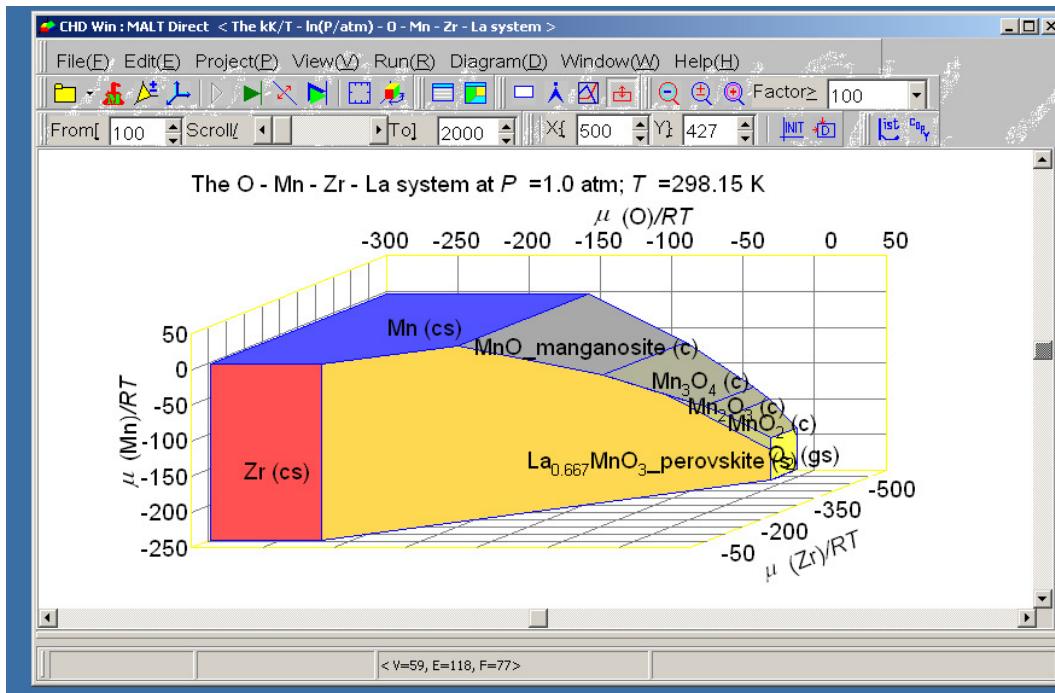
Chemical Potential Diagrams and Thermodynamic Equilibrium Calculations

MALT (MAterials oriented Little Thermodynamic database*)

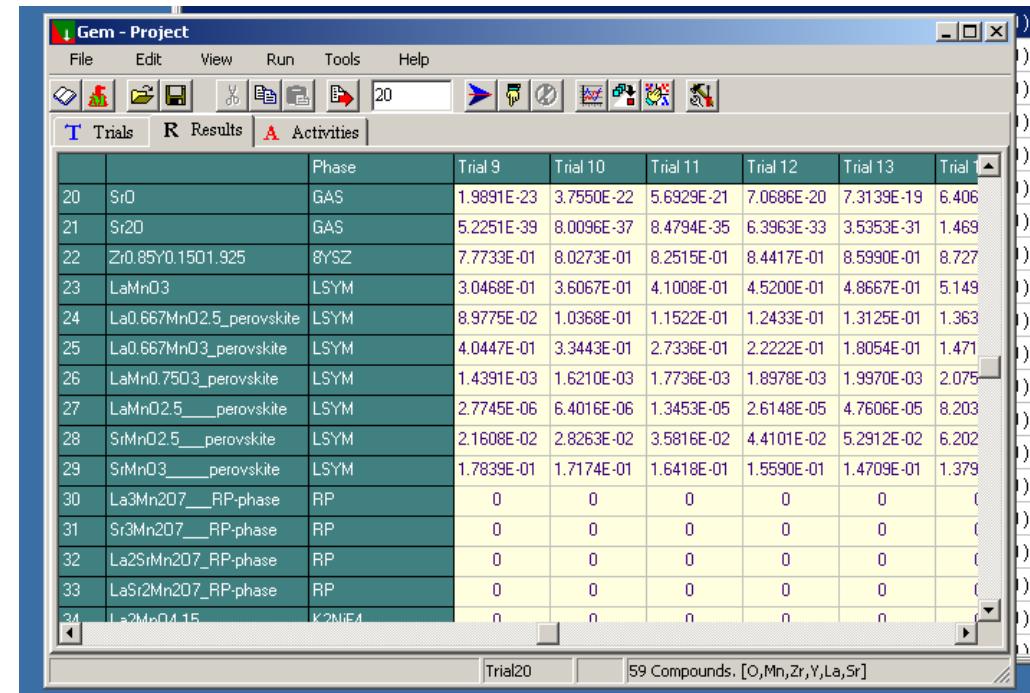
$\Delta_f G^\circ$, $\Delta_f H^\circ$, S° , C_p° at 298.15 K

temperature coefficients of heat capacities, transition temperatures, transition enthalpies

Calculation of Chemical Potential Diagrams (CHD)



Thermodynamic Equilibrium Calculations (GEM: Gibbs Energy Minimizer)



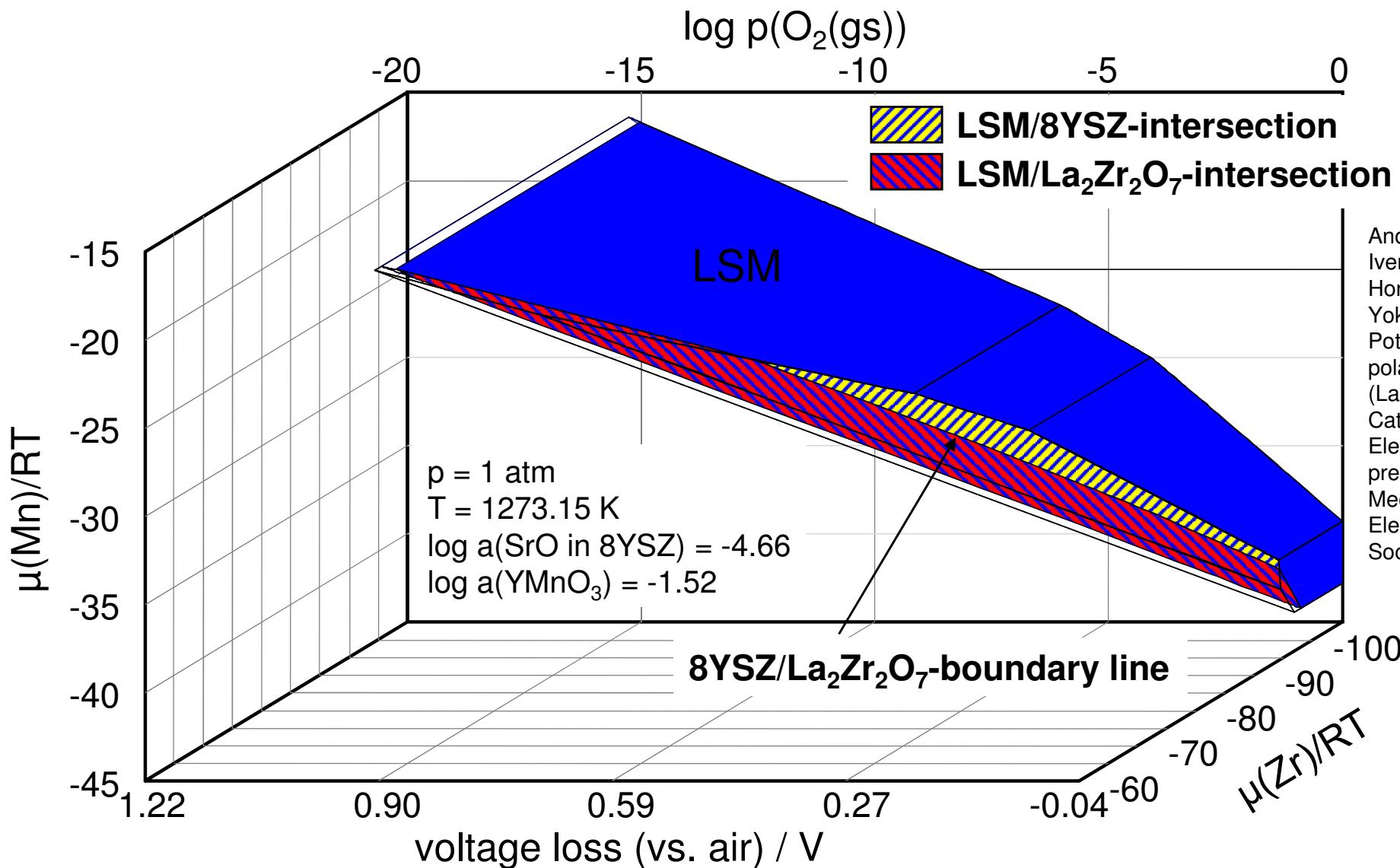
	Phase	Trial 9	Trial 10	Trial 11	Trial 12	Trial 13	Trial 14
20	SrO	1.9891E-23	3.7550E-22	5.6929E-21	7.0686E-20	7.3139E-19	6.406
21	Sr2O	5.2251E-39	8.0096E-37	8.4794E-35	6.3963E-33	3.5353E-31	1.469
22	Zr0.85Y0.15O1.925	7.7733E-01	8.0273E-01	8.2515E-01	8.4417E-01	8.5990E-01	8.727
23	LaMnO3	3.0468E-01	3.6067E-01	4.1008E-01	4.5200E-01	4.8667E-01	5.149
24	La0.667MnO2.5_perovskite	8.9775E-02	1.0368E-01	1.1522E-01	1.2433E-01	1.3125E-01	1.363
25	La0.667MnO3_perovskite	4.0447E-01	3.3443E-01	2.7336E-01	2.2222E-01	1.8054E-01	1.471
26	LaMn0.75O3_perovskite	1.4391E-03	1.6210E-03	1.7736E-03	1.8978E-03	1.9970E-03	2.075
27	LaMn0.25_perovskite	2.7745E-06	6.4016E-06	1.3453E-05	2.6148E-05	4.7606E-05	8.203
28	SrMnO2.5_perovskite	2.1608E-02	2.8263E-02	3.5816E-02	4.4101E-02	5.2912E-02	6.202
29	SrMnO3_perovskite	1.7839E-01	1.7174E-01	1.6418E-01	1.5590E-01	1.4709E-01	1.379
30	La3Mn2O7_RP-phase	0	0	0	0	0	0
31	Sr3Mn2O7_RP-phase	0	0	0	0	0	0
32	La2SrMn2O7_RP-phase	0	0	0	0	0	0
33	LaSr2Mn2O7_RP-phase	0	0	0	0	0	0
34	La2MnO4.15	n	n	n	n	n	n

* H. Yokokawa et al., *Thermochimica Acta* 245, 45-55 (1994)
MALT for Windows, see <http://www.kagaku.com/malt/index.html>



Modeling of Degradation Phenomena

Chemical Potential Diagram for the polarized LSM / 8YSZ - Interface

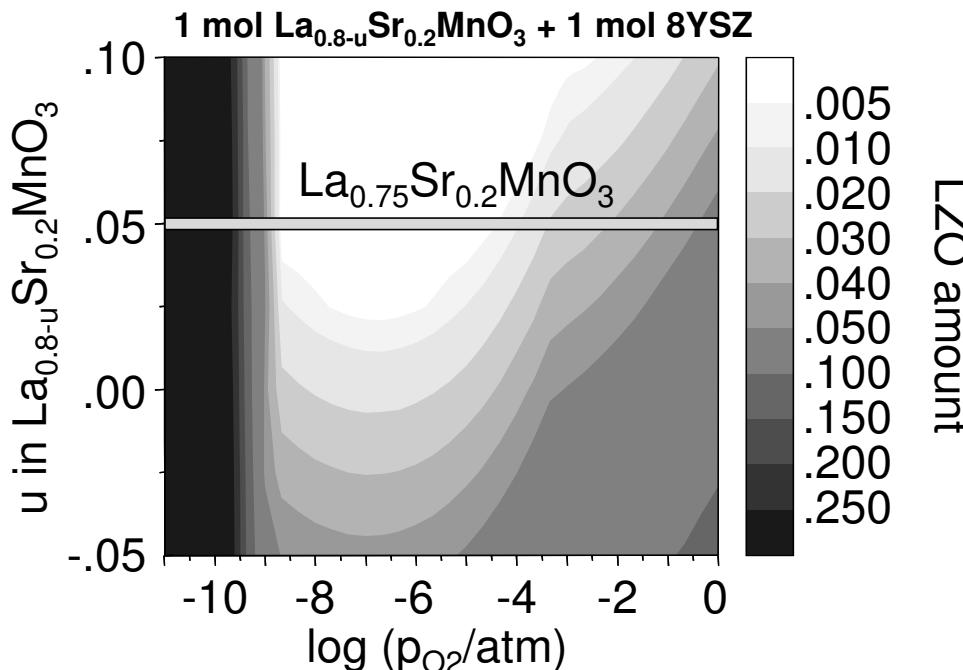


André Weber, Ellen Ivers-Tiffée, Teruhisa Horita, Harumi Yokokawa, Chemical Potential Diagrams for polarized (La,Sr)MnO_{3+d}-Cathode/8YSZ-Electrolyte-Interfaces", presented at the 206th Meeting of The Electrochemical Society, 2004

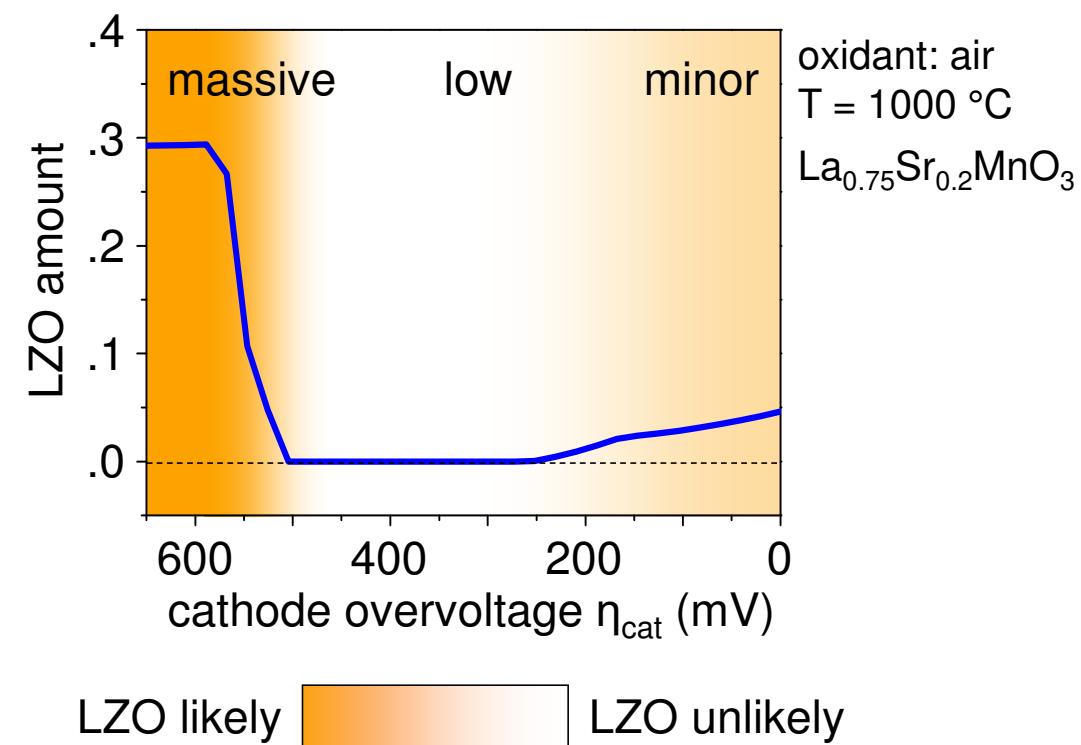
Modeling of Degradation Phenomena

Formation of insulating LZO-layers at the LSM/8YSZ Interface

calculated amount of LZO



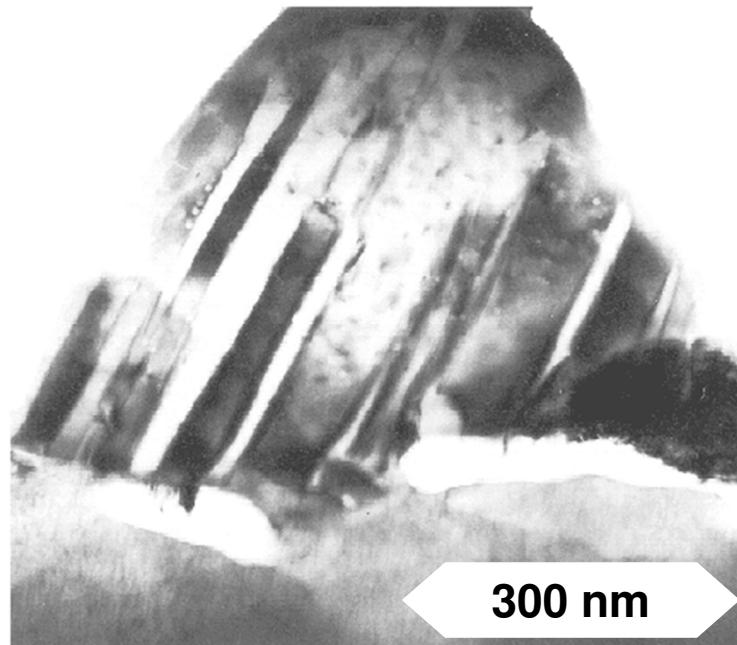
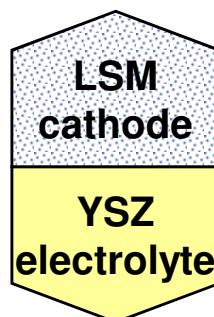
impact of cathode overvoltage η_{cat}



M. J. Heneka, E. Ivers-Tiffée: Degradation of SOFC Single Cells Under Severe Current Cycles. In: S. C. Singhal, J. Mizusaki (Hrsg.), Proc. 9th Int. Symp. on SOFC, PV 2005-07, The Electrochemical Society, 534-543 (2005)

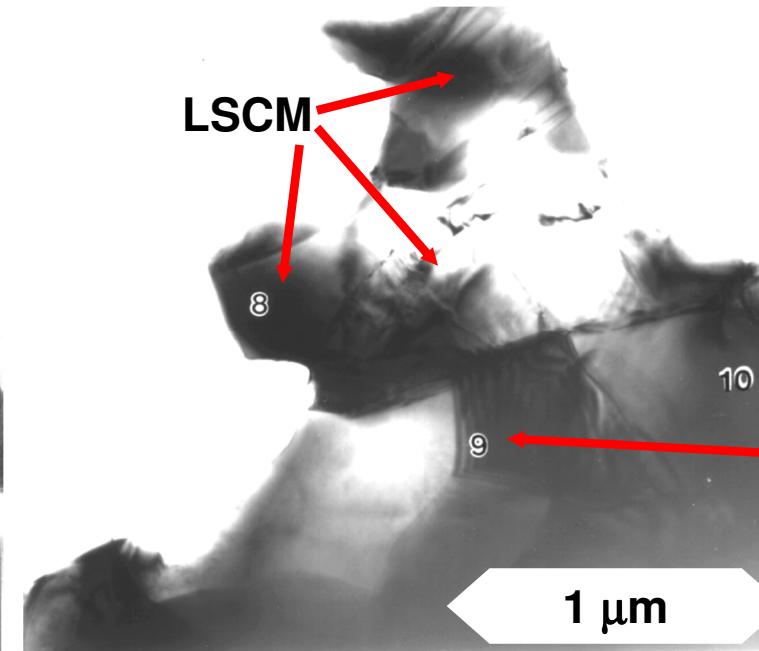
Cathode Degradation Growth of Secondary Phases at the Cathode/Electrolyte-Interface

Interface LSM/YSZ
(LSM: $\text{La}_{0.8}\text{Sr}_{0.2}\text{MnO}_{3+\delta}$)

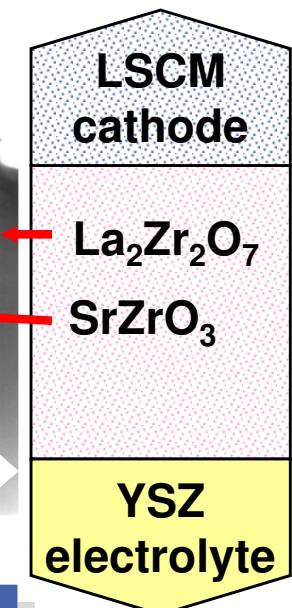


**Low Cathode Resistance
after Formation**

Interface LSCM/YSZ
(LSCM: $\text{La}_{0.8}\text{Sr}_{0.2}\text{Co}_{0.5}\text{Mn}_{0.5}\text{O}_{3+\delta}$)

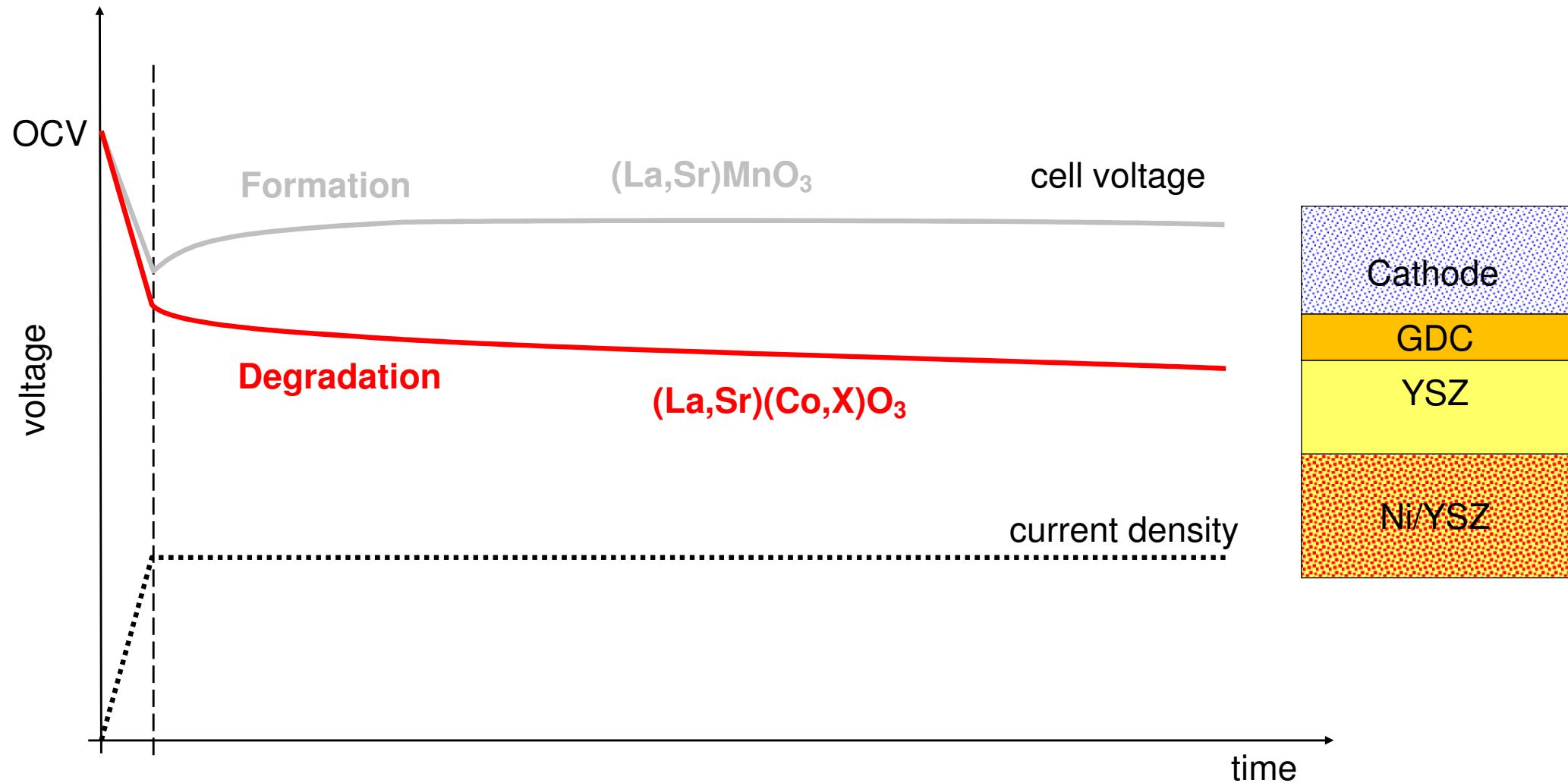


**High Cathode Resistance
Growth of Secondary Phases**

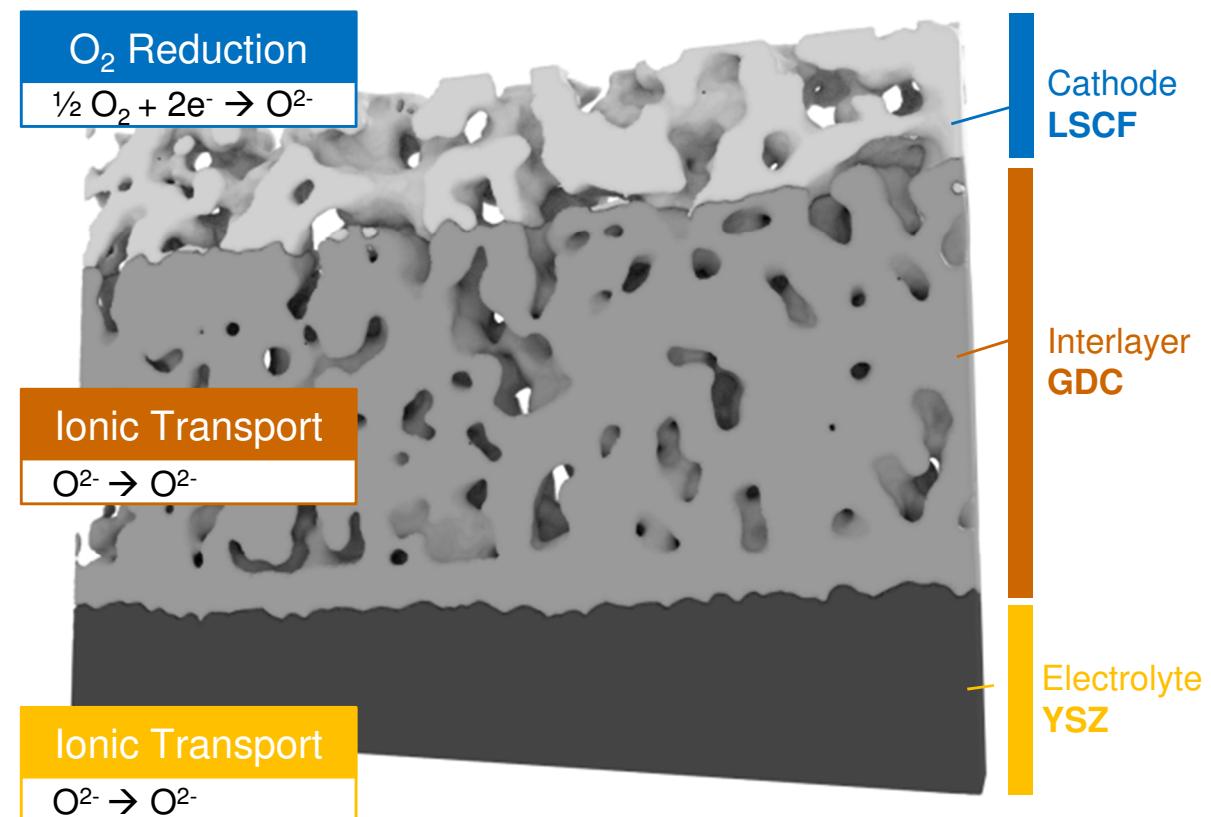
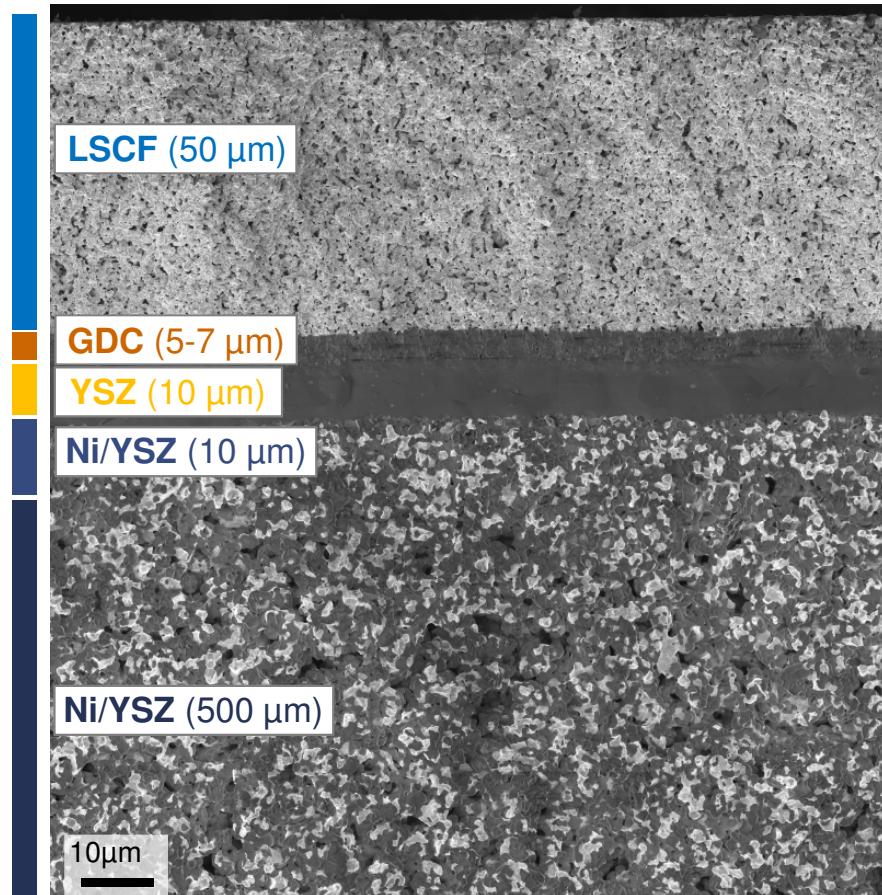


Startup Behavior of SOFC Single Cells and Stacks

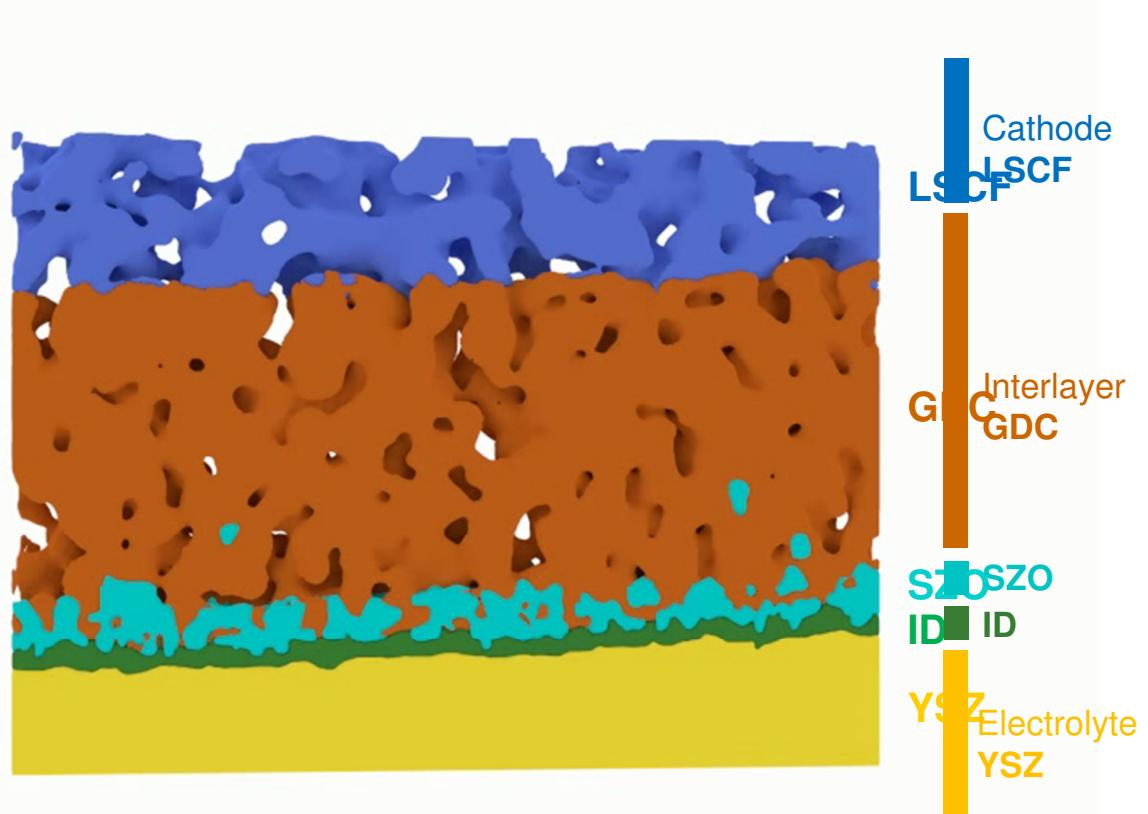
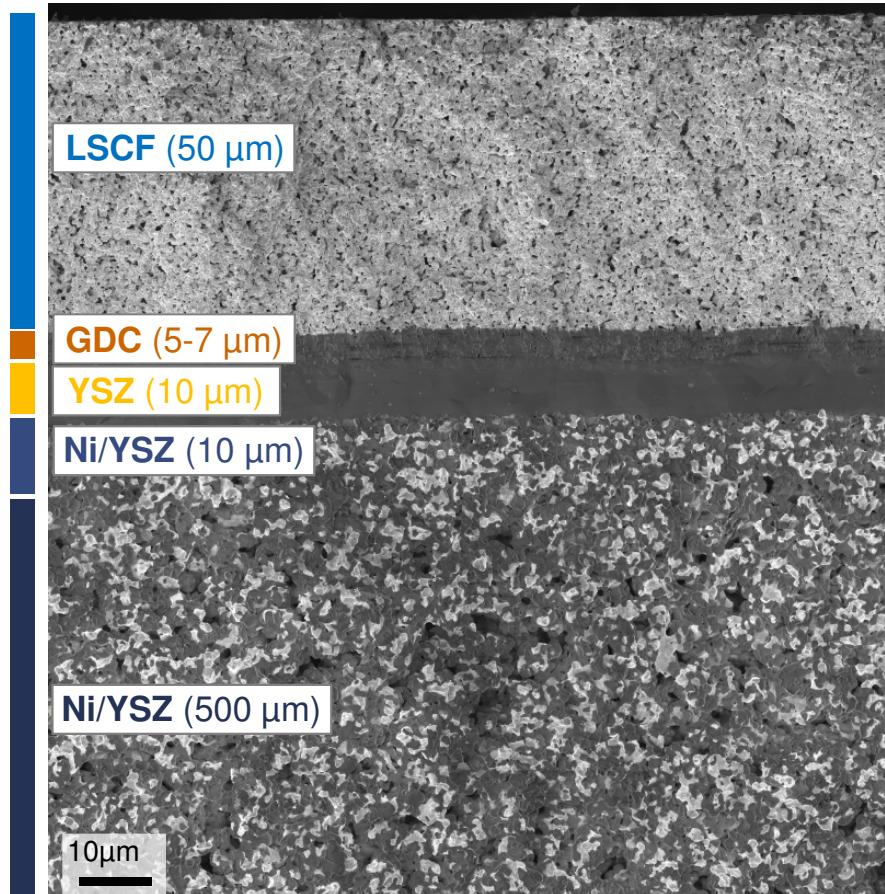
Formation and Degradation



Cathode/Electrolyte Interface

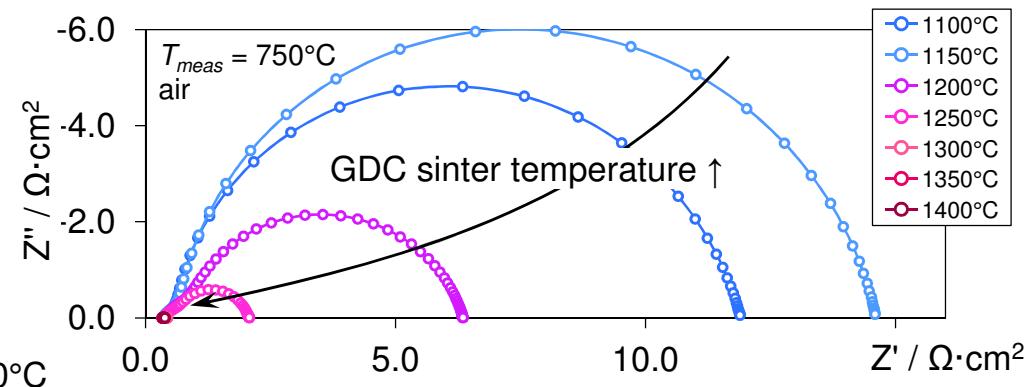
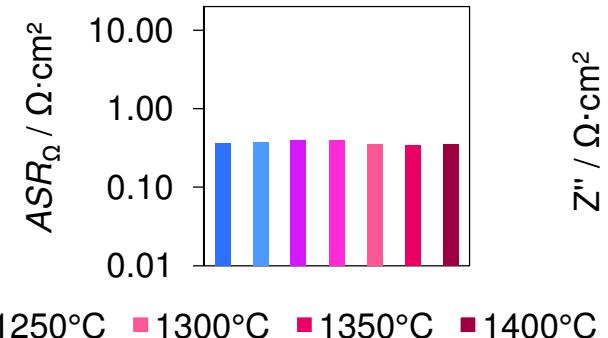
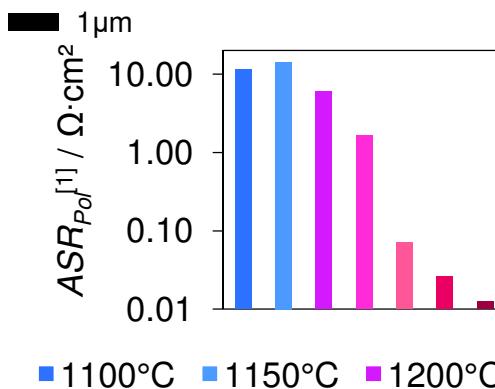
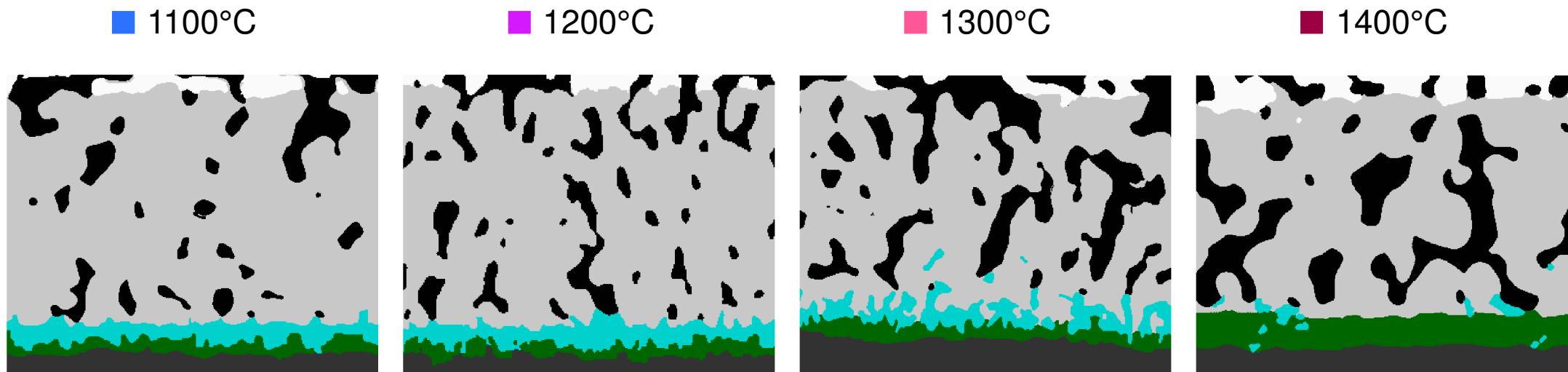


Cathode/Electrolyte Interface Distribution of Secondary Phases



Cathode/Electrolyte Interface Impact of Processing Conditions

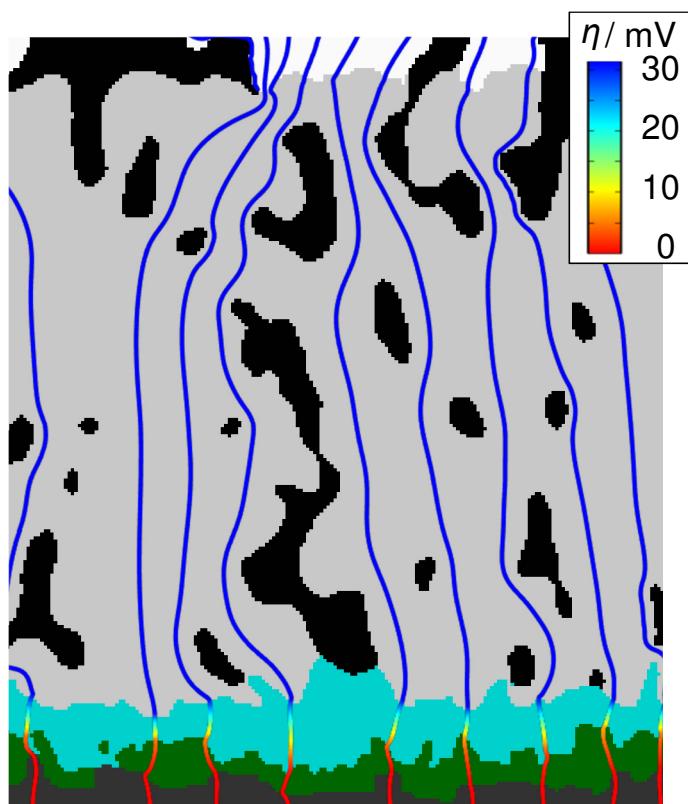
GDC sintering temperature



Cathode/Electrolyte Interface Cathode Overpotential and Current Pathways

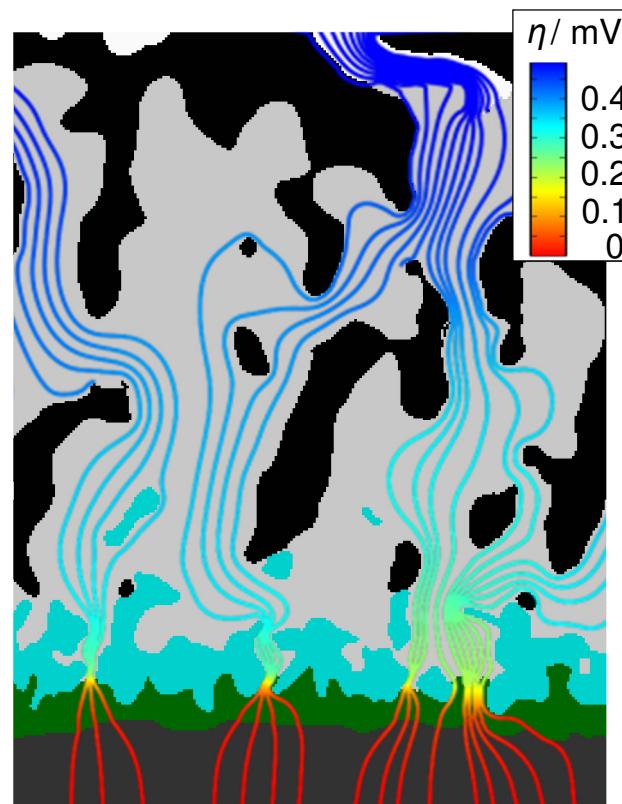
Continuous SrZrO₃

■ 1200°C



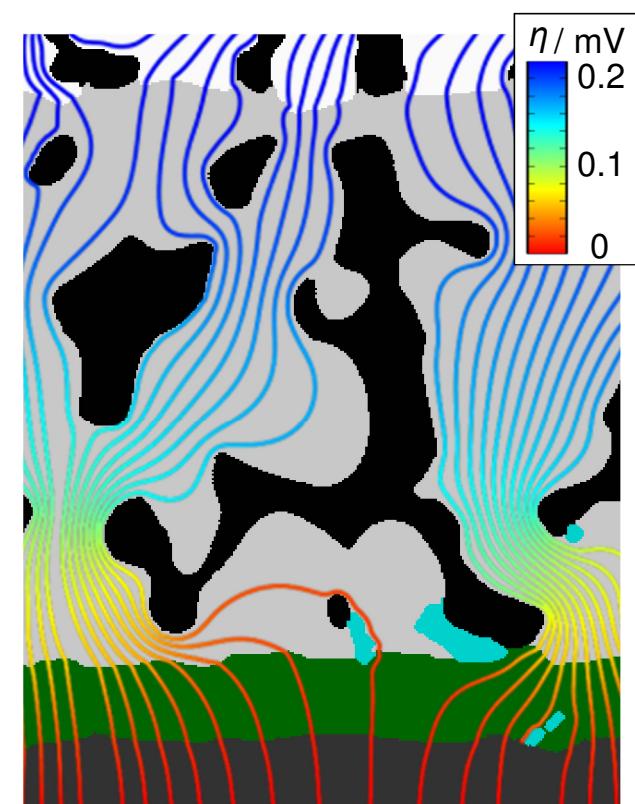
Intermittent SrZrO₃

■ 1300°C



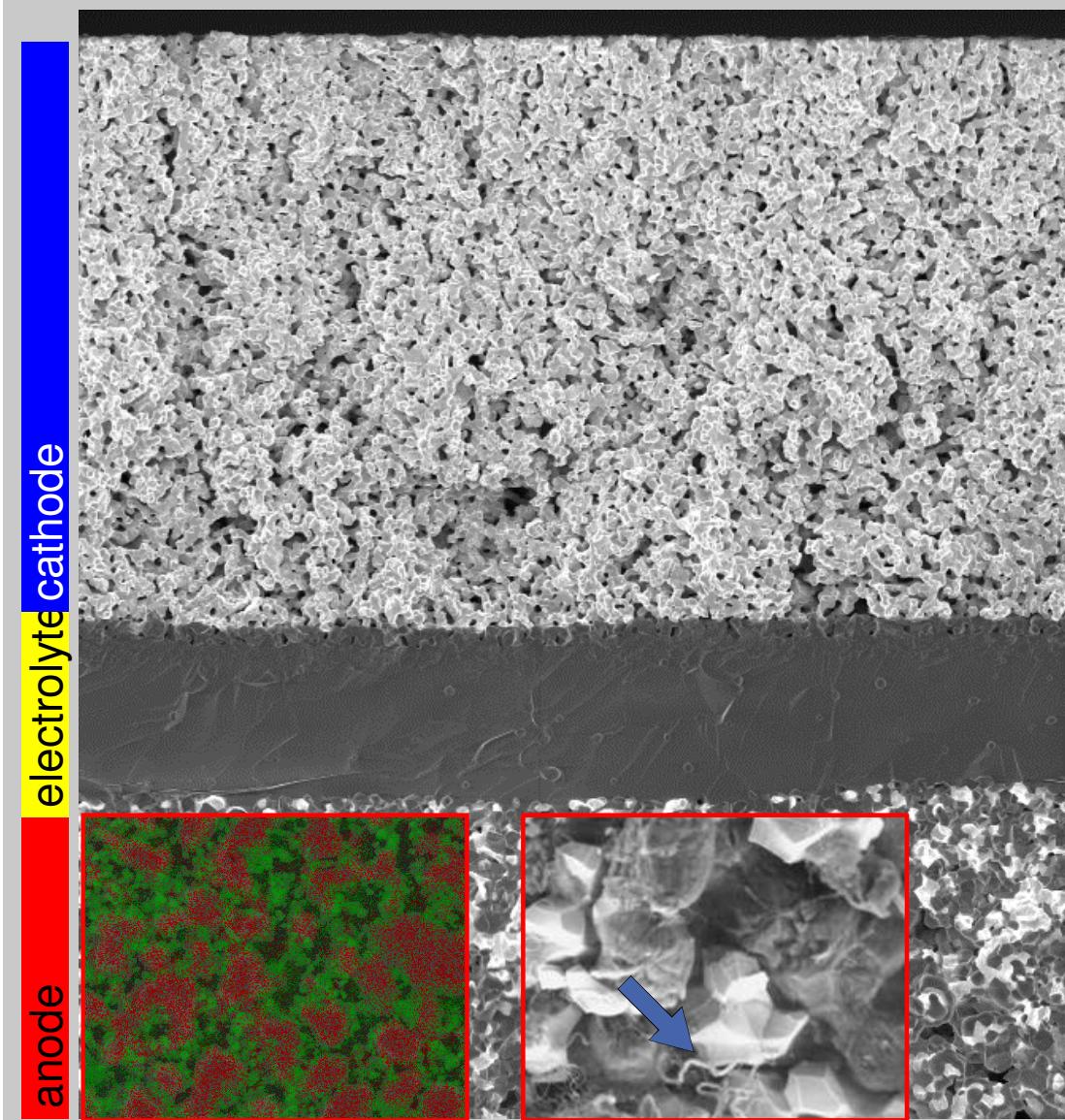
Continuous Interdiffusion

■ 1400°C



@ $f = 0.01$ Hz $j = 10\text{mA/cm}^2$ $T_{meas} = 800^\circ\text{C}$

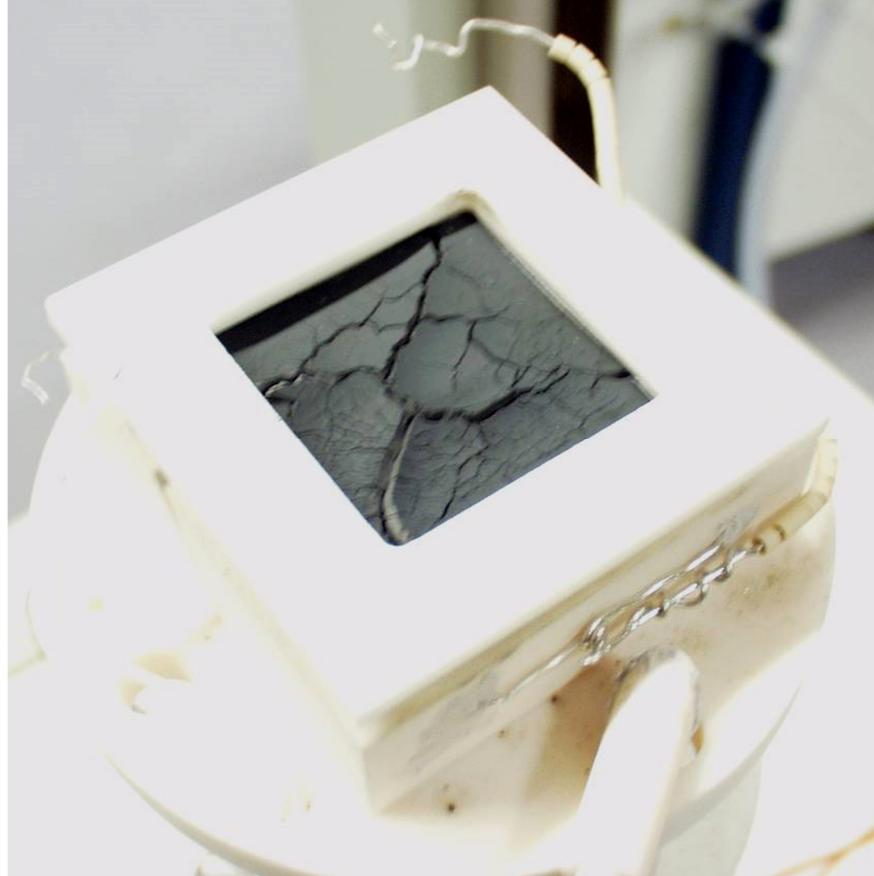
Degradation Processes in Solid Oxide Fuel Cells



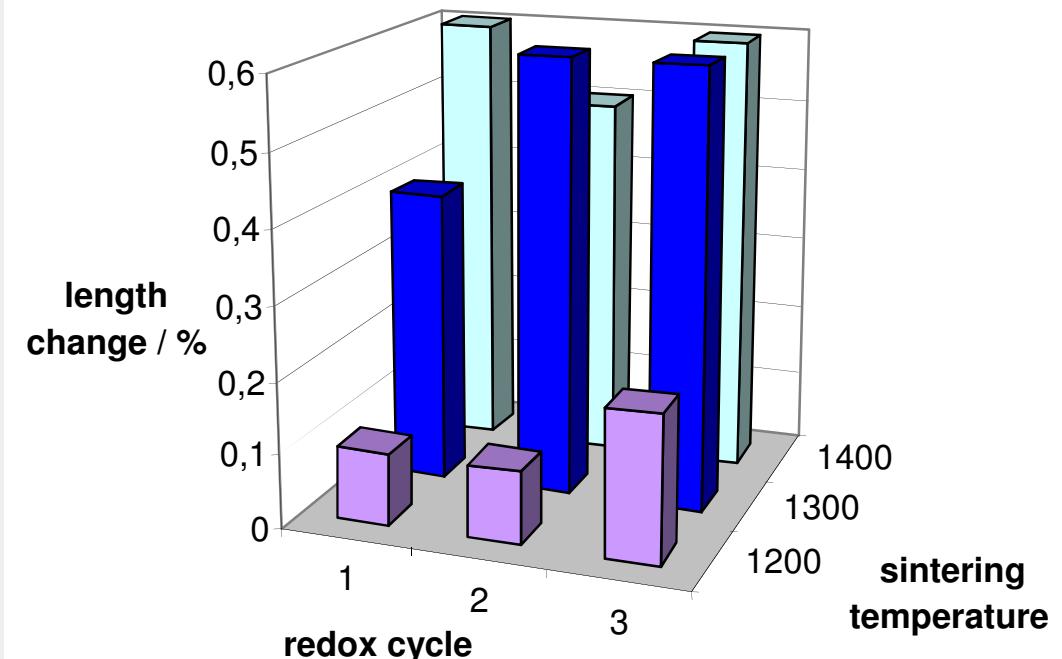
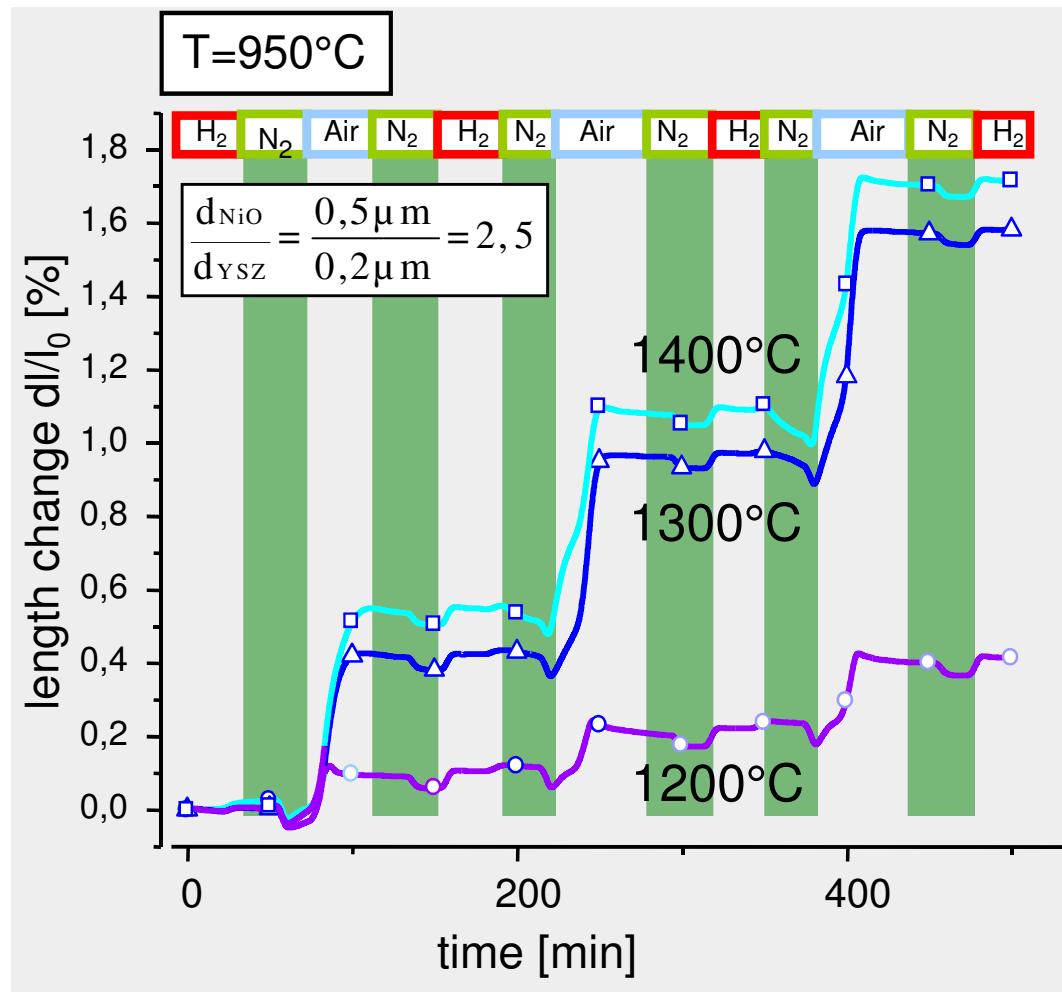
in SOFC many potentially competing degradation mechanisms are known:

- Ni-agglomeration
A. C. Müller et. al., *Proc. 3rd European SOFC Forum*, pp. 353-362 (1998).
- carbon deposition
E. Ivers-Tiffée et. al., *Handbook of Fuel Cells – Fundamentals, Technology and Applications*, pp. 933-956 (2009).

Extrinsic Degradation Cracking of an Anode Supported Electrolyte after Redox-Cycling

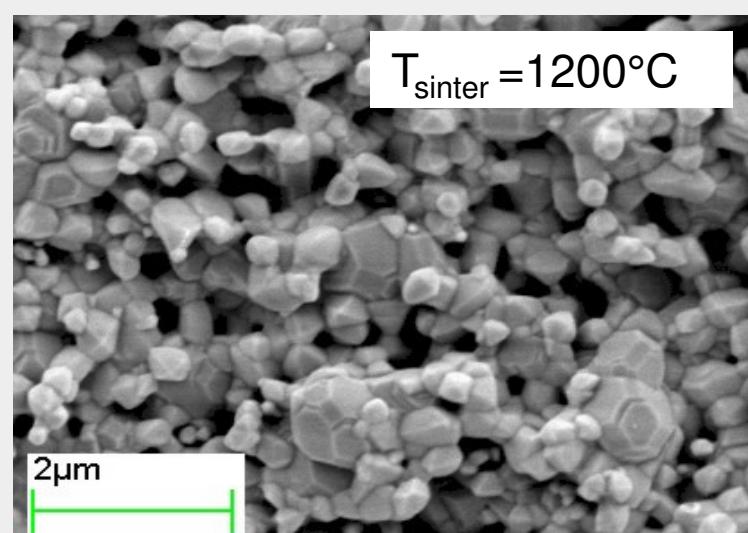


Impact of Redox Cycle on Length Change of Ni/YSZ Bulks: Different Sintering Temperatures



- minimum length change at lowest sintering temperature of 1200 °C

Impact of Redox Cycles on Microstructure of Ni/YSZ Bulks Different Sintering Temperatures

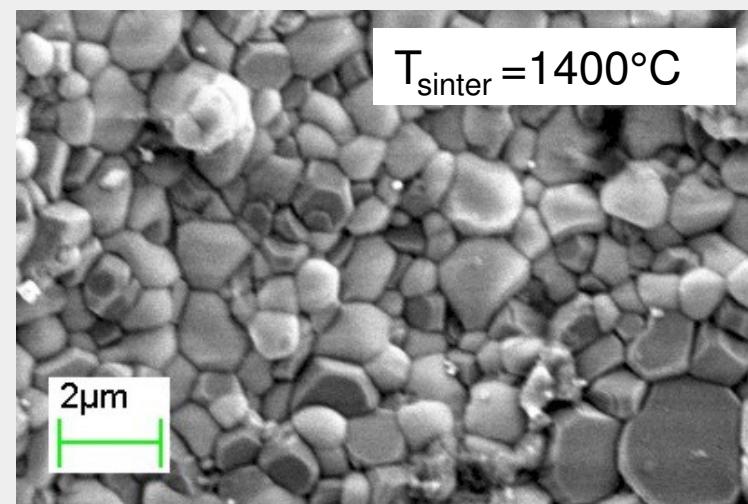


$T_{\text{sinter}} = 1200^\circ\text{C}$

$d_{50}: \text{NiO} = 0,5\mu\text{m}$
 $d_{50}: \text{YSZ} = 0,2\mu\text{m}$

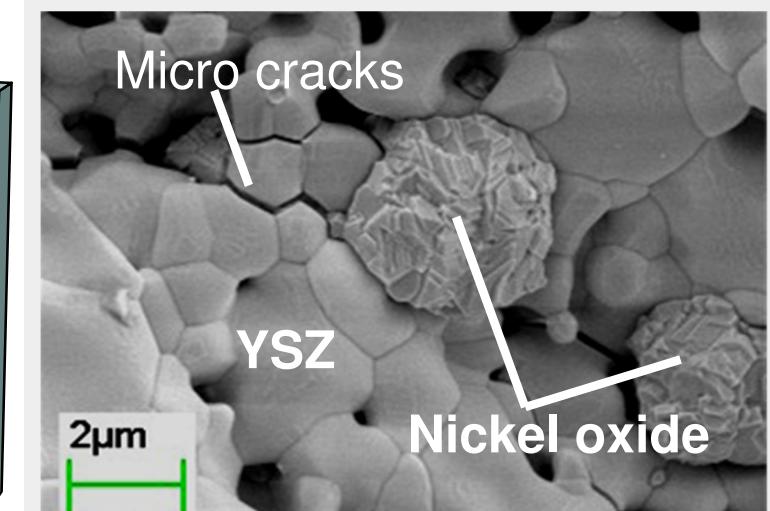
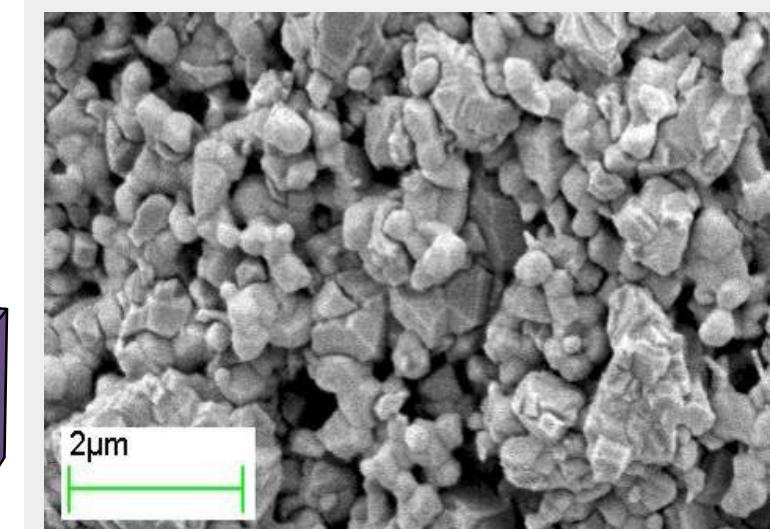
length change
 $\Delta l/l_0 = 0,4\%$

3 redox cycles

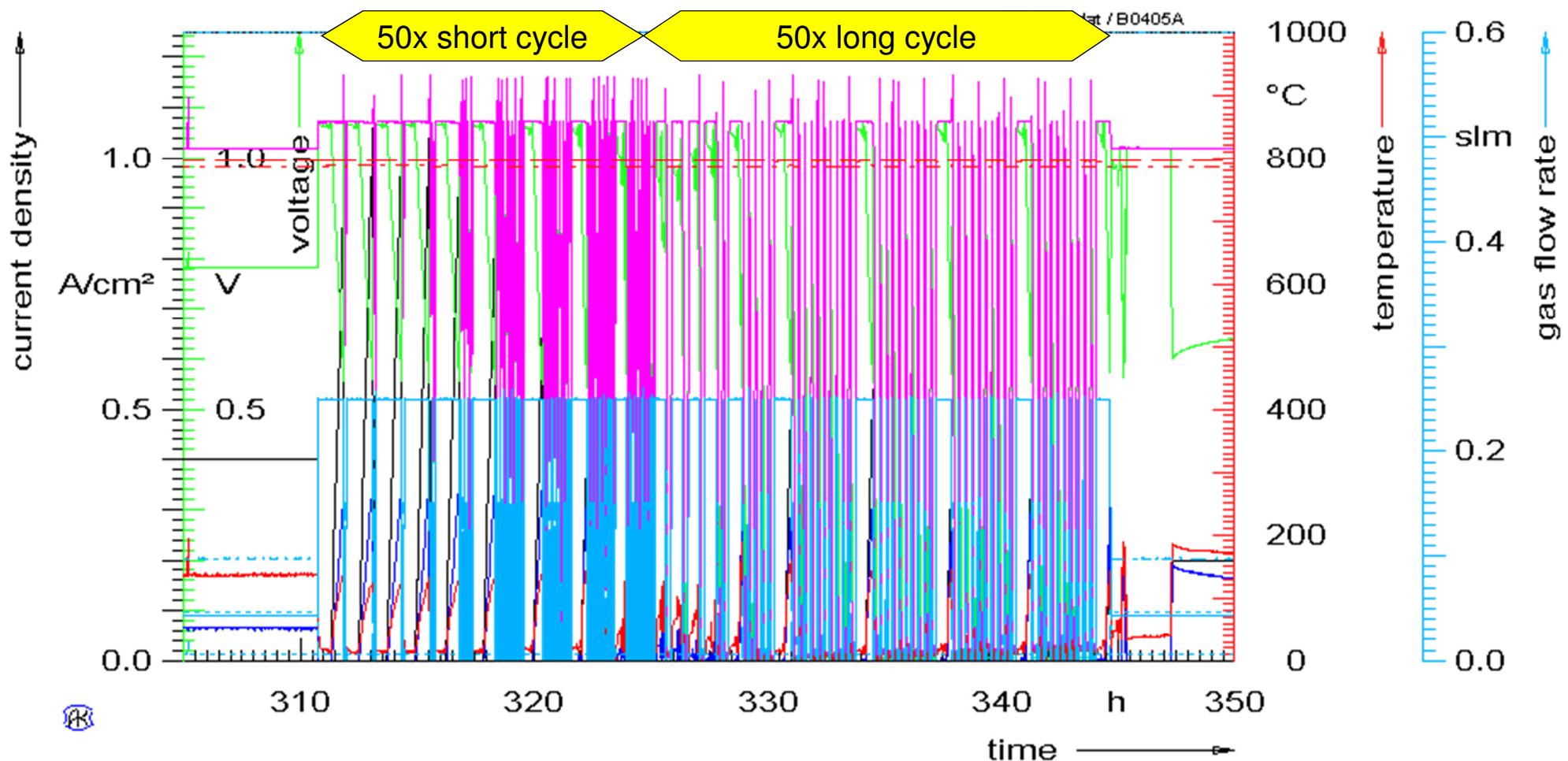


$T_{\text{sinter}} = 1400^\circ\text{C}$

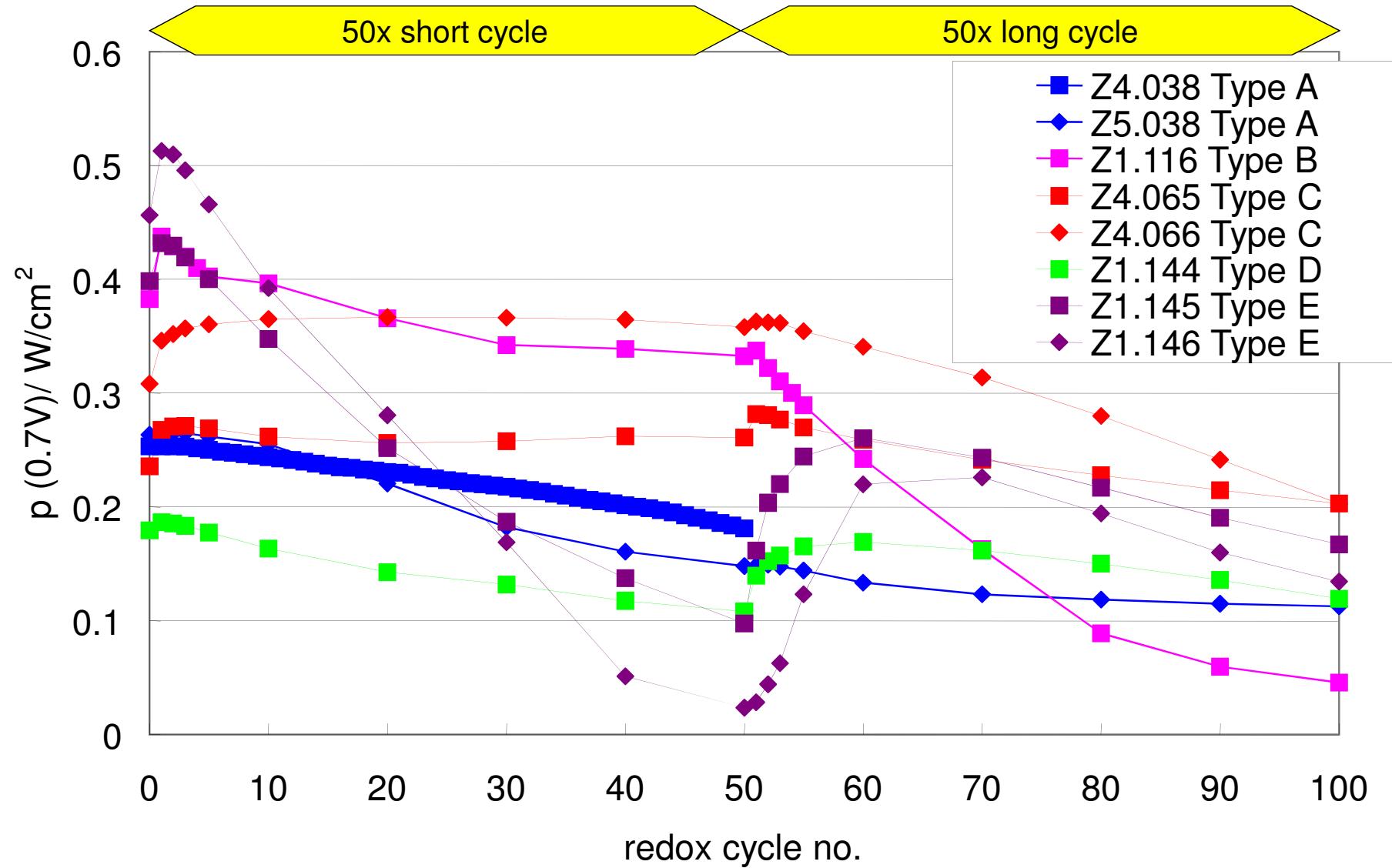
length change
 $\Delta l/l_0 = 1,7\%$



Degradation Tests for SOFC Redox Stability (100 Redox-Cycles in 35 h)

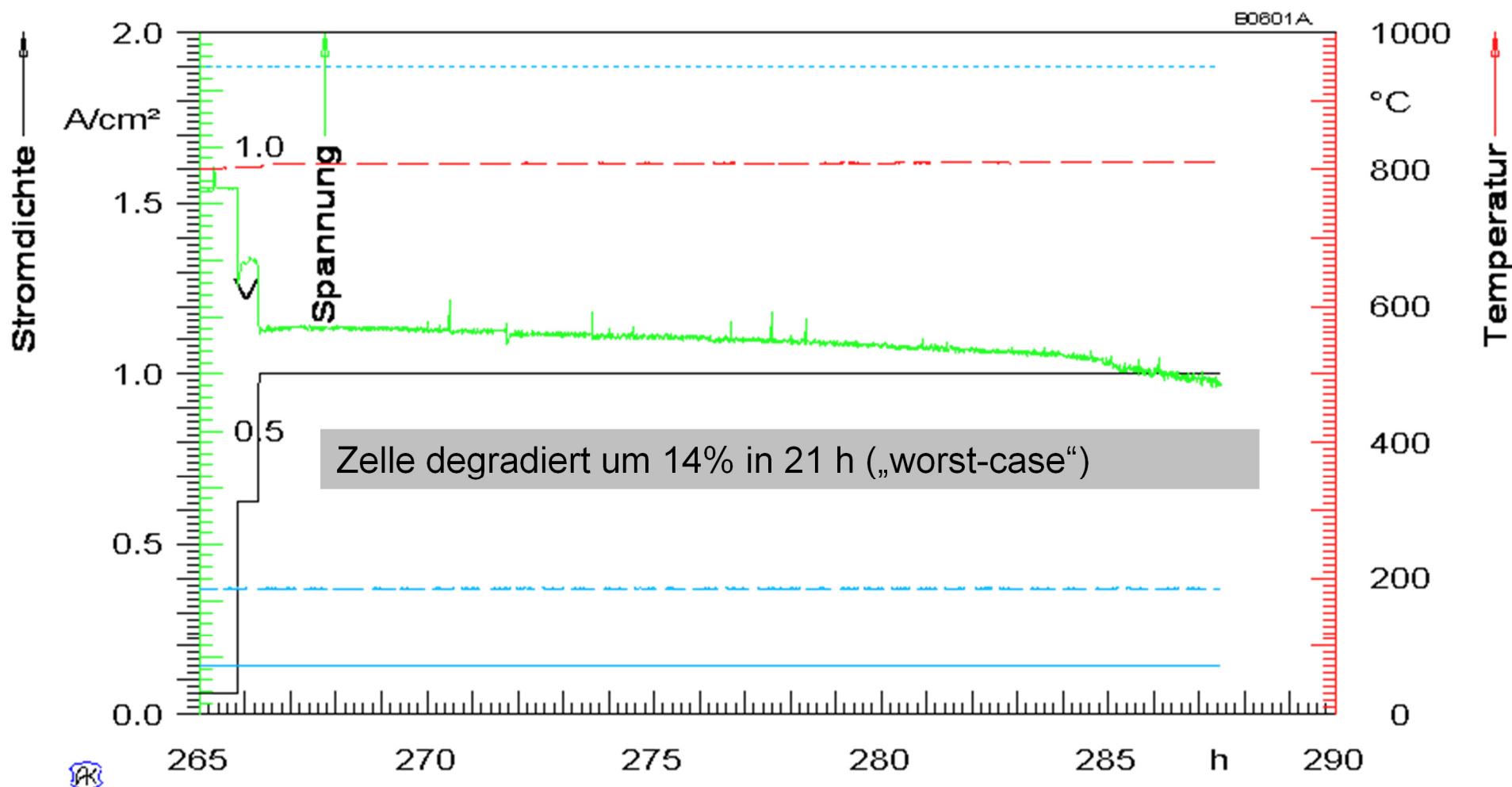


Redox Stability of Electrolyte Supported SOFC



Operation on Hydrocarbons

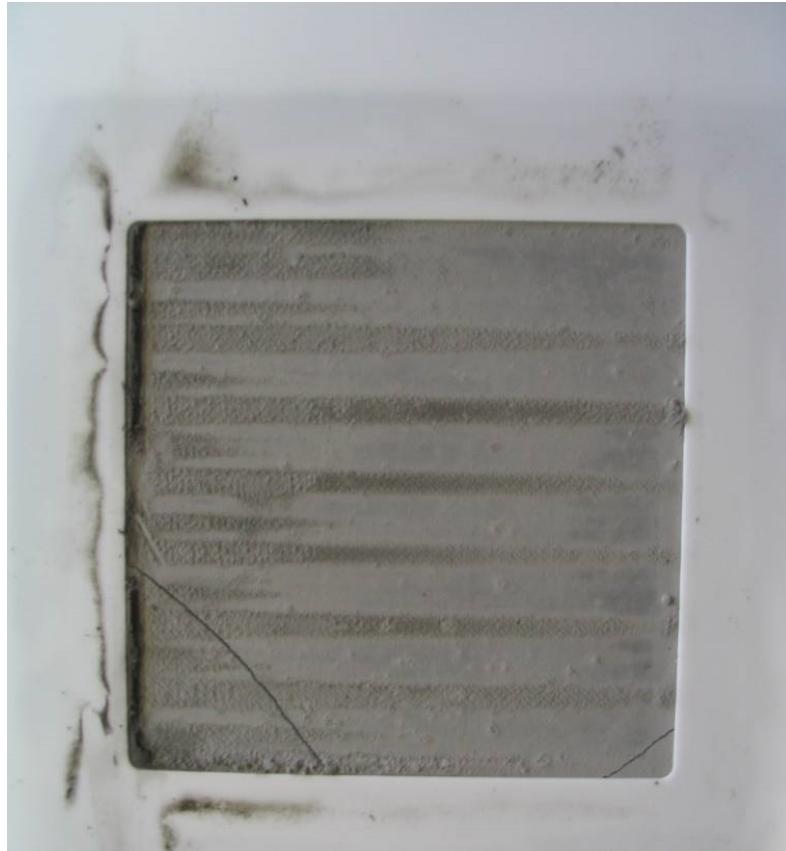
Severe Degradation with higher Hydrocarbons C_nH_m



Gasgemisch: H₂, CO, H₂O, CO₂, N₂, C₂H₂ (Ethin), C₇H₈ (Toluol), Methylnaphtalin (C₁₁H₁₀)

Operation on Hydrocarbons

Severe Degradation with higher Hydrocarbons C_nH_m

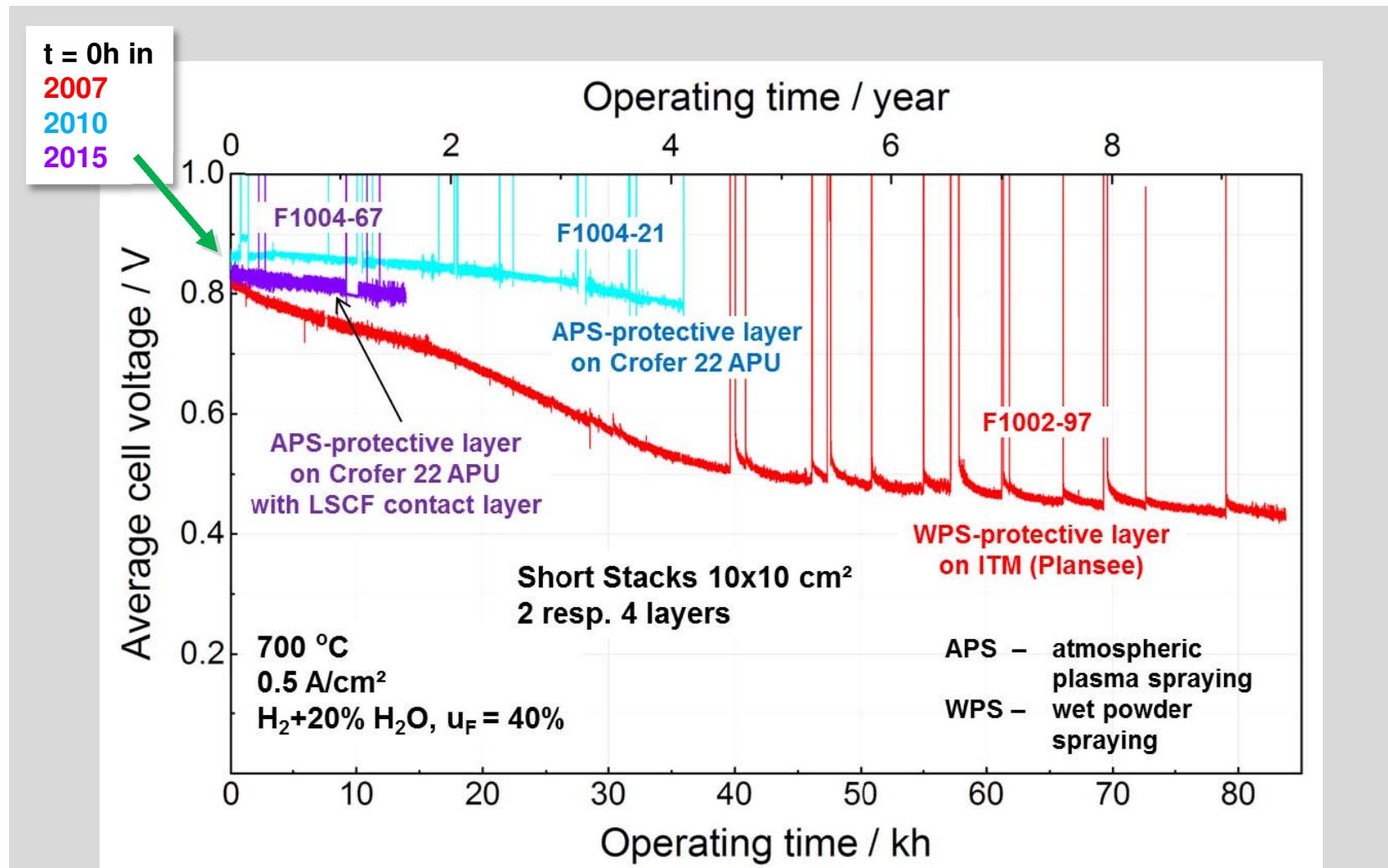


Flussrichtung
→

Aufgekohlte Zelle nach Betrieb mit einem Gasgemisch bestehend aus aus H₂, CO, H₂O, CO₂, N₂, C₂H₂ (Ethin), C₇H₈ (Toluol) und Methylnaphtalin (C₁₁H₁₀)

Durability Testing of Solid Oxide Cells

Research Center Jülich ASC-Stacks (2007 ff)



duration
> 80 000 h
~ 10 years
2007 – today

degradation
0.3 ... 0.6 % /
1 000 h

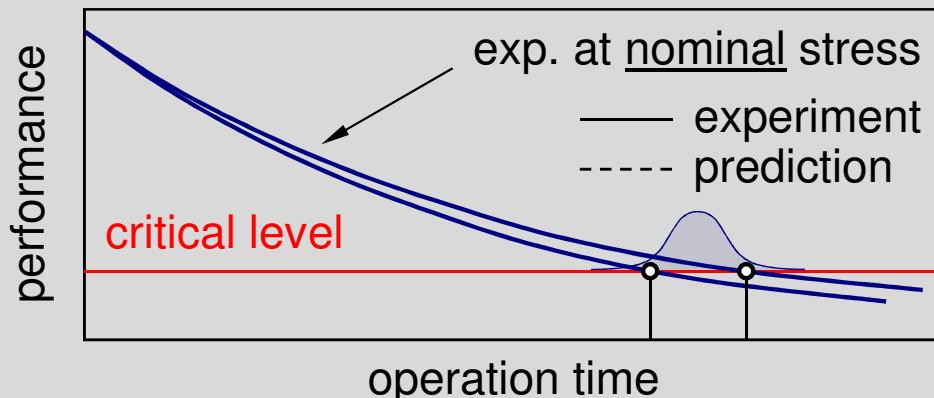
results
available after
5...10 years



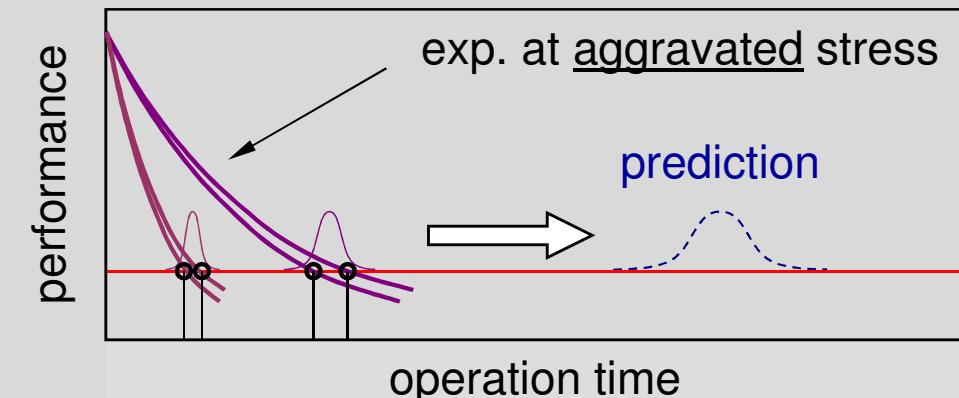
**accelerate
testing !**

L. Blum et al., *ECS Trans.*, **78** (1), p. 1791 (2017)

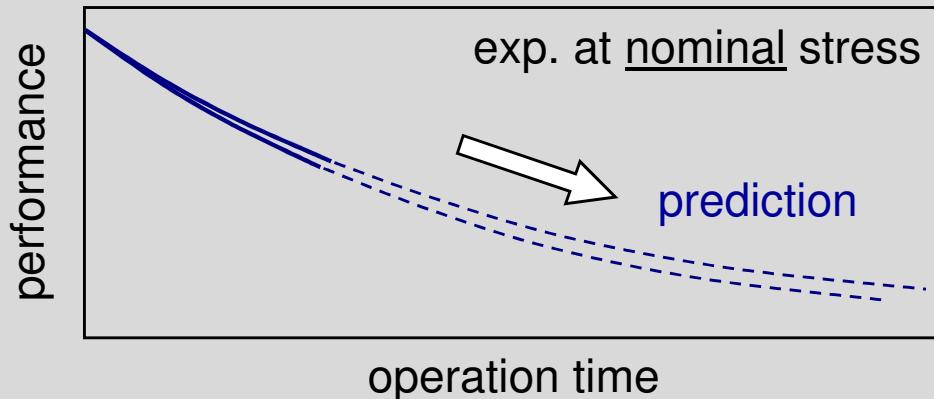
"conventional" life testing



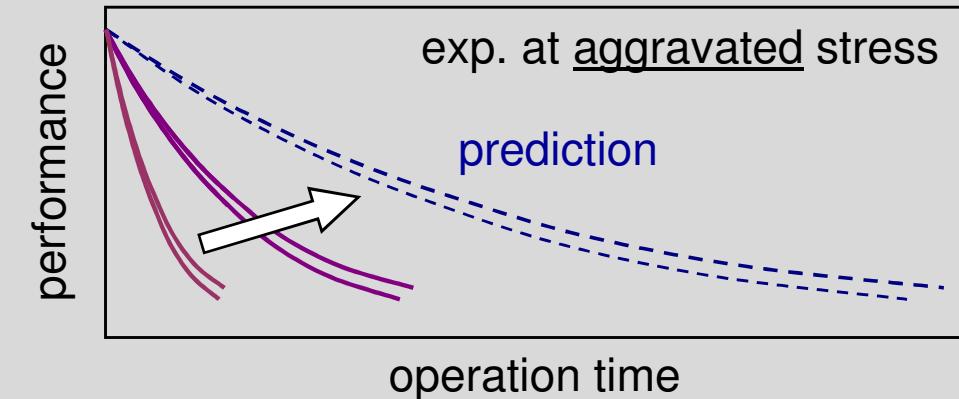
accelerated life testing



degradation testing



accelerated degradation testing



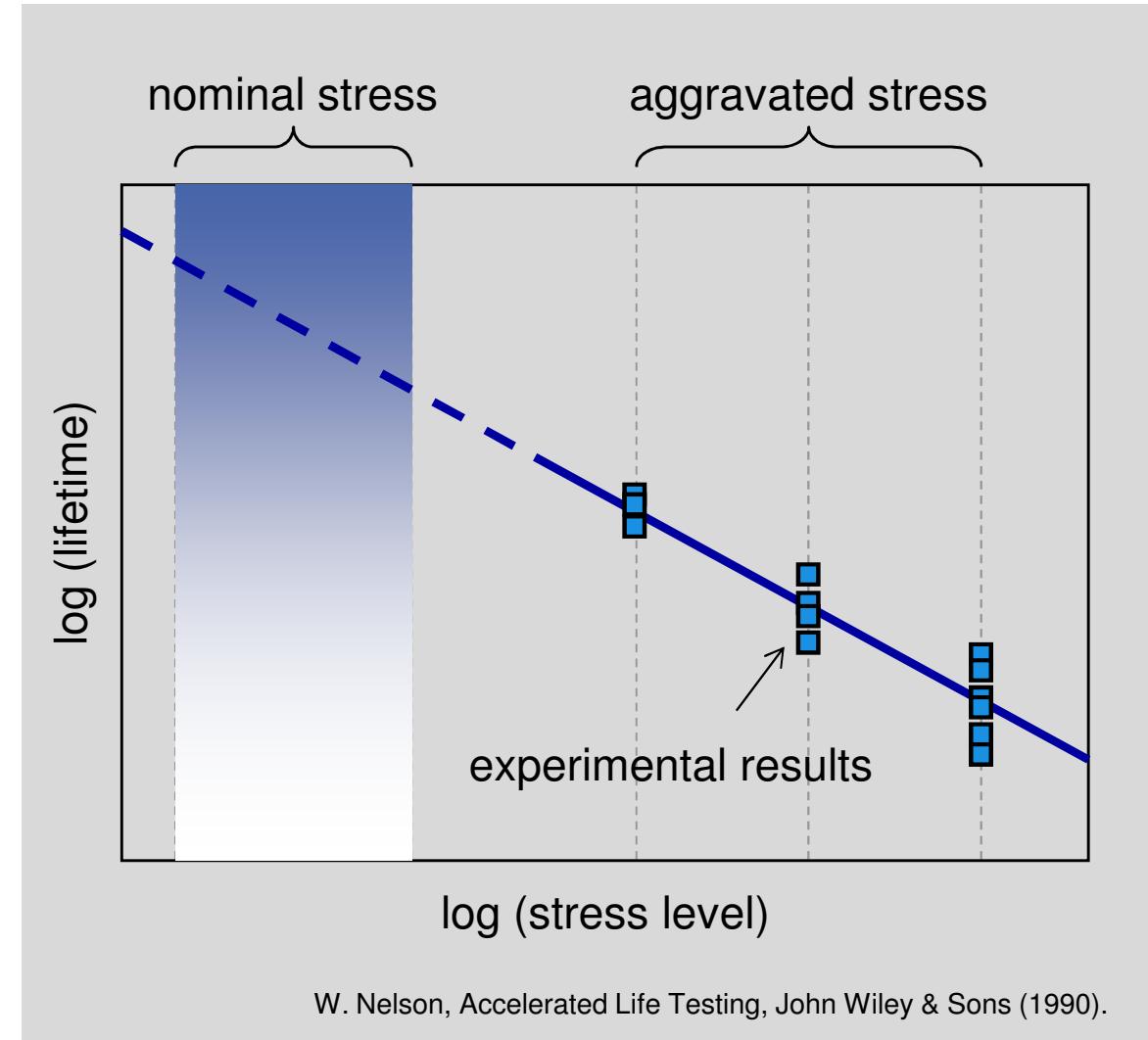
Accelerated Testing Approach and Advantages

approach

- degradation is accelerated by means of aggravated stress (current load, temperature, etc.).
- life-stress relationship is modelled by the use of failure data.
- extrapolation of life-stress model gives a prediction for service life at nominal stress.

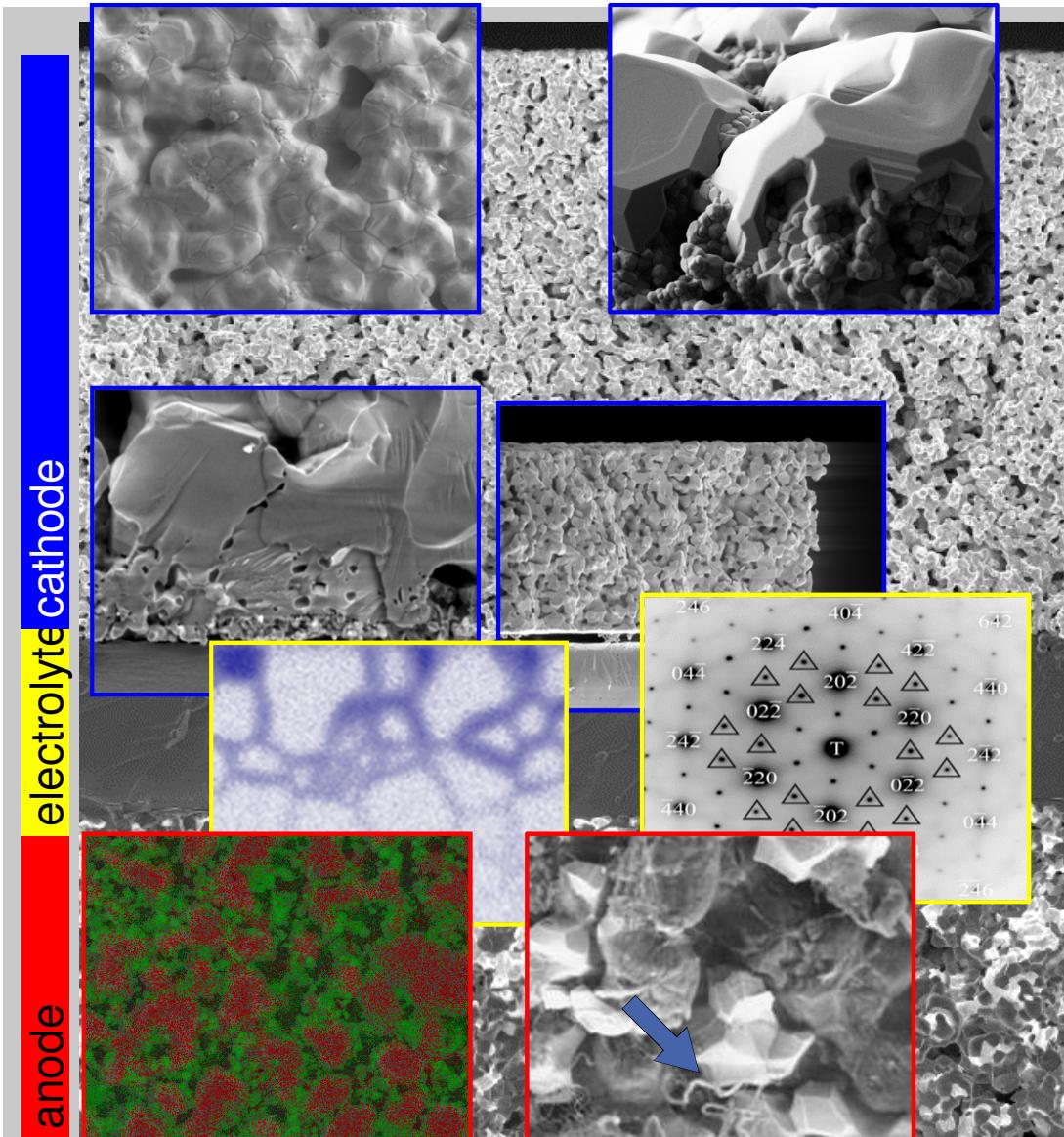
advantages of ALT

- physical consequences of degradation appear more clearly, mechanisms are identified more easily.
- precious measurement time is saved, lifetime is evaluated rapidly.



Accelerated Life Testing

Competing Degradation Processes in Solid Oxide Cells



in SOC many potentially competing degradation mechanisms are known:

- densification of electrodes
 - formation of micropores
A. Weber et. al., *Denki Kagaku* **64**, pp. 582-589 (1996).
 - formation of chromium compounds on the cathode
 - delamination of the cathode
M. J. Heneka et. al., Proc. 9th Int. Symp. on SOFC, pp. 534-543 (2005).
 - Mn-interdiffusion
A. Weber, in J. Garche (Ed.), *Encyclopedia of Electrochemical Power Sources*, Amsterdam: Elsevier, pp. 120-134 (2009).
 - intrinsic electrolyte degradation
B. Butz et. al., *Solid State Ionics* **177**, pp. 3275-3284 (2006).
 - Ni-agglomeration
A. C. Müller et. al., Proc. 3rd European SOFC Forum, pp. 353-362 (1998).
 - carbon deposition
E. Ivers-Tiffée et. al., *Handbook of Fuel Cells – Fundamentals, Technology and Applications*, pp. 933-956 (2009).
- ⇒ in addition, activation processes that improve the performance take place
A. Weber et. al., *Denki Kagaku* **64**, pp. 582-589 (1996).

Accelerated Life Testing

Competing Degradation Processes

aging mechanism A

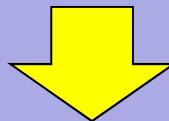
- experimental results
- life-stress model
- - - extrapolation

aging mechanism B

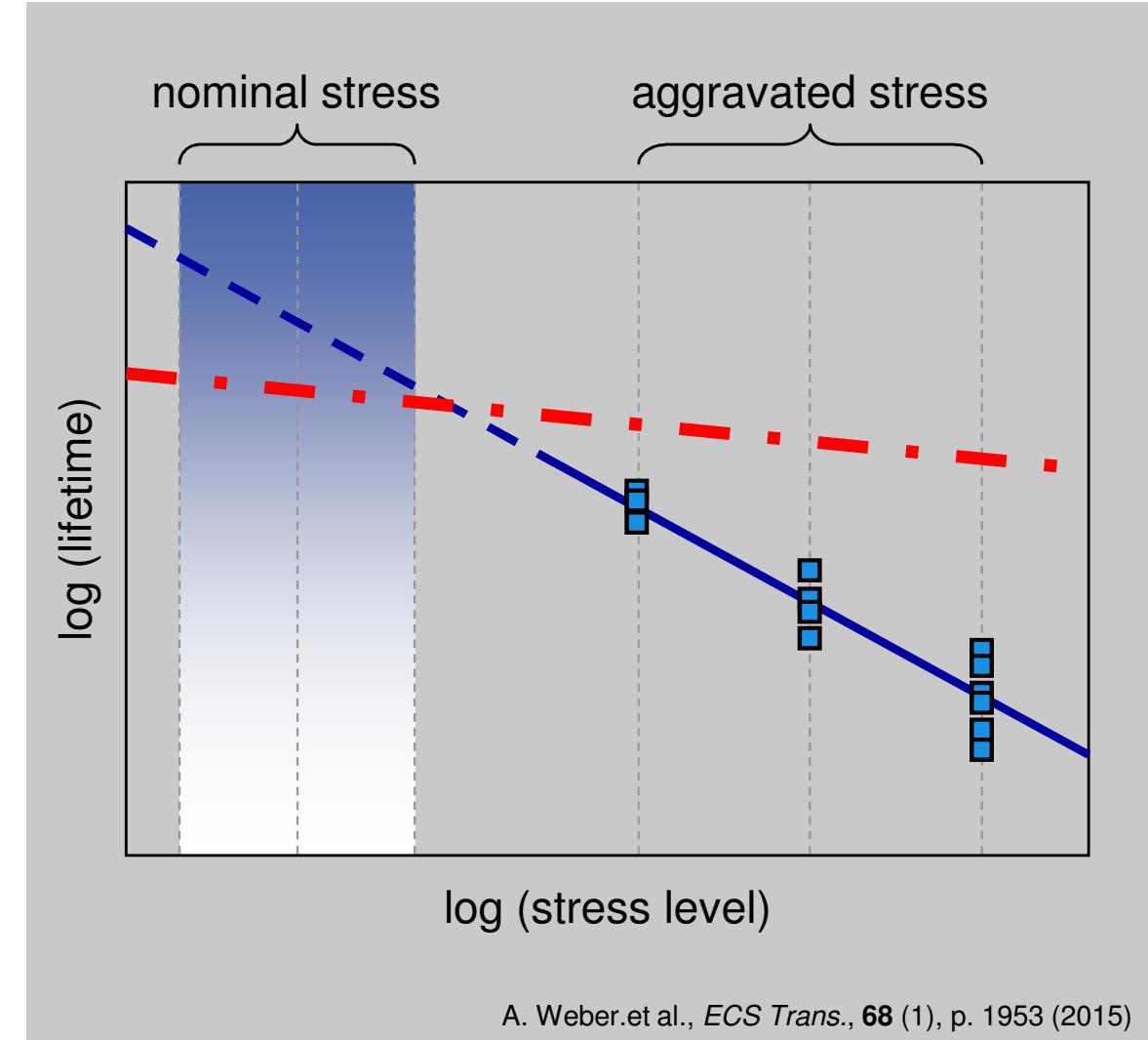
- · different life-stress dependency

aging mechanism B

- different dependency on stress level
- minor impact at aggravated stress
- lifetime limiting at nominal stress



ALT approach will fail



Degradation Testing

Deconvolution of Competing Degradation Processes

aging mechanism A

- experimental results
- life-stress model
- - extrapolation

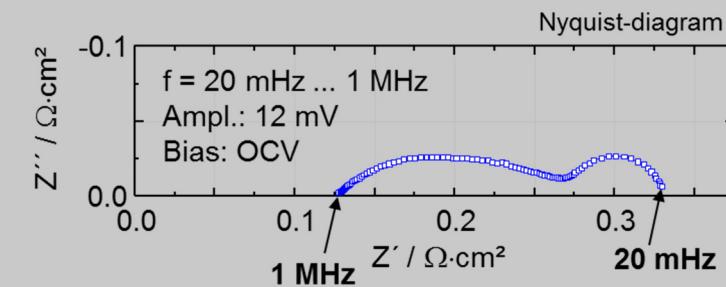
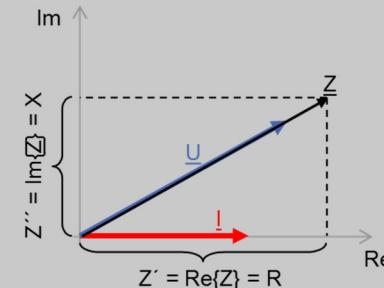
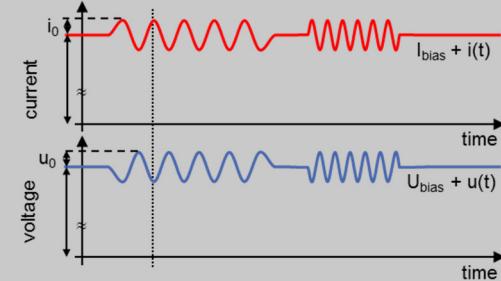
aging mechanism B

- · different life-stress dependency

→ deconvolute aging mechanisms

- ~~overall performance, lifetime
(summarized / averaged quantities)~~
- physicochemical processes acquired via their electrochemical impedance (additive quantities)

EIS: electrochemical impedance spectroscopy



electrochemical
impedance
spectrum

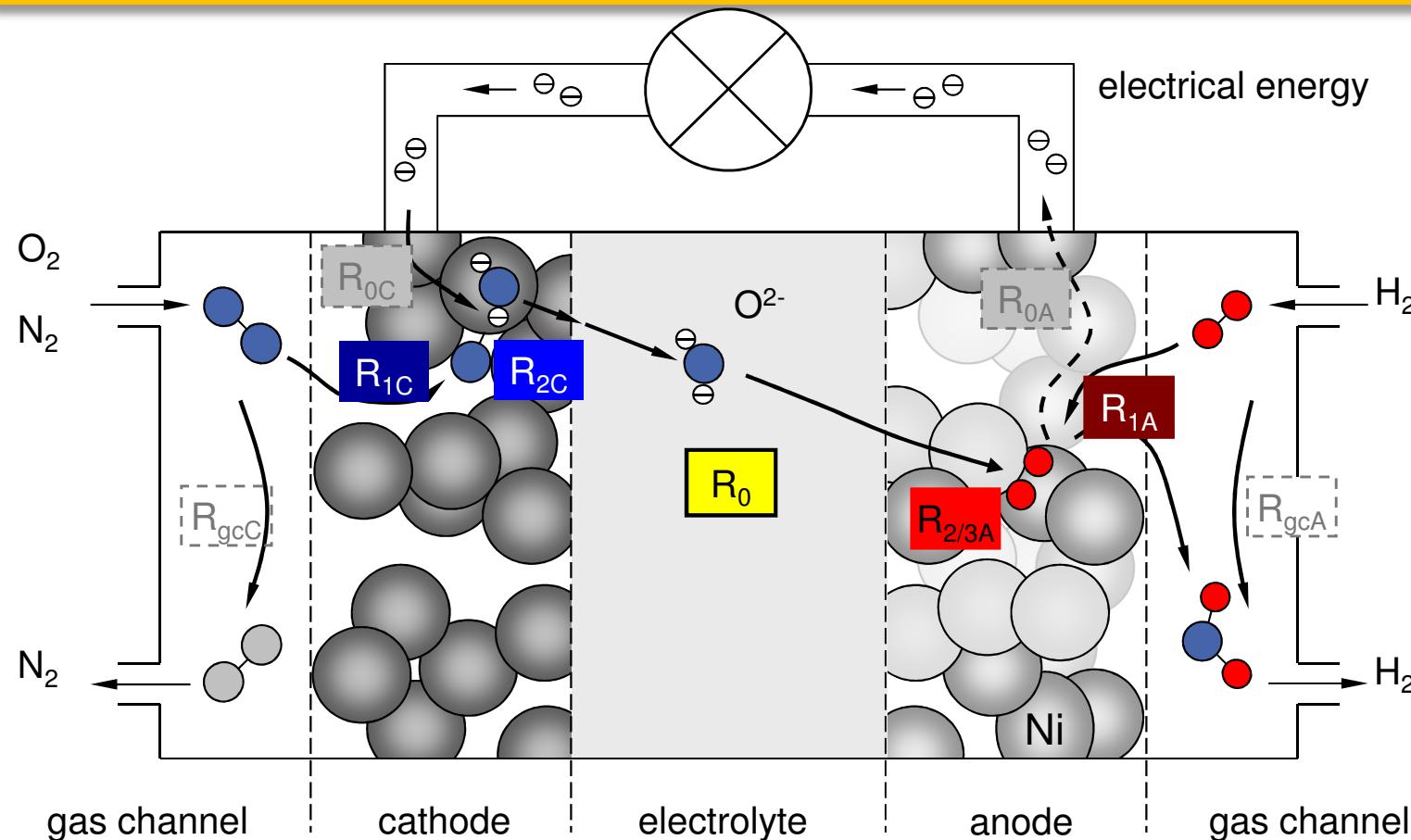
Loss mechanisms in SOFCs

Processes contributing to the Area Specific Resistance (ASR)

Transport in the gas phase: gas flow and diffusion

Transport in the solid phase: electronic and ionic current

Reactions: charge transfer, surface exchange, electrooxidation, surface catalysis



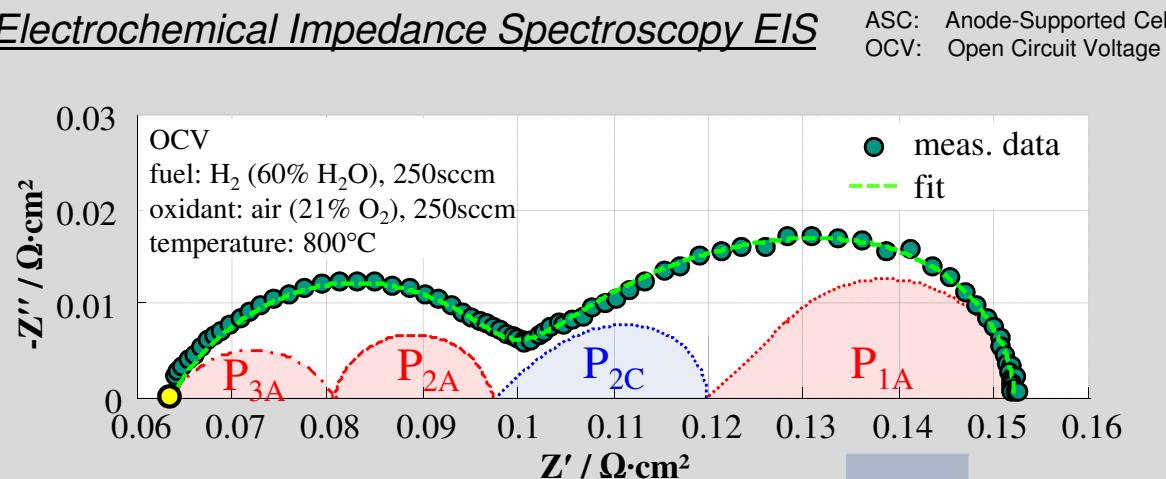
related resistances

R_{gcC}	gas conversion
R_{1C}	gas diffusion (cathode)
R_{0C}	electronic current (cathode)
R_{2C}	surface exchange and O^{2-} bulk diffusion
R_0	ionic current (electrolyte)
$R_{2/3A}$	electrooxidation and ionic transport in the anode
R_{0A}	electronic current (anode)
R_{1A}	gas diffusion (anode)
R_{gcA}	gas conversion

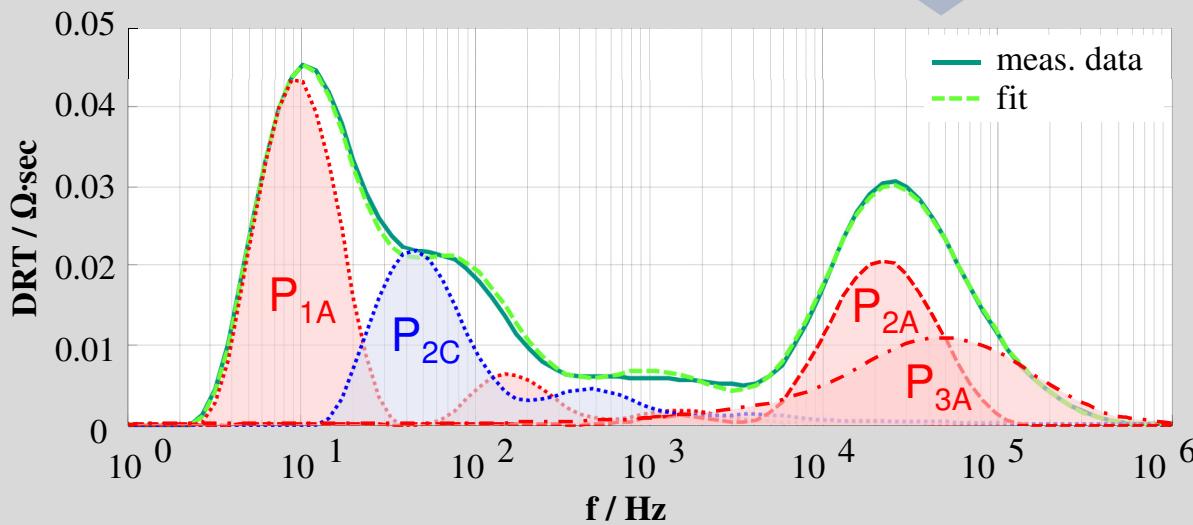
Impedance based Cell Model I

Deconvolution of electrochemical Processes

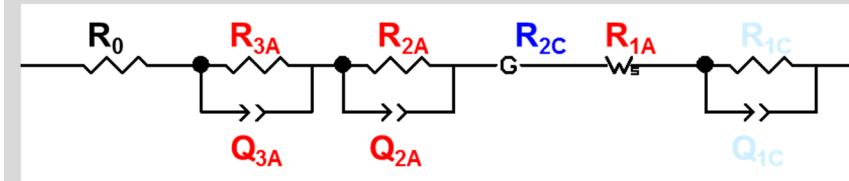
Electrochemical Impedance Spectroscopy EIS



Distribution of Relaxation Times DRT $\underline{Z}(\omega) = R_{Pol} \int_0^\infty \frac{\gamma(\tau)}{1+j\omega\tau} d\tau$



equivalent circuit model



process physicochemical origin

P_{1C} gas diffusion (cathode)

P_{2C} oxygen reduction reaction (cathode)

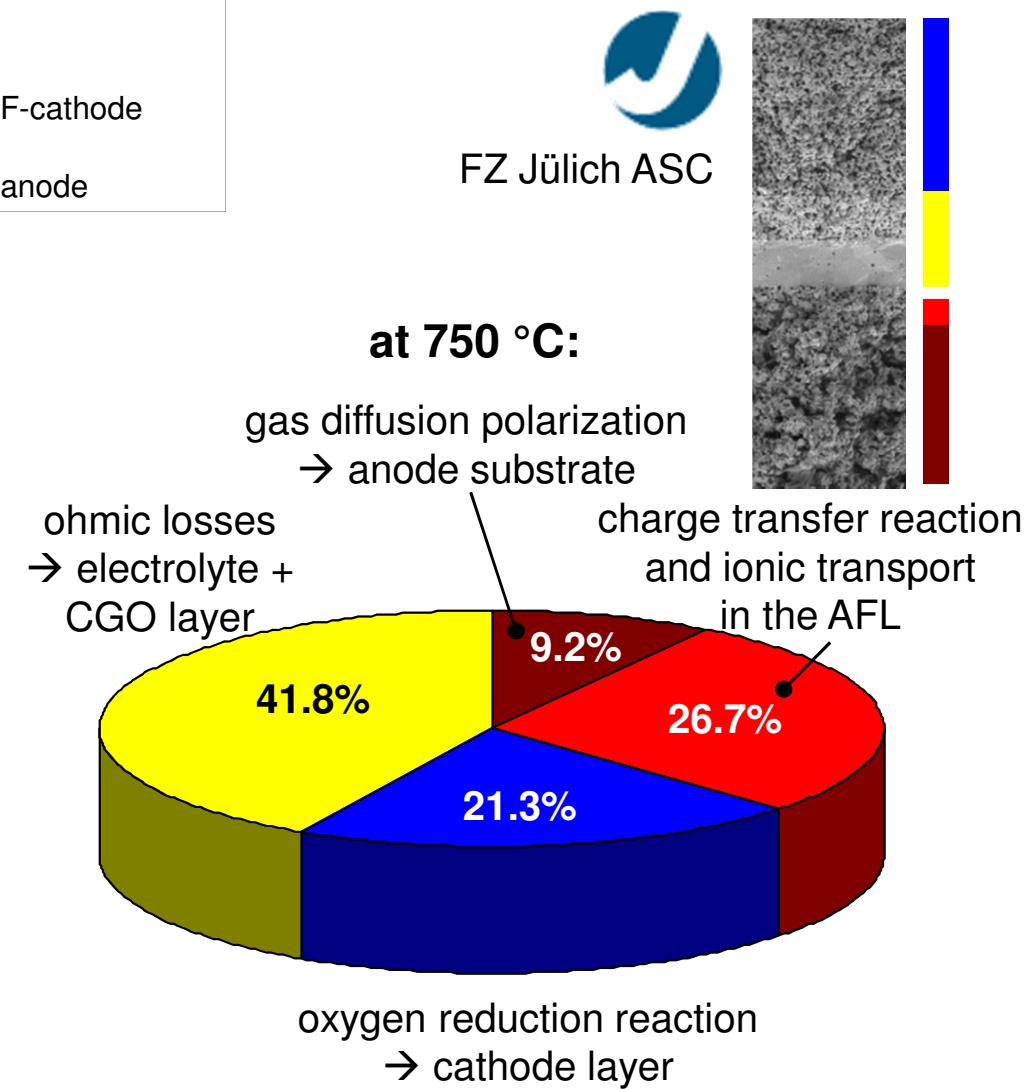
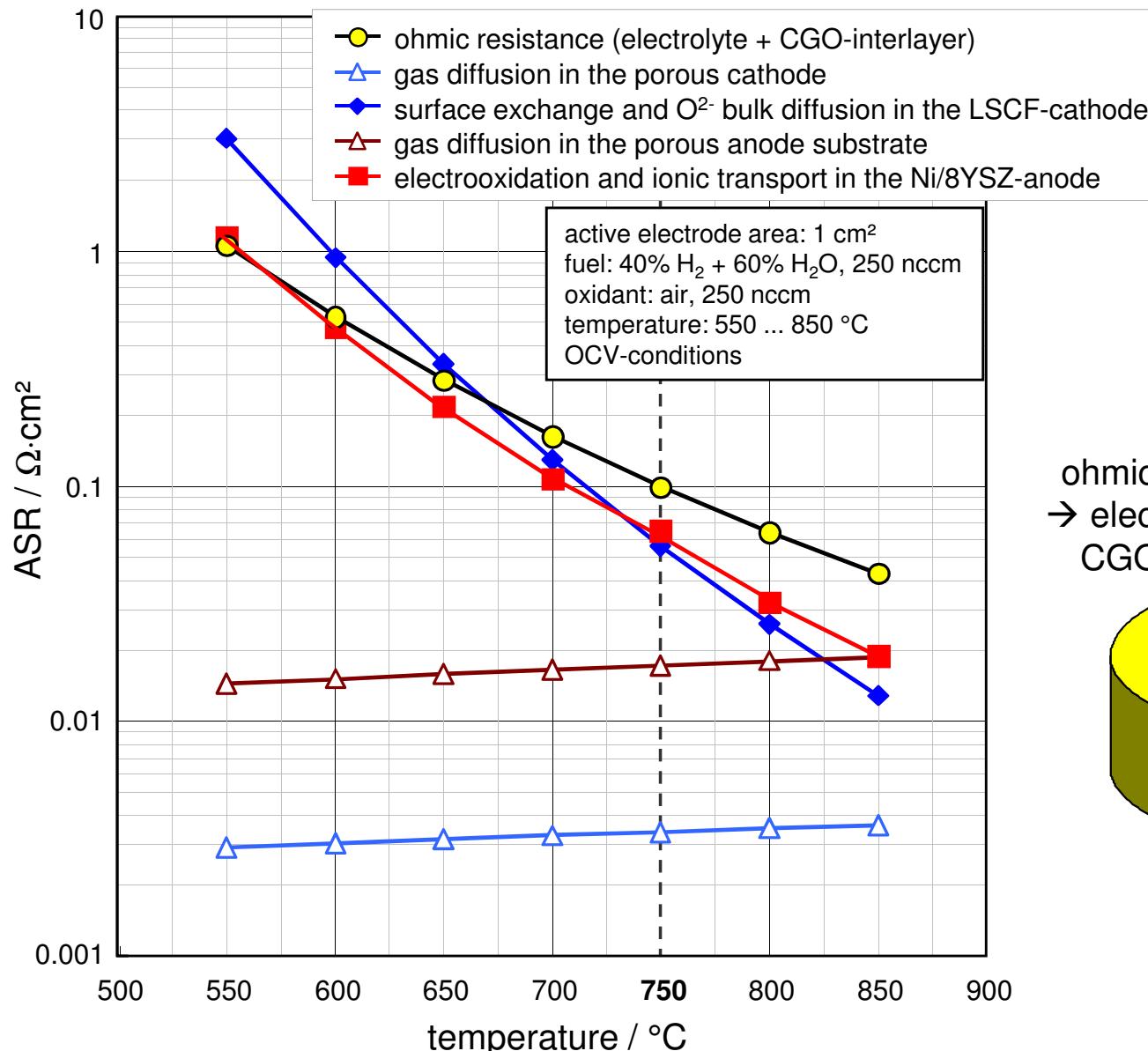
R_0 ionic (and electronic) conduction

P_{2A} / P_{3A} hydrogen electrooxidation coupled with gas diffusion and ionic transport

P_{1A} gas diffusion (porous substrate)

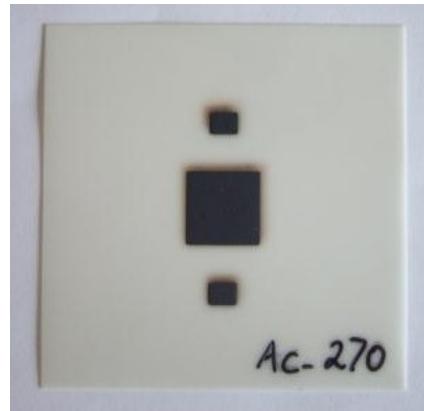
Model Application

ASRs of the Cell Components

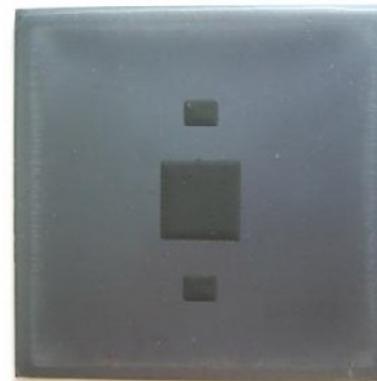


Applicability of the Model to different Cell Types

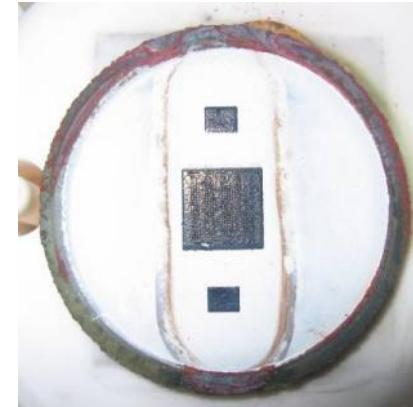
Cell Types analyzed in National and European Projects



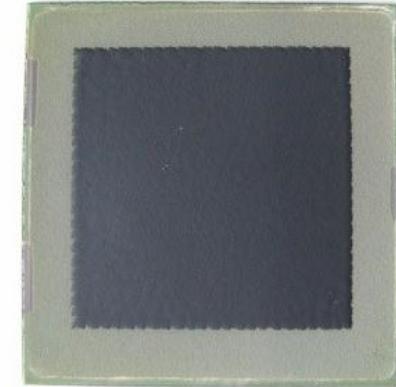
ESC



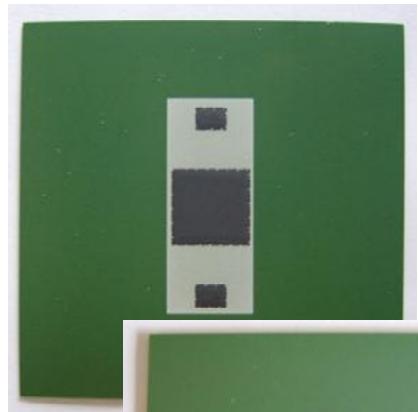
H.C. Starck



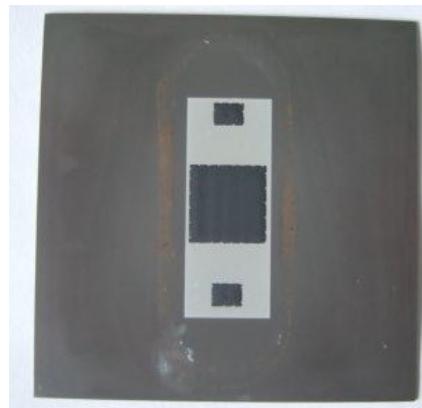
VPS-MSC



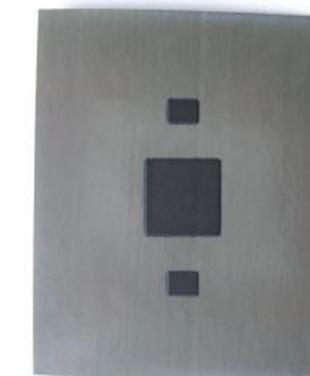
APS-MSC



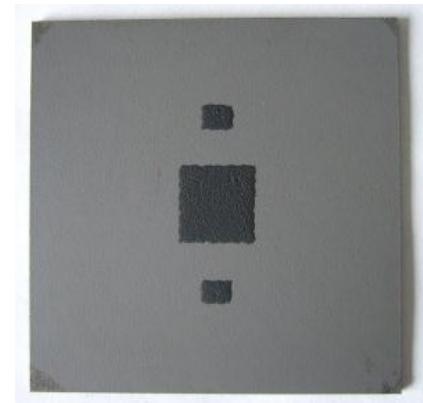
CeramTec
THE CERAMIC EXPERTS
SOLID POWER



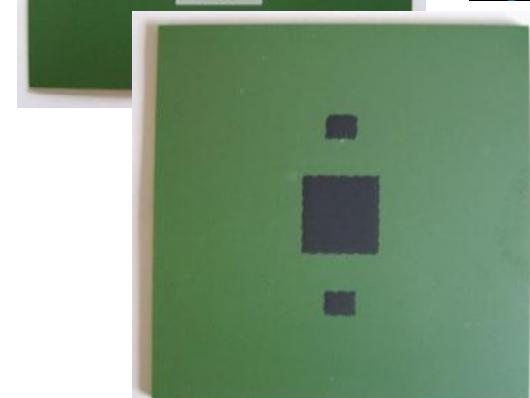
SrTiO₃-based ASC



cofired MSC



PVD MSC



ASC's (before reduction)

ASC (reduced)



- C. Endler-Schuck et al., *J. FC Science and Technology* **8**, p. 41001 (2011)
- Q. Ma et al., *J. Power Sources* **196**, p. 7308 (2011)
- P. Blennow et al., *J. Power Sources* **196**, p. 7117 (2011)
- F. Han et al., *J. Power Sources* **218**, p. 157 (2012)
- T. Franco et al., *Proc. 10th Eur. SOFC Forum*, p. A0906 (2012)
- A. Kromp et al., *Fuel Cells* **13**, p. 598 (2013)

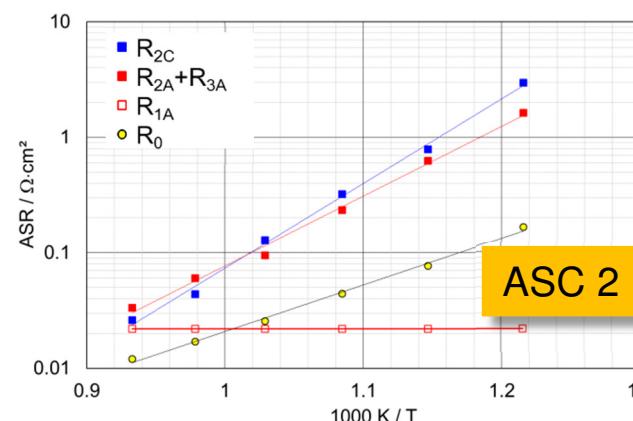
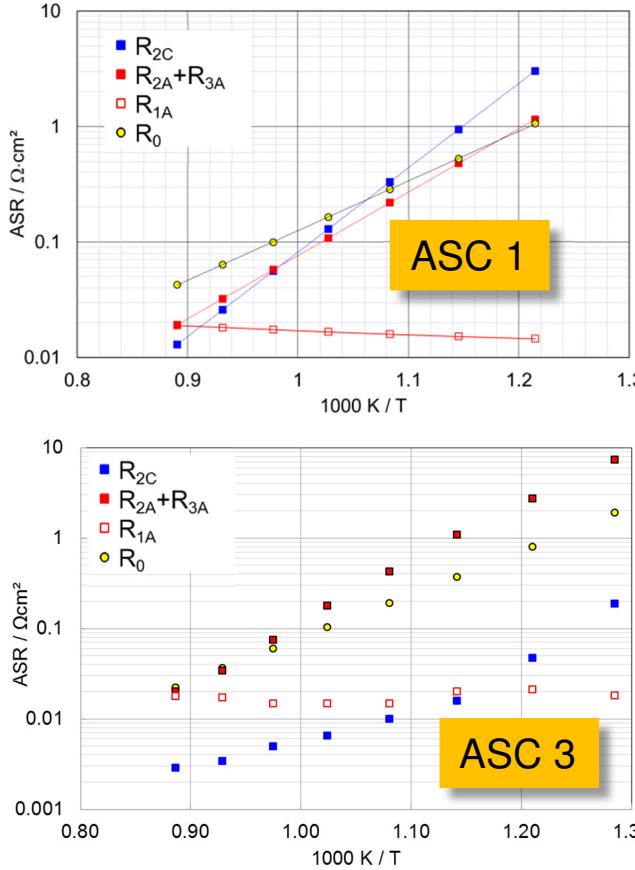


DTU



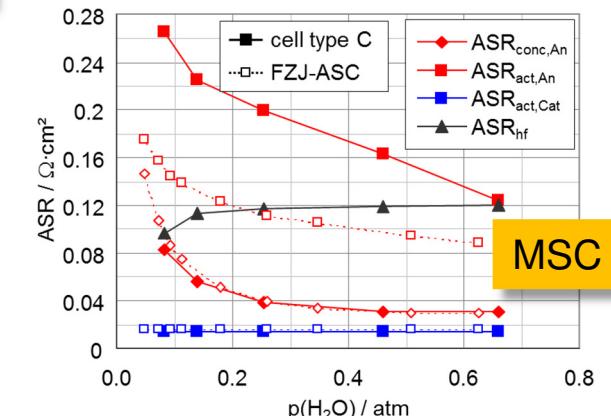
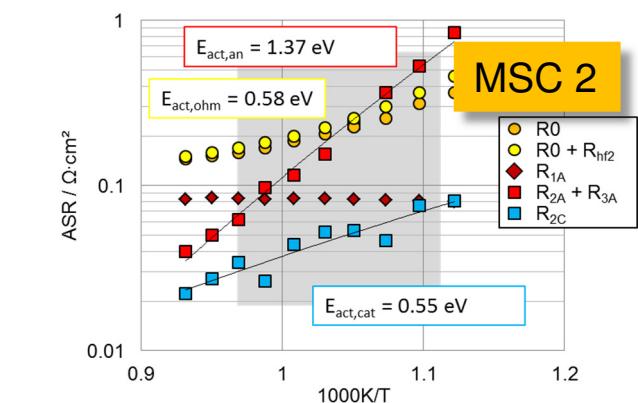
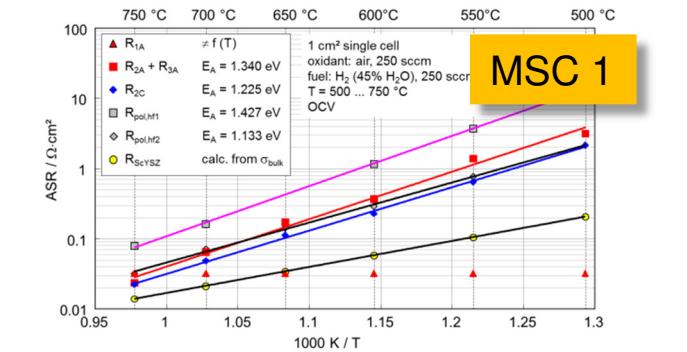
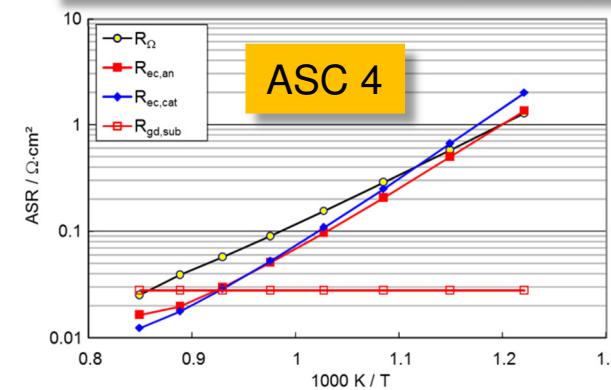
PLANSEE

Applicability of the Model to different Cell Types



quite similar processes in all cells

- gas diffusion in the pores
- electronic / ionic currents in solids
- a MIEC-cathode
- a cermet anode
- + polarization at insulating interlayers



- P. Blennow et al., *J. Power Sources* **196**, p. 7117 (2011)
 F. Han et al., *J. Power Sources* **218**, p. 157 (2012)
 T. Franco et al., *Proc. 10th Eur. SOFC Forum*, p. A0906 (2012)
 A. Kromp et al., *Fuel Cells* **13**, p. 598 (2013)

Impedance based Cell Model II

Physicochemical Performance Model → Power Density

model equations with 13 parameters from EIS

Nernst equation U_{OCV}

$$\text{ohmic loss } R_{ohm} = \frac{T}{B_{ohm}} \cdot \exp\left(\frac{E_{act,ohm}}{R \cdot T}\right)$$

Butler-Volmer equation

$$j = j_{0,El} \left[\exp\left(\alpha_{El} \frac{n_e F \eta_{act,El}}{RT}\right) - \exp\left(-(1 - \alpha_{El}) \frac{n_e F \eta_{act,El}}{RT}\right) \right]$$

$$j_{0,an} = \gamma_{an} (pH_{2,an})^a (pH_2 O_{an})^b \exp\left(-\frac{E_{act,an}}{RT}\right)$$

$$j_{0,cat} = \gamma_{cat} (pO_{2,cat})^m \exp\left(-\frac{E_{act,cat}}{RT}\right)$$

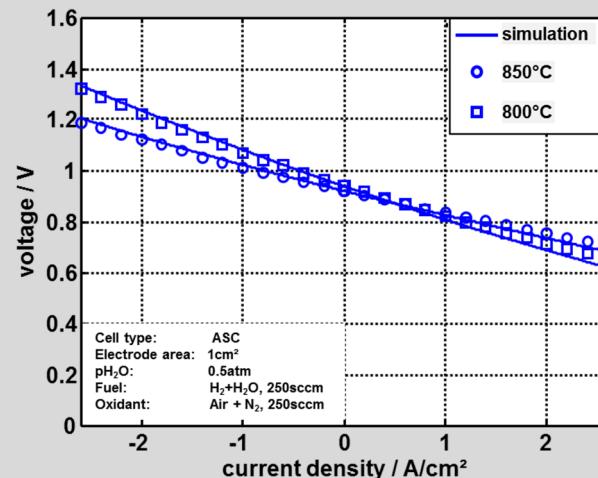
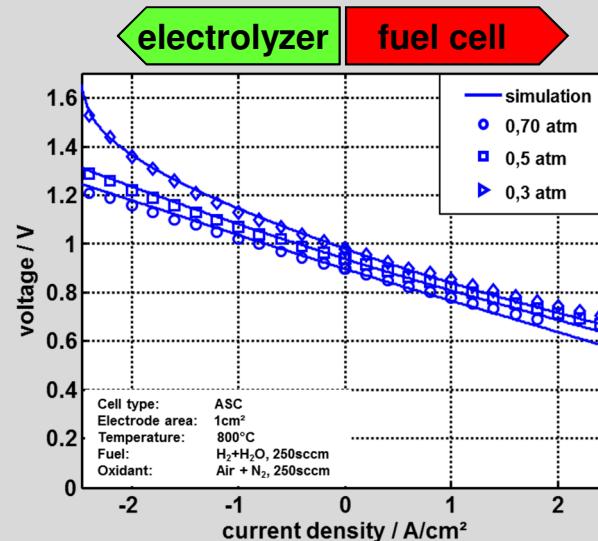
Fick's law inserted + Nernst equation with $D_i^{eff} = \psi_{el} \cdot D_i$

Cell voltage $U_{cell}(j) = U_{OCV} - (\eta_{ohm}(j) + \eta_{conc,an}(j) + \eta_{conc,cat}(j) + \eta_{act,an}(j) + \eta_{act,cat}(j))$

durability model ?

model parameters $p_i = f(\text{time, operating conditions})$

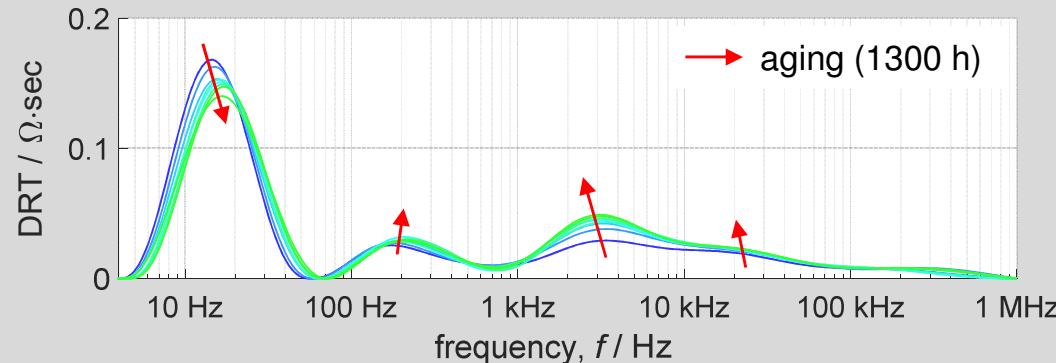
model validation



- excellent agreement
- different temperatures
- different gases
- fuel cell and electrolyzer mode

Modelling III

Single Cell Durability Model

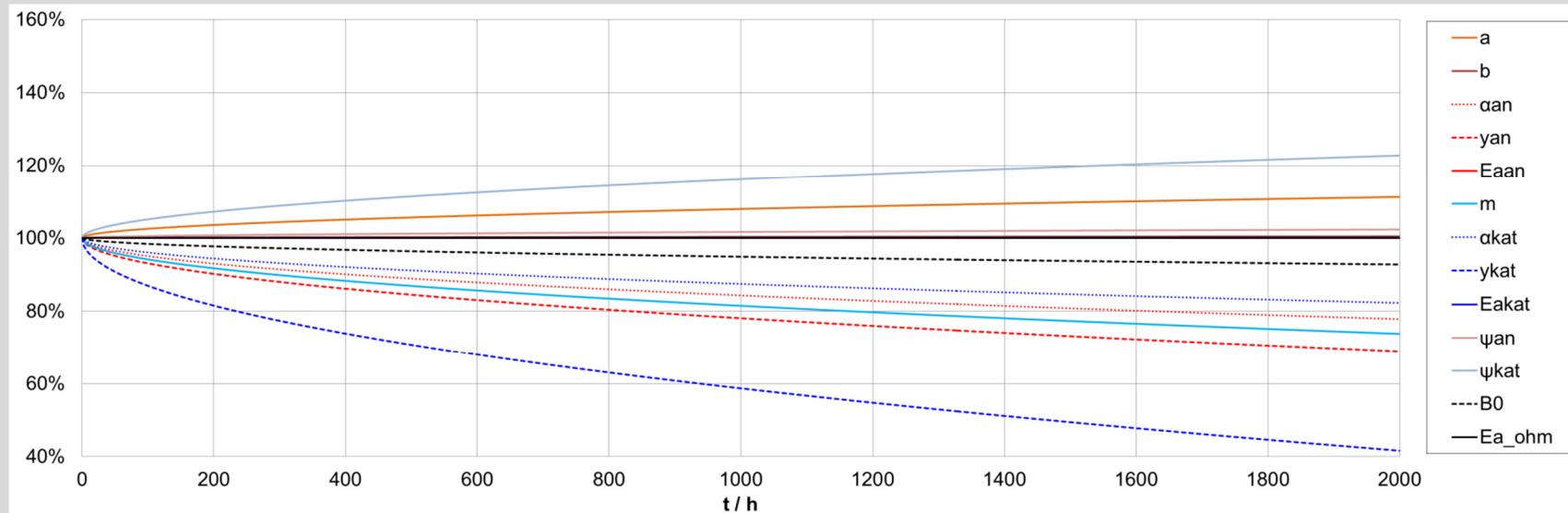


test matrix (varied parameters)

- temperature
- current density
- fuel composition
- oxidant composition
- ... to be extended

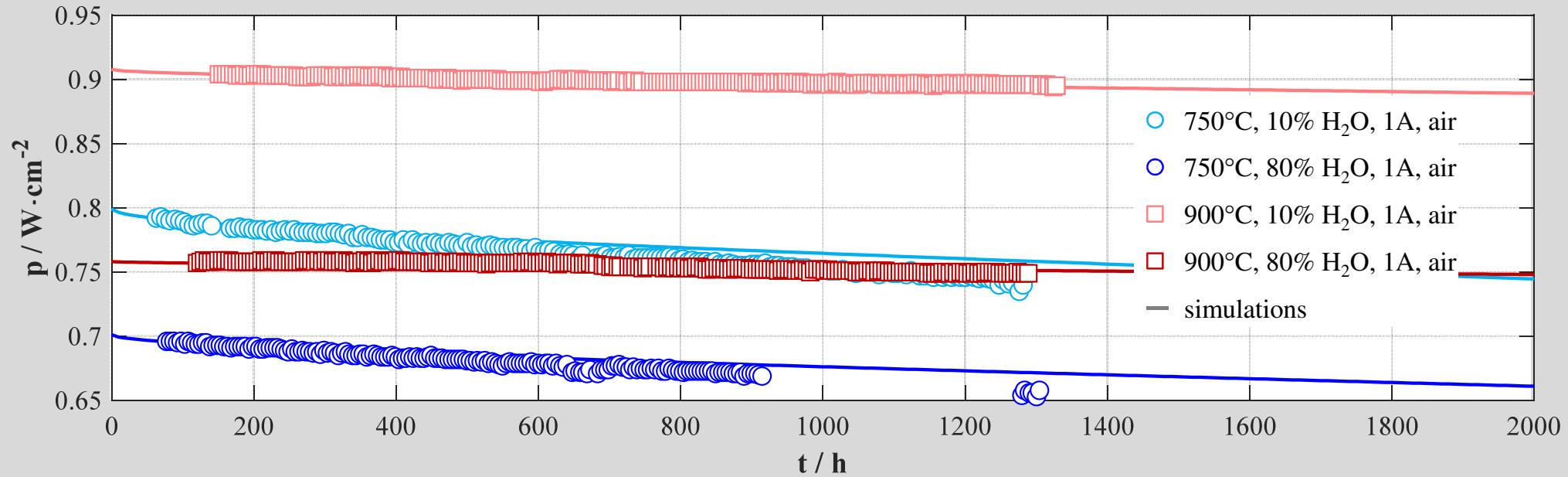
test duration: 1300 h (> 4000 h)

time dependency of model parameters $\rightarrow \sqrt{t}$



Single Cell Durability Model Validation

measured and simulated cell degradation

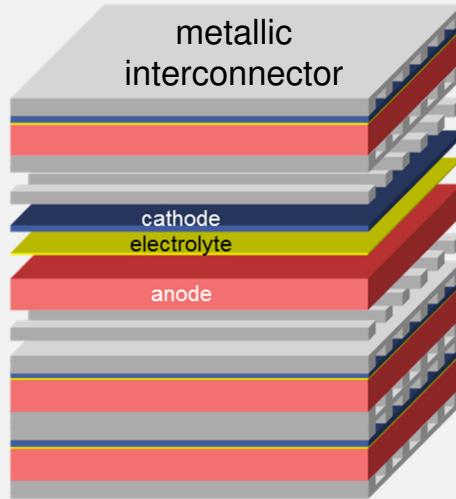


- simulations in good agreement with the measurements
- error < 2.5 mW·cm²

SOFC Stacks

Spatial Gradients → Localized Degradation

planar SOC stack

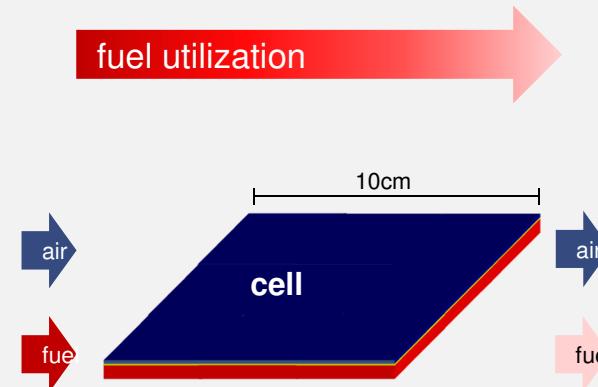


- in plane conduction
- gas conversion
- heat generation
- heat dissipation



spatial gradients

spatial gradients



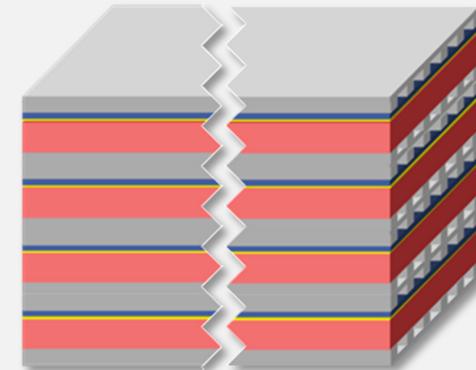
fuel utilization

temperature

electromotive force

current density

degradation model

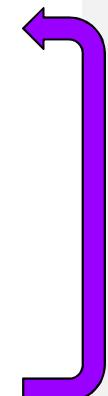


local operating conditions

local degradation

temporal change of local cell parameters

temporal change of local operating conditions

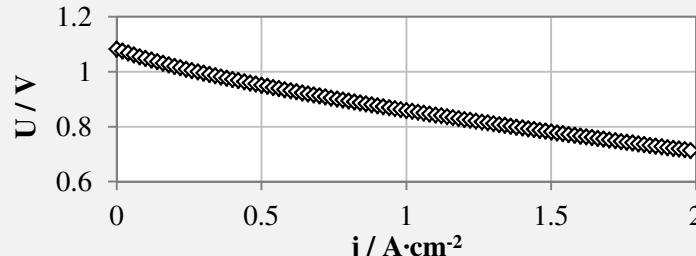


Modelling IV

FEM Repeat Unit Model → space-resolved Performance

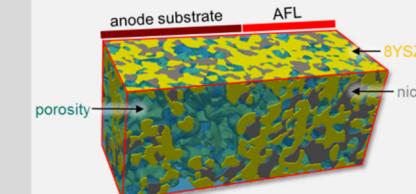
parametrization

0D stationary performance model¹

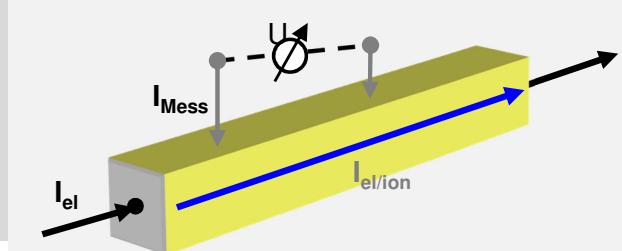


13 parameters

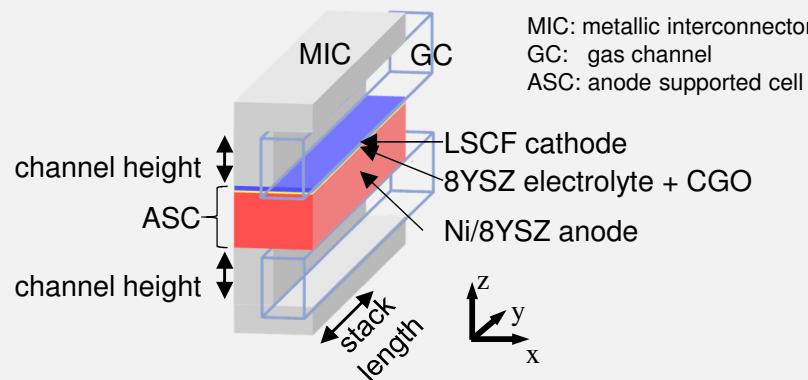
3D reconstruction^{2,3}



Conductivity measurements⁴



RPU of a stack layer



model equation^{5,6,7}

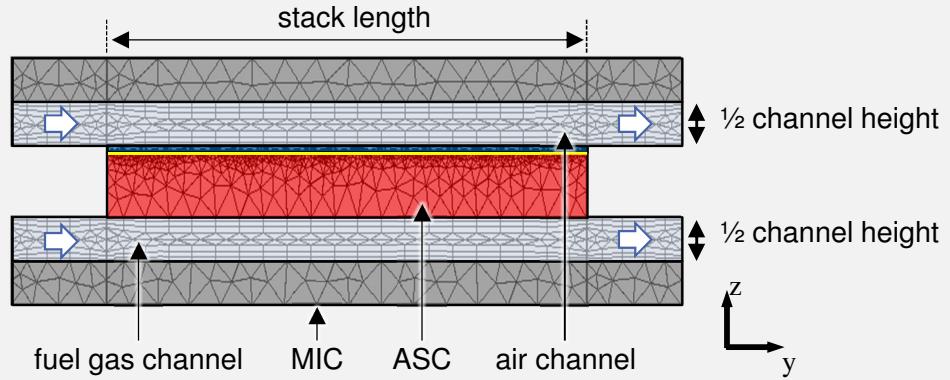
impulse balance – Navier Stokes

$$\rho(\vec{U} \cdot \nabla)\vec{U} = -\nabla p + \nabla \left(\eta (\nabla \vec{U} + (\nabla \vec{U})^T) \right)$$

electrochemistry – Butler-Volmer

$$j_{ct,el} = j_{0,el} \left[\exp \left(\alpha_{el} \frac{n_e F \cdot \eta_{act,el}}{RT} \right) - \exp \left(-(1 - \alpha_{el}) \frac{n_e F \cdot \eta_{act,el}}{RT} \right) \right]$$

2D model geometry



mass balance – dusty gas

$$\frac{N_i}{D_{i,Kn}} + \sum_{i \neq j} \frac{x_j N_i - x_i N_j}{D_{i,j}^{eff}} = -\frac{1}{RT} \left[\nabla x_i p + x_i \nabla p \frac{\kappa p}{\mu D_{i,Kn}} \right]$$

energy balance

$$\nabla \cdot (\rho \vec{u} c_p T) = -\nabla \cdot \vec{q} + \dot{S}_Q$$

charge transport – ohmic law

$$\nabla j = -\nabla \cdot (\sigma_{ec/ion}(T) \cdot \nabla \Phi)$$

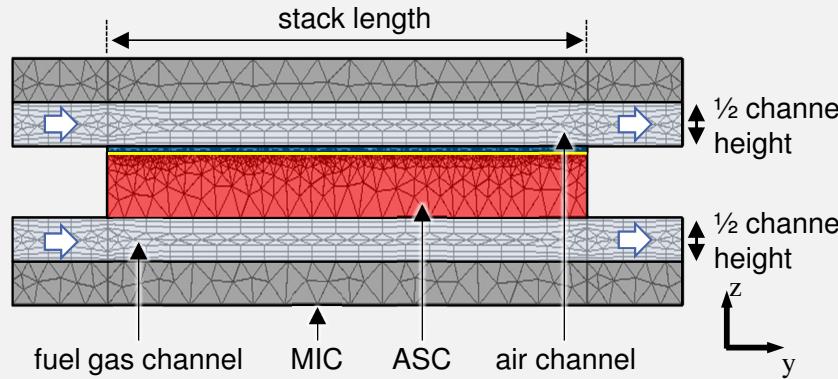
- [1] A. Leonide, et. al., *J. Power Sources*, **196**, p. 7343, (2011).
- [2] J. Joos, et. al., *J. Power Sources* **196**, pp. 7302-7307 (2011).
- [3] J. Joos, et. al., *J. Power Sources* **246**, pp. 819-830 (2014).
- [4] M. Kornely, et. al., *J. Power Sources* **196**, pp. 7209-7216 (2011).

- [5] H. Geisler, et. al., *J. Electrochem. Soc.*, **161** (6), p. F778-F788 (2014).
- [6] H. Geisler, et. al., *ECS Transaction*, **68** (1), p. 2151-2158 (2015).
- [7] N. Russner, et. al., *ECS Transaction*, **78** (1), p. 2673-2682 (2017).

Modelling V

Stack Durability Model

2D FEM model

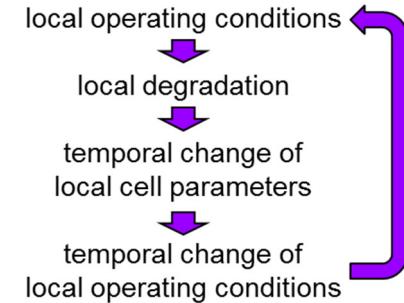


geometry

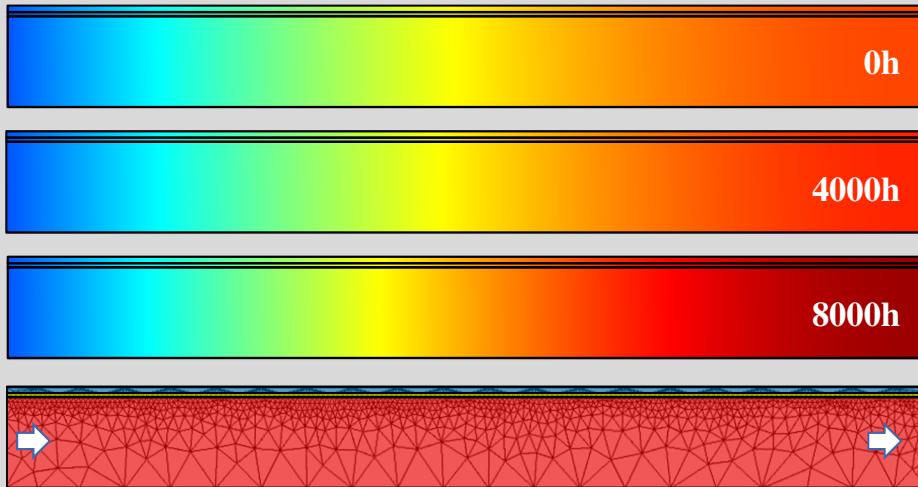
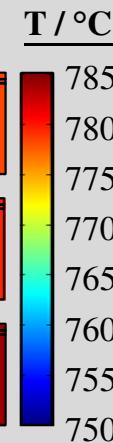
- stack width: 129mm
- stack length: 65mm
- channel height: 0.34mm

operation conditions (co-flow)

- 0.3 A/cm² (f.u. 36%)
- fuel: H₂/N₂ (50/50) + 3% H₂O, 1l/m
- oxidant: air (21% O₂), 2l/min
- temperature: 750°C



spatial distribution of the operating conditions

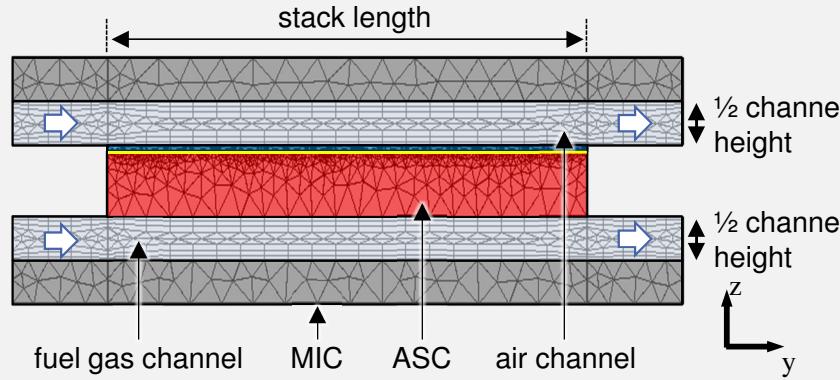


➤ T_{max} and T-gradient increases

Modelling V

Stack Durability Model

2D FEM model

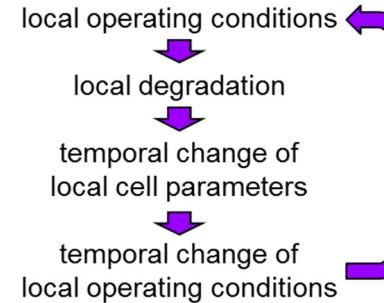


geometry

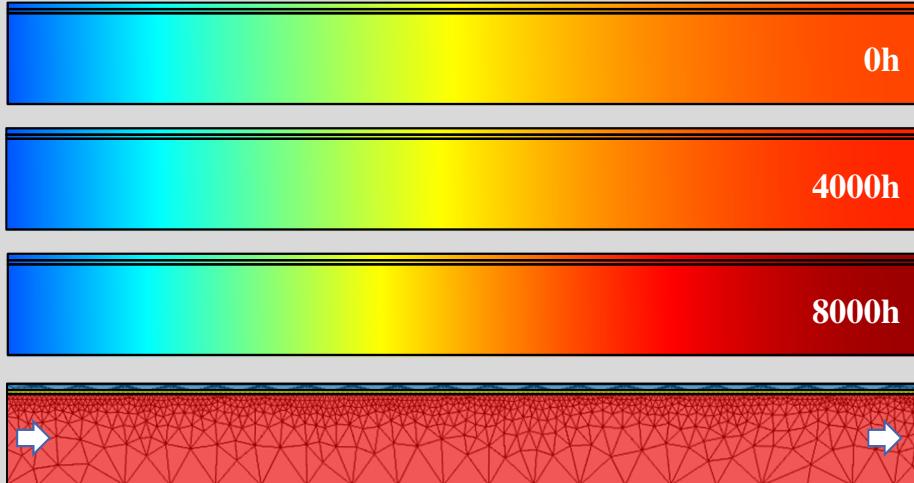
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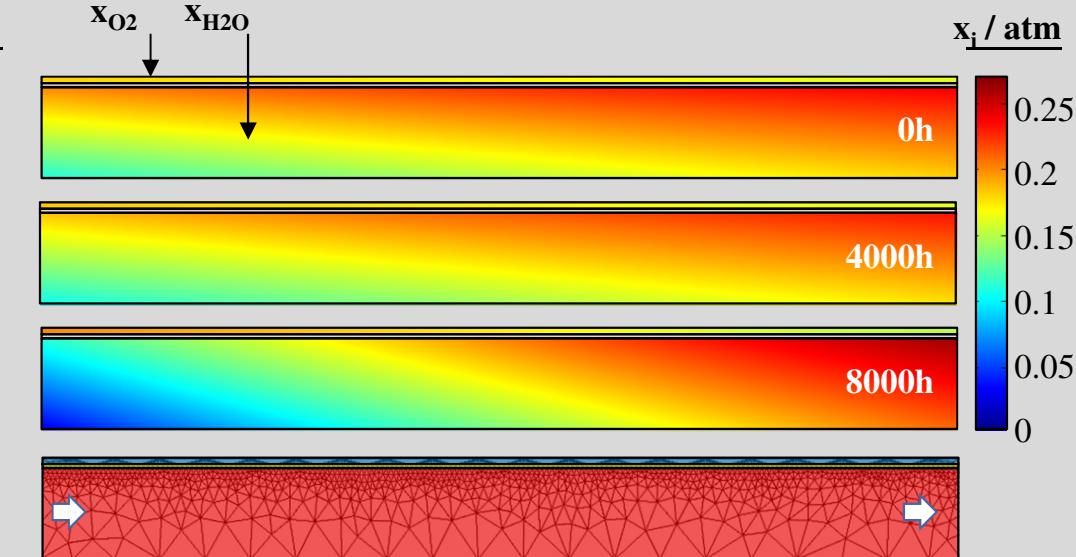


spatial distribution of the operating conditions



T / °C

785
780
775
770
765
760
755
750



x_i / atm

0.25
0.2
0.15
0.1
0.05
0

➤ T_{max} and T-gradient increases

➤ local changes of O₂- and H₂O-concentration